

Review

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# Ternary organic solar cells featuring polythiophene

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## Abstract

Benefiting from the creation of new photovoltaic materials and innovations in device architectures, organic photovoltaic (OPV) cells are booming. Nonetheless, their prosperity is also accompanied by challenges, such as tedious synthetic routes, increasing costs and insufficient operational stability under practical stresses. Polythiophene, with a simple chemical structure, high scalability and excellent charge transport ability, is expected to be the most promising candidate among all kinds of polymer donors. Ternary mixing, as a simple and effective method for improving the efficiency and stability of OPVs, has attracted significant attention in recent decades. This review provides an overview of the recent advances in ternary OPVs based on polythiophene and discusses the role of various third components in three types of OPV active layers, where polythiophene serves as either the host material or additive, and also clarifies how the third component plays a role in determining morphology and device performance, and finally proposes future research directions for ternary OPVs featuring polythiophene. In short, this review provides insights into polythiophene-based multicomponent systems and helps readers better understand the relationships between morphology, efficiency and stability.

**Keywords:** Ternary organic solar cells, third components, polythiophene, morphology, efficiency, stability



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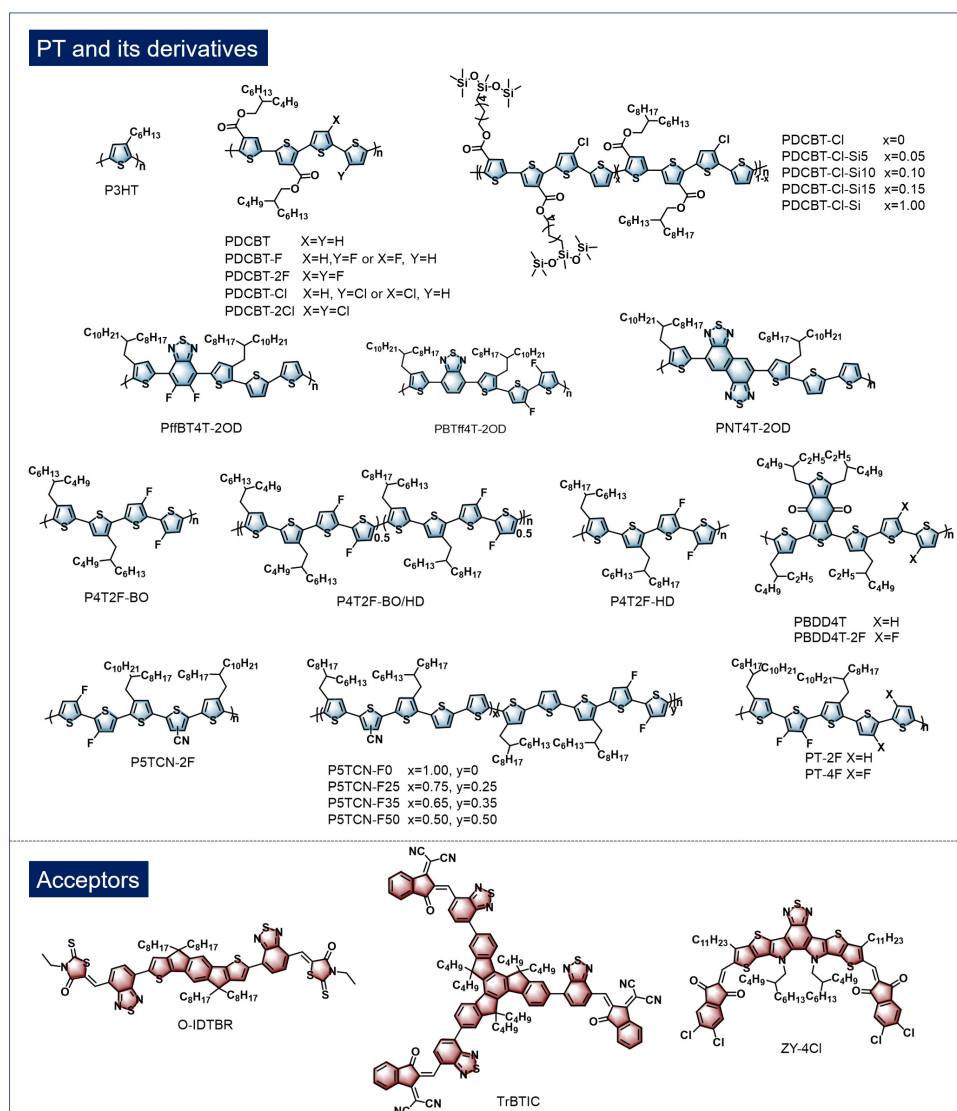


## INTRODUCTION

The increasing demand for energy sources and the continuous aggravation of environmental pollution compel us to seek alternative energy technologies to replace conventional fossil materials. Solar energy, as an inexhaustible and renewable energy source, is essential to this challenge. Organic solar cells (OSCs) demonstrate significant potential for the Internet of Things, electronic devices and portable and wearable sensors due to their merits of lightweight, flexibility and solution processibility<sup>[1-4]</sup>. Recent decades have witnessed rapid developments in OSCs. For instance, Wang *et al.* recently designed and synthesized three regioregular benzodithiophene-based polymer donors, namely, PBDTT, PBDTT1Cl and PBDTT2Cl<sup>[5]</sup>. The weak intramolecular charge transfer affords these polymers with large optical bandgaps. PBDTT1Cl-based OSCs gave the best device performance with an impressive power conversion efficiency (PCE) of 17.0%, which could be attributed to the ordered molecule stacking, balanced charge mobility and efficient charge transport. The PCE of single-junction OSCs has now exceeded 19%<sup>[6-9]</sup>, which is already approaching the commercialization application requirements. There is no doubt that the optimization of molecular structure and the improvement of device processing methods play critical roles<sup>[10-12]</sup>. However, several problems, such as more and more complicated molecular structures, tedious purification processes and increasing costs, also arise as a result. Herein, the design and synthesis of low-cost and high-efficiency photovoltaic materials are urgent for the development of OSCs.

Polythiophene (PT) has always been considered as the most promising polymer donor to be scaled up due to its straightforward synthesis, relatively high crystallinity and excellent batch reproducibility<sup>[13-17]</sup>. **Figure 1** summarizes the chemical structures of PT and its derivatives featured in this article. Among them, poly(3-alkylthiophene) (P3HT) is the most representative material. Acceptors, such as fullerene and its derivatives, have dominated the development of OSCs for a long time due to their advantages of relatively high electron affinity, excellent charge transfer characteristics and good compatibility with polymer donors<sup>[18]</sup>. As the classic blend system, the photovoltaic properties of P3HT:fullerene have been systemically investigated. Nonetheless, the efficiency of P3HT and fullerene is much lower than that of donor-acceptor-based conjugated polymers because of the intrinsic excessively high-lying energy levels of P3HT and the weak absorption in the visible region of fullerene derivatives<sup>[19]</sup>. Therefore, downshifting the energy levels of PT by introducing electron-withdrawing substituents and developing acceptors with matched energy levels, complementary absorption and excellent compatibility is critical for promoting the development of PT-based OSCs.

The emergence of non-fullerene acceptors has revived PT-based OSCs. Holliday *et al.* developed a pair of non-fullerene acceptors named O-IDTBR and O-IDFBR in 2016, boosting the efficiency of OSCs based on PT to 6.4%<sup>[20]</sup>. In 2019, Xu and coworkers<sup>[21]</sup> designed a novel acceptor matching P3HT, namely, TrBTIC. TrBTIC exhibited excellent solubility in common solvents. They found that the phase separation degree could be well controlled by changing the aging time of P3HT in a green solvent, namely, 1,2,4-trimethylbenzene. Consequently, the device performance was improved from 6.62% to 8.25% after aging for 40 min. The efficiency of PT-based OSCs has been refreshed again since the development of a new Y-series acceptor known as ZY-4Cl<sup>[22]</sup>. Compared with BTP-4Cl, ZY-4Cl exhibited appropriate phase separation due to the limited miscibility with P3HT. As a result, devices based on P3HT:ZY-4Cl achieved a high efficiency of 9.46% with an improved open-circuit voltage ( $V_{oc}$ ) of 0.88 V, a short-circuit current density ( $J_{sc}$ ) of 16.49 mA/cm<sup>2</sup> and fill factor (FF) of 0.65. Recently, our group demonstrated that the crystalline order of these photovoltaic materials could be manipulated by varying the annealing time<sup>[23]</sup>. Thus, the P3HT:ZY-4Cl blend gave a significantly improved PCE of 10.7% after annealing at 130 °C for 30 s.



**Figure 1.** Chemical structures of PT, PT derivatives and non-fullerene acceptors featured in this review.

PT derivatives demonstrate excellent potential for realizing low-cost and highly efficient OSCs<sup>[24-28]</sup>. An alkoxy-carbonyl-functionalized PT derivative known as PDCBT was developed by Zhang *et al.*<sup>[29]</sup>, which exhibited better performance compared with P3HT, either paired with fullerene or non-fullerene acceptors<sup>[30]</sup>. Since then, a series of studies focusing on PT derivatives have been carried out, contributing to the in-depth understanding of the molecular design and device processing technologies of PT-based OSCs<sup>[31-34]</sup>. Jia and colleagues<sup>[35]</sup> reported a series of fluorinated PT derivatives (P4T2F-BO, P4T2F-BO/HD and P4T2F-HD) in 2019. Among all the three polymers, P4T2F-HD gave the highest PCEs of 6.3% and 7.0% when combined with PC<sub>70</sub>BM and O-IDTBR, respectively. Shortly afterwards, the authors systematically investigated the effect of backbone fluorination degree on the photovoltaic performance of PT-based OSCs<sup>[36]</sup>. P6T-F100, with the highest fluorination degree, demonstrated enhanced exciton dissociation efficiency and reduced charge recombination and gave the best device performance. Yuan and coworkers<sup>[37]</sup> constructed a series of high-efficiency OSCs based on PT by designing a cyano group-substituted P5TCN-2F paired with Y-series acceptors. The incorporation of the cyano group not only deepened the energy levels of PT but also enhanced the interchain interaction and polymer crystallinity. As a result, the P5TCN-

2F:Y6-based device achieved a champion PCE of 16.1%, which was much higher than those of previously reported PT-based OSCs. There is no doubt that these successful explorations have laid a solid foundation for the practical application of OSCs.

To further increase the efficiency and improve the stability of OSCs, various methods have been proposed<sup>[38,39]</sup>. Ternary OSCs have attracted increasing attention in the past decade due to their simple device fabrication process<sup>[40-42]</sup>. The third component in the form of quantum dots, dye molecules<sup>[43]</sup>, insulating polymers<sup>[44]</sup>, polymer/small molecule donors<sup>[45]</sup> and fullerene/non-fullerene acceptors<sup>[46]</sup> is incorporated into the binary systems to form complementary absorption, matched energy levels with host photovoltaic materials or acquire appropriate morphology. Recent advances in ternary OSCs based on non-fullerene acceptors have been clarified, while reviews focusing on PT-based ternary OSCs are still deficient. Therefore, this review aims to summarize the up-to-date advances of PT-based ternary OSCs with the category of application of third components, including quantum dots, insulating polymers, small molecules, fullerene derivatives and polymer donors in fullerene, non-fullerene and polymer acceptors systems. Furthermore, the role of the third component in different systems is also stressed from the perspective of light harvesting, energy transfer, charge transfer and morphology control<sup>[47,48]</sup>. [Figure 2](#) summarizes the recent advances in polythiophene-based ternary OSCs and the content of this review is depicted in [Figure 3](#).

## PT:FULLERENE-BASED TERNARY OSCS

### PT:quantum dot:fullerene systems

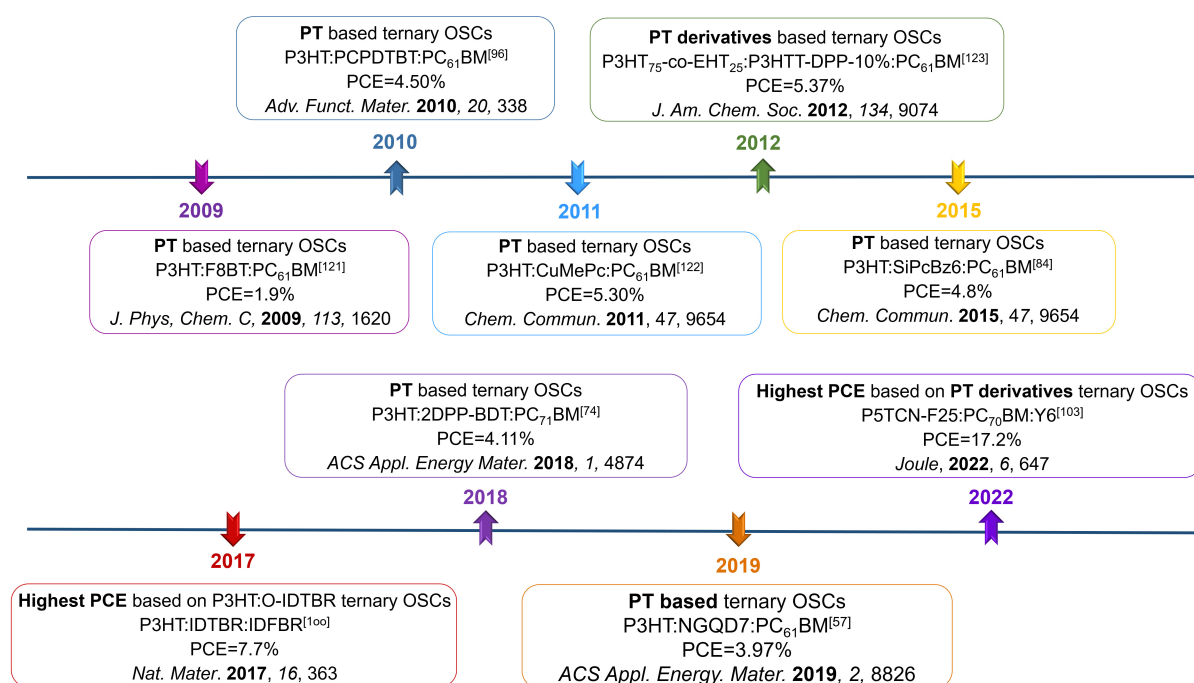
Quantum dot (QD) solar cells have attracted increasing attention in recent decades due to their advantages of low cost, tunable energy levels and strong light absorption capacity<sup>[49]</sup>. Research has demonstrated that QDs can effectively extend light absorption and improve the charge transfer of OSCs, contributing to enhanced device performance. The photovoltaic parameters of PT:fullerene-based OSCs with and without QDs are summarized in [Table 1](#). Park and coworkers<sup>[50]</sup> investigated the effect of ligand type on the performance of P3HT:PC<sub>61</sub>BM:CdSe-based ternary OSCs by introducing two CdSe nanoparticles capped with different ligands, namely, pyridine and fatty acid-oleic acid, into the active layers. Both CdSe nanoparticles had positive effects on the efficiency of OSCs but with different mechanisms. The aggregation of pyridine CdSe nanoparticles in the active layer provided enhanced charge transport pathways, leading to an increased  $V_{oc}$ . The morphology of P3HT:PC<sub>61</sub>BM, however, was almost unchanged after incorporating CdSe nanoparticles capped with the small-sized oleic acid. The enhanced PCE of P3HT:PC<sub>61</sub>BM:CdSe capped with oleic acid was ascribed to the increased  $J_{sc}$ . In addition to the common semiconductor nanoparticles, such as PbS and CdSe, Si nanoparticles also can be applied in OSCs as the third component. Zhao *et al.* improved the efficiency of P3HT:PC<sub>61</sub>BM by a factor of 1.4 with 5% PC<sub>61</sub>BM replaced by Si nanoparticles<sup>[51]</sup>. Nonetheless, ternary OSCs with P3HT partially being substituted demonstrated a decreased PCE. The difference originated from the better absorption of Si nanoparticles in the short wavelength region compared with PC<sub>61</sub>BM.

Fan and coworkers<sup>[52]</sup> demonstrated that the incorporation of 2 mg/mL of carbon QDs provided additional carriers at the donor-acceptor interface and a further increase in the carbon QD content resulted in enhanced defects in the active layer. Compared with fullerene and its derivatives, QDs exhibit broader spectral absorption and more appropriate energy levels. Substituting fullerenes with small amounts of QDs contributes to enhanced spectral response and reduced charge recombination, leading to increased device performance. Despite these positive effects, the complex purification process and the relatively low yield of QDs restrict their large-scale application in OSCs.

**Table 1. Photovoltaic parameters of PT:fullerene-based OSCs with and without QDs**

System	Binary system				quantum dot	Ternary system				Ref.
	$V_{oc}$ (V)	$J_{sc}$ (mA cm <sup>-2</sup> )	FF	PCE (%)		$V_{oc}$ (V)	$J_{sc}$ (mA cm <sup>-2</sup> )	FF	PCE (%)	
P3HT:PC <sub>61</sub> BM	0.62	7.35	0.59	2.74	CdSe	0.64	7.50	0.61	2.96	[50]
P3HT:PC <sub>61</sub> BM	0.59	9.14	0.54	2.93	Si NCs	0.57	16.15	0.47	4.11	[51]
P3HT:PC <sub>61</sub> BM	0.53	14.90	0.32	2.50	CQDs	0.55	18.90	0.32	3.32	[52]
P3HT:PC <sub>61</sub> BM	0.36	6.31	0.34	0.78	PbS	0.45	7.88	0.32	1.15	[53]
P3HT:PC <sub>61</sub> BM	/	9.30	0.55	3.08	BaTiO <sub>3</sub>	/	9.55	0.56	3.27	[54]
P3HT:PC <sub>70</sub> BM	0.61	7.69	0.51	2.39	PbSSe	0.55	9.26	0.48	2.40	[55]
P3HT:PC <sub>61</sub> BM	0.56	4.56	0.51	1.28	CdSe	0.55	5.21	0.53	1.53	[56]
P3HT:PC <sub>61</sub> BM	0.57	10.30	0.52	3.09	NGQD7	0.57	11.60	0.58	3.79	[57]
					NGQD8	0.57	11.40	0.57	3.69	
					rGQD	0.57	11.00	0.56	3.52	
P3HT:PC <sub>60</sub> BM	0.57	8.77	0.50	2.51	CrO <sub>3</sub>	0.59	9.94	0.63	3.67	[58]
P3HT:PC <sub>61</sub> BM	0.59	9.51	0.53	2.96	GQDs	0.60	12.31	0.56	4.13	[59]
P3HT:PC <sub>61</sub> BM	0.58	3.47	0.40	0.81	NCs-EHT	0.52	6.34	0.48	1.60	[60]

OSCs: Organic solar cells; QDs: quantum dots; PCE: power conversion efficiency.



**Figure 2.** Development of polythiophene-based ternary OSCs. Milestone studies are marked in the boxes<sup>[121-123]</sup>.

### PT:insulating polymer:fullerene systems

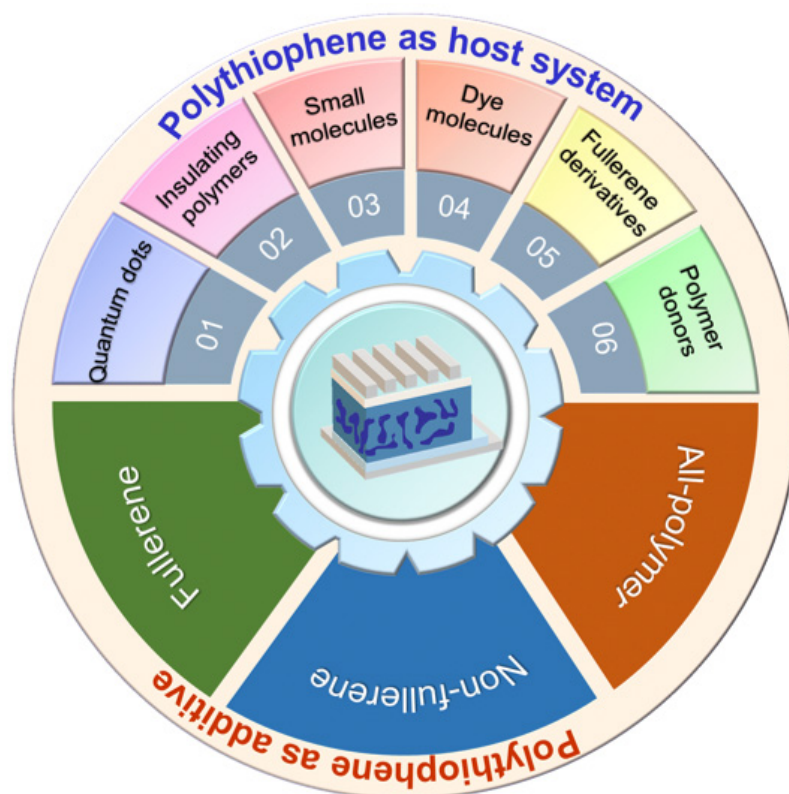
Insulating polymers, such as commodity plastics, elastomers and resins, have been introduced into OSCs to improve their efficiency, stability and flexibility by controlling the film morphology or forming buffer layer by self-organization. Table 2 summarizes the photovoltaic parameters of PT:fullerene-based OSCs with and without insulating polymers. Ferenczi and coworkers<sup>[68]</sup> demonstrated that the incorporation of semicrystalline insulators, such as high-density polyethylene (HDPE) and isotactic polystyrene (i-PS), was favorable for the formation of interpenetrated networks under lower fractions, leading to an increased  $J_{sc}$ . OSCs with 50% HDPE or i-PS achieved comparable efficiency but increased the active layer thickness,



**Table 2. Photovoltaic parameters of PT:fullerene-based OSCs with and without insulating polymers**

System	Binary system				Insulating polymer	Ternary system				Ref.
	$V_{oc}$ (V)	$J_{sc}$ ( $\text{mA cm}^{-2}$ )	FF	PCE (%)		$V_{oc}$ (V)	$J_{sc}$ ( $\text{mA cm}^{-2}$ )	FF	PCE (%)	
P3HT: PC <sub>61</sub> BM	0.39	4.01	0.50	0.78	PMMA	0.39	5.40	0.52	1.10	[61]
PfFBT4T: PC <sub>61</sub> BM	0.75	17.00	0.73	9.24	TP-C <sub>6</sub>	0.75	18.40	0.72	9.41	[62]
P3HT: PC <sub>61</sub> BM	0.59	12.12	0.56	3.97	PEG	0.59	12.67	0.57	4.24	[63]
P3HT: PC <sub>61</sub> BM	0.61	9.92	0.50	3.03	PVP	0.61	13.04	0.49	3.90	[64]
P3HT: PC <sub>61</sub> BM	0.59	8.80	0.56	2.90	P3HT-b-P3FAT	0.60	11.30	0.65	4.40	[65]
P3HT: PC <sub>61</sub> BM	0.56	7.99	0.54	2.40	PVDF	0.56	8.63	0.63	3.07	[66]
P3HT: PC <sub>61</sub> BM	0.57	8.31	0.65	3.05	PDMS-b-PMMA	0.61	8.68	0.67	3.56	[67]

OSCs: Organic solar cells; PCE: power conversion efficiency.



**Figure 3.** Structure of this review.

which facilitated the large-scale preparation of OSCs. Dauzon and colleagues<sup>[69]</sup> systematically investigated the effect of two approaches, namely, crosslinking and blending, on the mechanical properties of OSCs. Three crosslinkers (C<sub>12</sub>N<sub>3</sub>, PEG<sub>3</sub>N<sub>3</sub> and PEG<sub>n</sub>N<sub>3</sub>) with different polarities and chain lengths and two elastomers with different net types, i.e., polystyrene-block-poly(ethylene-ran-butylene)-block-polystyrene (SEBS) and poly(dimethylsiloxane) (PDMS) were incorporated into the P3HT:PC<sub>61</sub>BM blends. The bis-azide molecules reacted selectively with PC<sub>61</sub>BM and suppressed the crystallization of acceptor under thermal stress, contributing to a uniform morphology. Nonetheless, blend films with a SEBS and PDMS weight ratio over 50% exhibited severe phase separation resulting from poor miscibility. Compared with ternary OSCs based on SEBS and PDMS, the efficiency of bis-azide molecules crosslinked devices could be well retained due to the enhanced aggregation of P3HT. Active layers crosslinked with bis-azide molecules

or blended with SEBS demonstrated improved mechanical properties originating from the softening effect induced by the plasticization or the low interaction. Notably, the application of SEBS could be extended to non-fullerene blends, demonstrating excellent potential in the field of flexible electronic devices.

In addition to forming uniform three-phase blend systems, functionalized insulating polymers incorporated into the active layers can also act as the buffer layer due to the surface energy or chemical interaction. Yamakawa *et al.* developed a simple approach to improve the efficiency of P3HT:PC<sub>61</sub>BM with a low surface energy insulator poly(dimethyl-siloxane)-block-poly(methyl methacrylate) (PDMS-b-PMMA)<sup>[67]</sup>. The PDMS-b-PMMA segregated at the active layer, self-organized to the buffer layer during the spin-coating process and suppressed the charge recombination between active layer and electrode, with the PCE of P3HT:PC<sub>61</sub>BM increasing from 3.05% to 3.56%. Yu and coworkers<sup>[66]</sup> improved the device performance of P3HT:PC<sub>61</sub>BM by employing a nonvolatile processing additive polyvinylidene fluoride (PVDF). The addition of PVDF improved the absorption in the range of 300-600 nm and enhanced the phase separation of active layer, resulting in improved light-harvesting capacity and interfacial contact. Insulating polymers demonstrate excellent potential for improving the efficiency, thermal stability and mechanical properties of OSCs and the incorporation of commercialized insulating polymers can also reduce the cost of device preparation. Some studies have reported that insulating polymers with a glass transition temperature ( $T_g$ ) lower than the annealing temperature during device fabrication facilitate the enhancement of PCE since the film morphology can be easily manipulated, while insulating polymers with  $T_g$  over the annealing temperature generally have a negative influence on the device performance. Nevertheless, circumstances that employ high-temperature resistant resins as the third component to improve the efficiency of OSCs are not uncommon. Therefore, clarification of the mechanism of insulating polymers affecting the efficiency of OSCs and the construction of the selection principle of the third component are urgently required.

#### **PT:small molecule:fullerene systems**

The application of small molecules, including small molecule donors, non-fullerene acceptors and dye molecules, in polythiophene OSCs is featured in this section. Table 3 summarizes the photovoltaic parameters of PT:fullerene-based OSCs with and without conjugated small molecules. Zhang *et al.* systemically investigated the effect of crystallinity of non-fullerene acceptors on the microstructure and photovoltaic performance of PT:fullerene-based OSCs<sup>[70]</sup>. Two non-fullerene acceptors with different crystallinity, namely, EH-OIDTBR and O-IDTBR, were incorporated into the P3HT:PC<sub>71</sub>BM blend separately. Ternary blends with both EH-OIDTBR and O-IDTBR demonstrated extended absorption band and enhanced absorption intensity at 602 nm, corresponding to the improved  $\pi$ - $\pi$  interaction of P3HT. Furthermore, the addition of EH-OIDTBR and O-IDTBR provided more charge transport pathways. Compared with the P3HT:PC<sub>71</sub>BM:EH-OIDTBR ternary blend, the P3HT:PC<sub>71</sub>BM:EH-OIDTBR-based system exhibited smoother morphology, which originated from the more ordered structure of O-IDTBR. Furthermore, the coherent length of P3HT increased with increasing O-IDTBR weight ratio. Both ternary OSCs based on EH-OIDTBR and O-IDTBR achieved an improved average efficiency (4.05% and 4.01%, respectively), which could be attributed to the increased FF and  $J_{sc}$ , respectively. Accordingly, the enhanced charge transport efficiency and more balanced electron and hole mobility were responsible for the increased  $J_{sc}$  and FF.

Apart from small molecular donors and non-fullerene acceptors, dye molecules are another important component for improving the light-harvesting capacity. The device performances of PT:fullerene-based OSCs with and without dye molecules are listed in Table 4. Honda and coworkers<sup>[79]</sup> reported a ternary OSC based on P3HT:PCBM:SiPc with enhanced device performance. The ternary blend films exhibited similar absorption before and after annealing, with the maximum absorption peaks at ~460 and ~670 nm, indicating the identical aggregation behavior of P3HT. As expected, the ternary OSCs achieved a PCE of

**Table 3. Photovoltaic parameters of PT:fullerene-based OSCs with and without conjugated small molecules**

System	Binary system				Conjugated small molecule	Ternary system				Ref.
	$V_{oc}$ (V)	$J_{sc}$ (mA cm <sup>-2</sup> )	FF	PCE (%)		$V_{oc}$ (V)	$J_{sc}$ (mA cm <sup>-2</sup> )	FF	PCE (%)	
P3HT:PC <sub>71</sub> BM	0.60	9.10	0.60	3.30	EH-IDTBR	0.59	10.86	0.63	4.05	[70]
					O-IDTBR	0.59	10.93	0.62	4.01	
P3HT:PC <sub>61</sub> BM	0.62	8.91	0.68	3.77	TT-TTPA	0.62	10.30	0.67	4.26	[71]
P3HT:PC <sub>71</sub> BM	0.57	8.47	0.62	3.00	p-DTS(FBTTH <sub>2</sub> ) <sub>2</sub>	0.62	9.35	0.64	3.71	[72]
P3HT:PC <sub>61</sub> BM	0.56	7.54	0.67	2.84	SMA1	0.65	8.00	0.70	3.70	[73]
P3HT:PC <sub>71</sub> BM	0.57	8.80	0.59	2.98	2DPP-BDT	0.59	11.90	0.55	3.80	[74]
P3HT:PC <sub>61</sub> BM	0.53	10.50	0.54	3.00	DPP4T-Cz	0.56	12.90	0.56	4.04	[75]
P3HT:PC <sub>61</sub> BM	0.55	7.05	0.66	2.58	BTT-OT-ORD	0.57	8.25	0.67	3.13	[76]
P3HT:PC <sub>61</sub> BM	0.53	7.53	0.61	2.45	BTT-OT-OTZDM	0.56	7.45	0.62	2.65	
P3HT:PC <sub>61</sub> BM	0.64	7.03	0.46	2.07	p-DPP-PhCN	0.66	7.30	0.51	2.48	[77]
P3HT:PC <sub>61</sub> BM	0.58	10.40	0.57	3.44	DH4T	0.60	11.21	0.62	4.17	[78]

OSCs: Organic solar cells; PCE: power conversion efficiency.

**Table 4. Photovoltaic parameters of PT:fullerene-based OSCs with and without dye molecules**

System	Binary system				Dye molecule	Ternary system				Ref.
	$V_{oc}$ (V)	$J_{sc}$ (mA cm <sup>-2</sup> )	FF	PCE (%)		$V_{oc}$ (V)	$J_{sc}$ (mA cm <sup>-2</sup> )	FF	PCE (%)	
P3HT:PC <sub>61</sub> BM	0.58	6.50	0.59	2.20	ZnPc	0.48	6.20	0.37	1.10	[79]
					SiPc	0.58	7.90	0.59	2.70	
P3HT:PC <sub>61</sub> BM	0.55	8.96	0.71	3.50	SiPc:SiNc	0.57	10.90	0.69	4.30	[80]
P3HT:PC <sub>61</sub> BM	0.58	7.00	0.65	2.64	Aza-BODIPY	0.56	7.85	0.65	2.81	[81]
P3HT:PC <sub>61</sub> BM	0.52	4.93	0.40	1.02	CuPc	0.55	5.70	0.46	1.44	[82]
P3HT:N2200	0.45	3.50	0.52	0.82	SiNc10	0.50	5.70	0.50	1.40	[83]
					SiNc6	0.50	4.50	0.54	1.20	
P3HT:PC <sub>61</sub> BM	0.55	10.70	0.64	3.80	SiPcBz6	0.58	13.10	0.60	4.40	[84]
P3HT:PC <sub>61</sub> BM	0.58	7.80	0.51	2.30	VOPcPhO	0.56	13.50	0.45	3.40	[85]
P3HT:PC <sub>61</sub> BM	0.60	6.80	0.70	3.60	Zn-TTBP	0.61	8.10	0.70	4.30	[86]
P3HT:PC <sub>61</sub> BM	0.54	10.52	0.60	3.41	(HxN <sub>3</sub> ) <sub>2</sub> -SiPc	0.58	11.42	0.60	3.58	[87]
P3HT:PC <sub>61</sub> BM	0.62	4.97	0.35	0.78	ZnPc-1py	0.45	4.97	0.35	0.78	[88]
P3HT:PC <sub>61</sub> BM	0.53	10.06	0.53	2.84	(3HS)2-SnPc	0.49	9.82	0.50	2.43	[89]
P3HT:PC <sub>61</sub> BM	0.63	6.77	0.52	2.22	BODIPY 1	0.67	9.45	0.54	3.43	[90]
P3HT:PC <sub>61</sub> BM	0.49	7.28	0.40	1.41	Ru(N-P)2 (O-O)	0.56	7.60	0.50	2.12	[91]
P3HT:PC <sub>61</sub> BM	0.56	8.22	0.59	2.73	SiPc-Py-1	0.62	9.84	0.61	3.75	[92]
					SiPc-Py-2	0.62	9.91	0.62	3.79	
					SiPc-Py-3	0.62	9.99	0.61	3.80	
					SiPc-Py-4	0.62	10.29	0.65	4.13	

OSCs: Organic solar cells; PCE: power conversion efficiency.

2.7% with an increased  $J_{sc}$  of 7.9 mA/cm<sup>2</sup> and a comparable FF of 0.59. They attributed the improved  $J_{sc}$  to the enhanced photocurrent promoted by SiPc.

Part of the photocurrent originated from the photoexcitation of SiPc itself and the rest was ascribed to the increased charge separation induced by SiPc. Notably, only SiPc located at the interface of P3HT:PCBM could contribute to photocurrent generation, which could also be a principle for the selection of photovoltaic materials.



### PT:fullerene derivative:fullerene systems

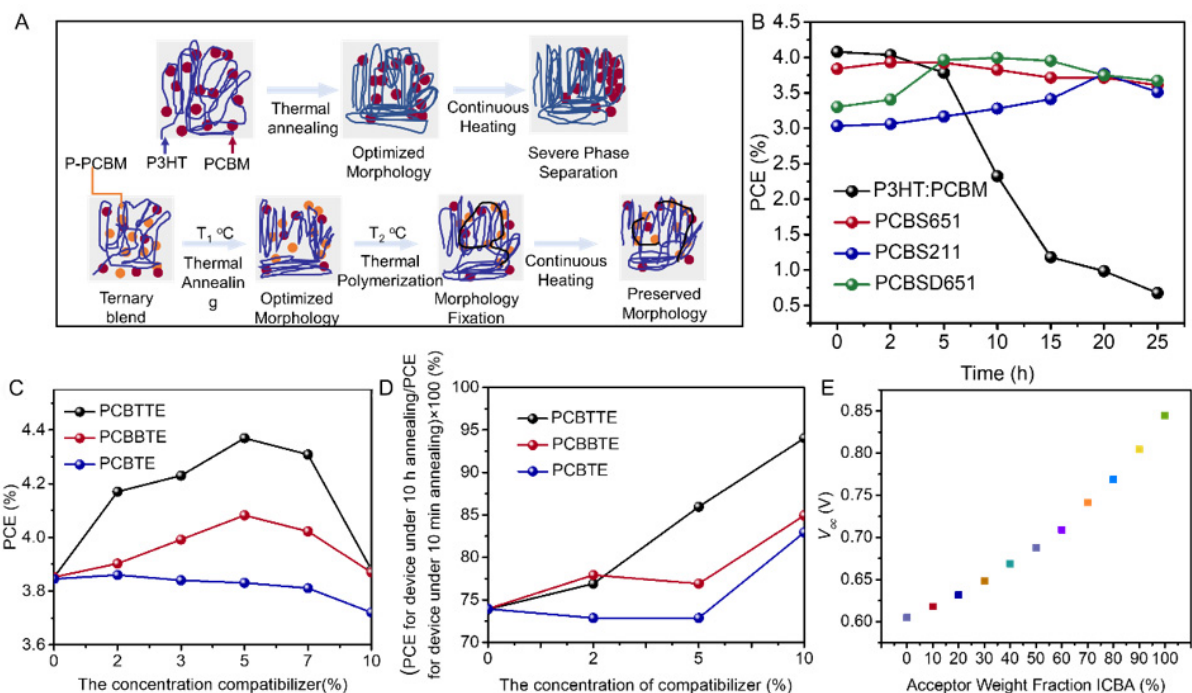
Cheng and coworkers<sup>[93]</sup> constructed efficient and thermal stable PT-fullerene ternary OSCs via the in situ polymerization of fullerene derivatives. Two fullerene derivatives (PCBSD and PCBS) with different styrene group contents were incorporated into P3HT:PC<sub>61</sub>BM to manipulate the film morphology. As depicted in [Figure 4A](#), an optimized morphology was acquired after annealing at 110 °C for 10 min and the second annealing stage triggered the in situ polymerization. As shown in [Figure 4B](#), the initial efficiencies of the control P3HT:PC<sub>61</sub>BM device and P3HT:PC<sub>61</sub>BM:PCBSD (6:5:1) and P3HT:PC<sub>61</sub>BM:PCBS (6:5:1) ternary OSCs were 4.08%, 3.32% and 3.84%, respectively.

The PCE of the P3HT:PC<sub>61</sub>BM binary system dropped to 0.69% after continuous heating for 25 h, while the efficiencies of all the ternary OSCs were well retained or even increased. It was discovered that the amorphous PCBS and PCBSD reduced the crystallinity and restricted the mobility of PC<sub>61</sub>BM and suppressed the aggregation of the acceptor under thermal stress. Furthermore, the crystallization behavior of P3HT was almost unchanged during the first stage of annealing, while the order of P3HT was significantly reduced after annealing at 150 °C.

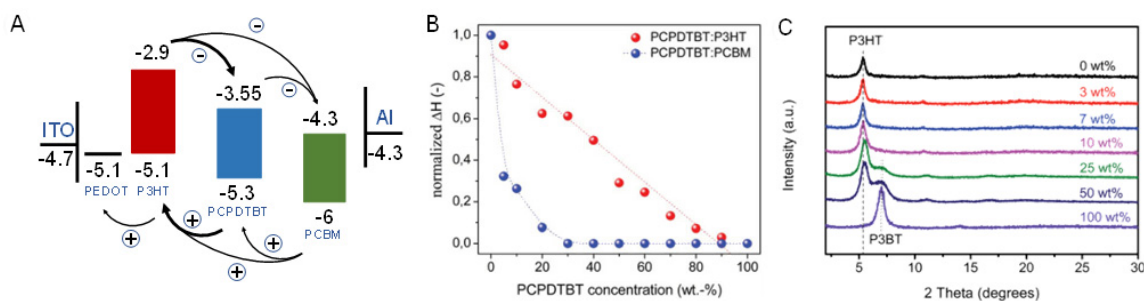
Lai *et al.* synthesized a series of PC<sub>61</sub>BM derivatives containing different thiophene units, namely, PCBTE, PCBBTE and PCBTTE, acting as the surfactants for OSCs<sup>[94]</sup>. Compared with the control P3HT:PC<sub>61</sub>BM, the ternary blends exhibited reduced surface energy and enhanced interfacial adhesion. The champion PCE of 4.37% was achieved by ternary OSCs with 5% PCBTTE content, as shown in [Figure 4C](#). The improved efficiency could be ascribed to the enhanced absorption coefficient originating from the increased order of the P3HT polymer chains. Moreover, ternary OSCs with a 10% PCBTTE content maintained 94% of their initial efficiency after annealing at 150 °C for 10 h, which was much higher than that of the binary system (74%) [[Figure 4D](#)]. The authors attributed the improved thermal stability to the suppressed PC<sub>61</sub>BM aggregation. Apart from functionalized fullerene derivatives, ICBA is another widely used fullerene acceptor. Khlyabich *et al.* demonstrated that the  $V_{oc}$  of P3HT:PC<sub>61</sub>BM:ICBA was not determined by the smallest  $V_{oc}$  of the corresponding binary systems but could be tuned by varying the weight ratio of the two acceptors<sup>[95]</sup>. The PCEs of the P3HT:PC<sub>61</sub>BM and P3HT:ICBA binary systems were 3.57% and 3.98%, respectively, and the  $V_{oc}$  of the ternary system increased from 0.605 to 0.844 V, with the ICBA content increased from 0 to 1, as shown in [Figure 4E](#). In addition, all the blend systems achieved comparable FFs, illustrating the balanced charge separation and transport.

### PT:donor polymer:fullerene systems

Low bandgap polymers with extended absorption to the near-infrared region are also promising candidates for promoting photocurrent generation. Koppe *et al.* investigated the mechanism of enhanced photosensitivity using a series of characterization methods<sup>[96]</sup>. They found that the absorption of the blend extended with increasing PCPDTBT fraction, while the film morphology was almost unchanged after annealing. Ternary OSCs with 20% PCPDTBT gave the highest PCE. Neither the energy transfer nor the charge transfer was responsible for the increased  $J_{sc}$ . The photoinduced charge transfer from PCPDTBT to P3HT and from PCPDTBT to PC<sub>61</sub>BM via the P3HT and PC<sub>61</sub>BM matrices, respectively, was the dominant process, as shown in [Figure 5A](#). Li and coworkers<sup>[97]</sup> investigated the phase diagram of binary and ternary OSCs based on P3HT:PC<sub>61</sub>BM and P3HT:PCPDTBT:PC<sub>61</sub>BM, respectively, by DSC and constructed a thermal-electrical property relationship. Small amounts of (less than 20%) PCPDTBT had no obvious effect on the P3HT:PC<sub>61</sub>BM phase diagram, while higher contents of PCPDTBT led to decreased crystallinity. Ternary OSCs with the blending ratio around the eutectic of the phase diagram exhibited the highest  $J_{sc}$ . Using a combination of DSC, GIWAXS and SCLC analysis, Machui *et al.* further explored the effect of the low bandgap polymer PCPDTBT on the crystallization behavior of the P3HT:PC<sub>61</sub>BM blend<sup>[98]</sup>. As shown in [Figure 5B](#), the normalized enthalpy of the PCPDTBT:P3HT blend decreased linearly with increasing



**Figure 4.** (A) Morphological evolution of P3HT:PC<sub>61</sub>BM with and without PCBSD or PCBS under thermal stress. (B) PCE of P3HT:PC<sub>61</sub>BM with different PCBSD or PCBS fractions as a function of annealing time. Reproduced with permission. Copyright, 2011 John Wiley and Sons<sup>[93]</sup>. (C) PCE and (D) thermal stability of P3HT:PC<sub>61</sub>BM with different PCBTTE, PCBBTE and PCBTE contents. Reproduced with permission. Copyright, 2011 Elsevier<sup>[94]</sup>. (E) Variation tendency of Voc of P3HT:PC<sub>61</sub>BM with various ICDA contents. Reproduced with permission. Copyright, 2011 American Chemical Society<sup>[95]</sup>.



**Figure 5.** (A) Energy levels of P3HT, PC<sub>61</sub>BM and PCPDTBT. Reproduced with permission. Copyright, 2010 John Wiley and Sons<sup>[96]</sup>. (B) Normalized enthalpy of PCPDTBT:P3HT and PCPDTBT:PC<sub>61</sub>BM blends with various PCPDTBT contents. Reproduced with permission. Copyright, 2012 Royal Society of Chemistry<sup>[98]</sup>. (C) X-ray spectra of P3HT:PC<sub>61</sub>BM blends as a function of P3BT content. Reproduced with permission. Copyright, 2015 Royal Society of Chemistry<sup>[99]</sup>.

PCPDTBT content, indicating that PCPDTBT mainly affected the acceptor crystallinity rather than the P3HT ones. Devices with a PCPDTBT content of less than 20% exhibited slightly higher efficiency and a further increase in PCPDTBT content led to a decreased  $J_{sc}$  and FF, which was mainly due to the decreased crystallinity of PC<sub>61</sub>BM.

In addition to low bandgap polymers, P3BT nanowires were also incorporated into the P3HT:PC<sub>61</sub>BM blend to modulate the film morphology by Zhang and colleagues<sup>[99]</sup>. Compared with P3BT, the P3BT nanowires demonstrated higher crystallinity and a significantly interconnected structure. As expected, the intensity of the diffraction peaks increased with P3BT nanowire content increasing from 0 to 10%, as shown in

**Figure 5C.** Moreover, the intensity of the absorption peaks at 605 and 550 nm corresponding to the 0-0 and 0-1 transfers of P3HT, respectively, was significantly enhanced after introducing 10% P3BT nanowires. Consequently, a ternary blend with 7% P3BT nanowires achieved the highest efficiency of 4.2%, which could be ascribed to the increased charge transport pathways provided by the P3HT crystalline phase and PC<sub>61</sub>BM phase segregation.

### PT:NON-FULLERENE-BASED TERNARY OSCS

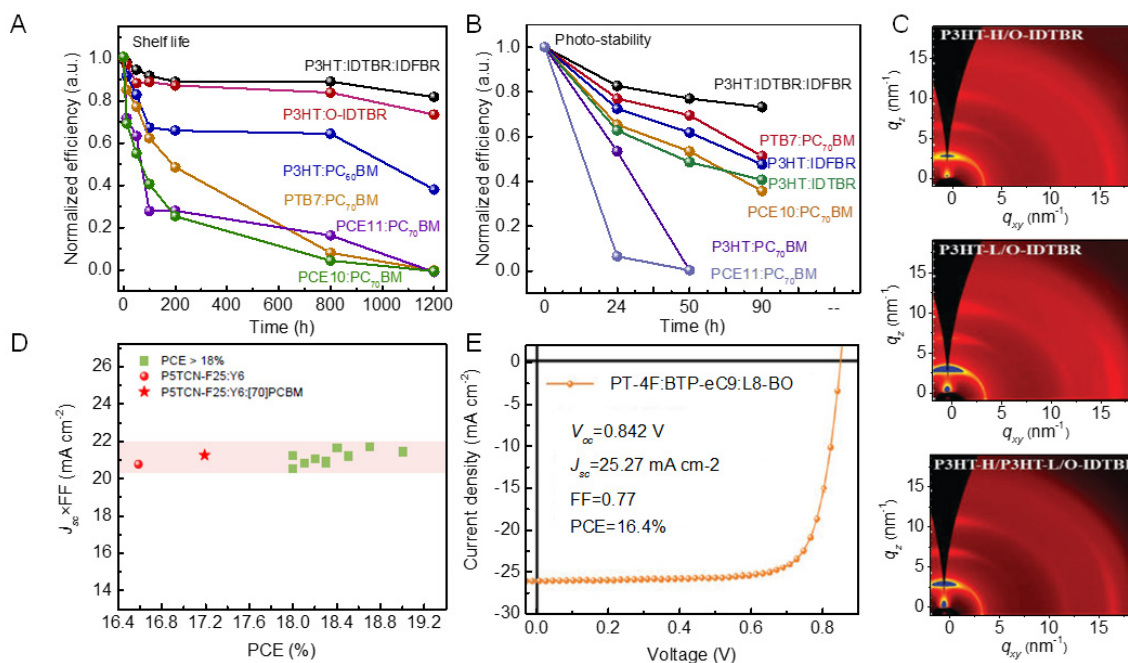
Non-fullerene acceptors play a critical role in promoting the efficiency of OSCs. Research based on PT:non-fullerene mainly focuses on the classical P3HT:O-IDTBR blend. Baran and coworkers<sup>[100]</sup> demonstrated highly efficient and thermal stable OSCs by optimizing the ratio of the two non-fullerene acceptors, O-IDTBR and O-IDFBR. The O-IDFBR dispersed in the crystalline P3HT and O-IDTBR phases was favorable for the preservation of the three-phase microstructure and the generation of photocurrent, contributing to increased device performance. Consequently, ternary OSCs based on P3HT:O-IDTBR:O-IDFBR gave a champion PCE of 7.7%. Furthermore, the efficiency of ternary OSCs could be well retained either in air or under illumination [Figure 6A and B].

Furthermore, improving the crystalline phase of the donor and acceptor is another effective method to facilitate charge transport. Liang *et al.*, for example, optimized the crystallinity and vertical phase separation of P3HT:O-IDTBR by separating the crystallization process with a high boiling point solvent (1,2,4-trichlorobenzene)<sup>[101]</sup>. Recently, Liang and colleagues<sup>[102]</sup> demonstrated a novel method to simultaneously extend the coherence lengths of both P3HT and O-IDTBR by mixing P3HT with different molecular weights. The low molecular weight P3HT relieved the entanglement of polymer chains, leading to a more ordered structure [Figure 6C]. Furthermore, the addition of lower molecular weight P3HT almost had no effect on the intermixed phase amount due to the excellent miscibility. As a result, the PCE of P3HT:O-IDTBR increased from 7.03% to 7.80%, which is the record value for P3HT:O-IDTBR blends.

Regardless of the various processing methods, the device performance of PT-based OSCs is still much lower than the theoretical value. Yuan and coworkers<sup>[103]</sup> reported a series of cyano group-substituted PTs with varied fluorination degrees named as P5TCN-Fx. Ternary OSCs based on P5TCN-F25:Y6:PC<sub>71</sub>BM achieved the highest PCE of 17.2% with a  $V_{oc}$  of 0.80 V, a  $J_{sc}$  of 27.55 mA/cm<sup>2</sup> and a FF of 0.777, as shown in Figure 6D. This is the present efficiency record for PT OSCs and represents a milestone in the field. Recently, Jeong *et al.* developed two fluorinated PT derivatives, PT-2F and PT-4F<sup>[104]</sup>. The highest occupied molecular orbital energy levels of PT were effectively downshifted due to the strong electron-withdrawing property of fluorine. Furthermore, the incorporation of fluorine enhanced the rigidity and planarity of the backbone, affording PT with improved crystallinity. Compared with PT-2F, PT-4F demonstrated an intense aggregation degree and distinct preaggregation behavior. Benefiting from the optimized morphology, the PT-4F-based devices demonstrated higher charge generation efficiency and the ternary OSCs based on PT-4F:BTP-eC9:L8-BO achieved a PCE of 16.4% with a  $V_{oc}$  of 0.842 V, a  $J_{sc}$  of 25.27 mA/cm<sup>2</sup> and a FF of 0.77, as shown in Figure 6E. Overall, PT derivatives demonstrate excellent potential for the realization of high-efficiency and low-cost OSCs.

### PT AS AN ADDITIVE FOR OSCS

In addition to the above circumstances, PT and its derivatives can also be incorporated into other systems as guest materials to manipulate phase separation and charge transport. For example, Lin *et al.* demonstrated an improved device performance of Si-PCPDTBT:PC<sub>71</sub>BM by introducing small amounts of P3HT<sup>[105]</sup>. They found that a blend with 1% P3HT had an enhanced absorption in the range of 375-575 nm, which was complementary to the Si-PCPDTBT:PC<sub>71</sub>BM blend. Furthermore, the addition of P3HT led to increased



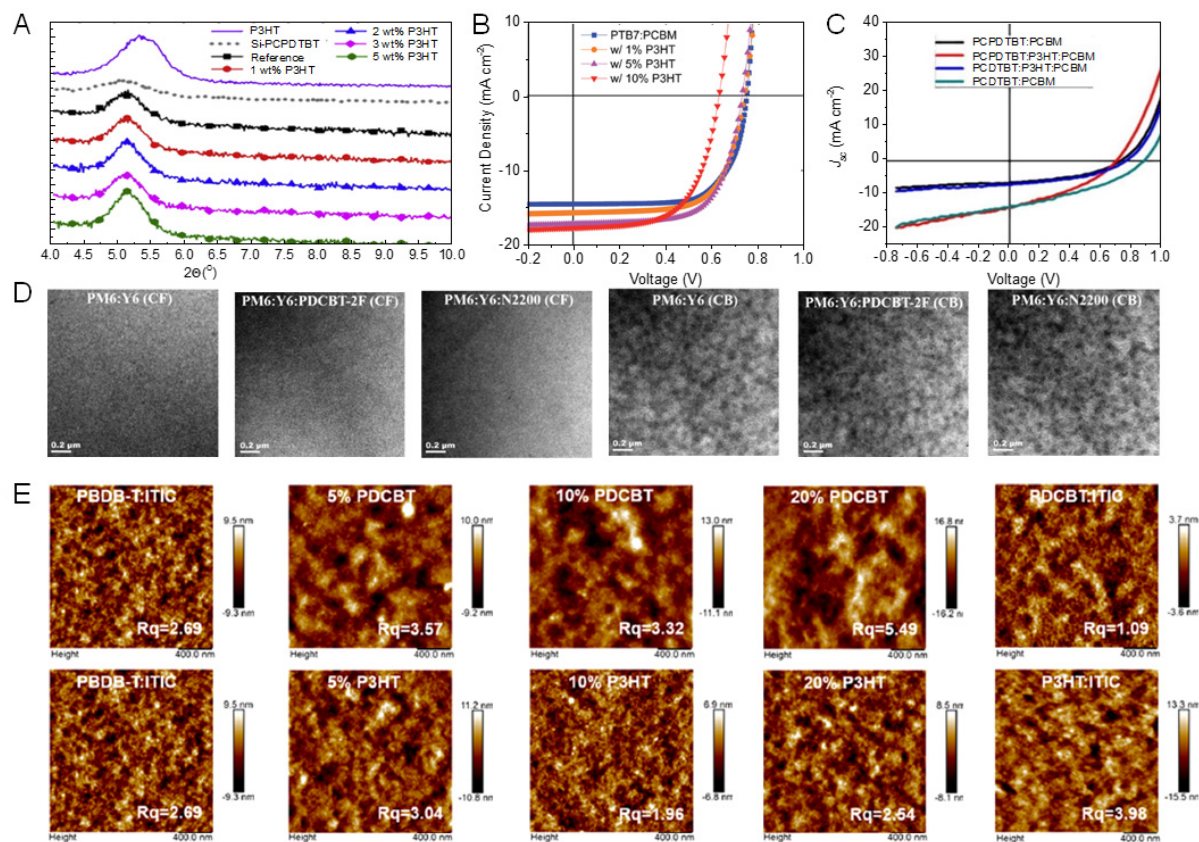
**Figure 6.** Normalized PCE of P3HT:IDTBR:IDFBR blend and other polymer:fullerene systems as a function of (A) storage and (B) illumination time. Reproduced with permission. Copyright, 2017 Macmillan Publishers Limited, part of Springer Nature<sup>[100]</sup>. (C) Two-dimensional grazing incidence wide-angle X-ray scattering (GIWAXS) patterns of P3HT-L:P3HT-H:O-IDTBR and ternary OSCs. Reproduced with permission. Copyright, 2021 John Wiley and Sons<sup>[102]</sup>. (D)  $J_{sc} \times FF$  as a function of PCE of OSCs based on P5TCN-F25 and other OSCs with PCE >18% in the literature. Reproduced with permission. Copyright, 2022 Elsevier<sup>[103]</sup>. (E) J-V characteristic curve of PT-4F:BTP-eC9:L8-BO. Reproduced with permission. Copyright, 2022 John Wiley and Sons<sup>[104]</sup>.

crystalline size and more significant phase separation [Figure 7A]. Despite the fact that the P3HT dispersed in the blend as the recombination centers led to increased charge recombination, the device performance of the control blend was finally increased due to the simultaneously improved  $J_{sc}$  and FF.

Studies have demonstrated that P3HT could replace conventional additives to control the film morphology and crystalline behavior of PTB7-based OSCs. Goh and coworkers<sup>[106]</sup> reported inverted OSCs based on P3HT:PTB7:PC<sub>71</sub>BM in 2015 for the first time. According to them, the influence resulting from the energy level differences between the two donors could be finely controlled by varying the blending ratios. The photodynamic, photovoltaic and microstructural properties of PTB7:PC<sub>71</sub>BM with a P3HT weight ratio in the range of 0%-10% were systemically investigated. The  $J_{sc}$  of the 5% ternary blend exhibited a remarkable increase due to the extended spectral response and the reduced recombination loss resulting from the enhanced Förster resonance energy transfer, as shown in Figure 7B. However, the FF of the ternary blend decreased rapidly with increasing P3HT weight ratio due to the distinct different orientations of the two donors. Solvent vapor annealing reduced the interfacial resistance, thereby improving the efficiency of the control device from 8.22% to 8.72%.

When acting as hole-cascade materials, PT and its derivatives can provide additional charge transport pathways via a bridging effect and therefore contribute to the improvement of device performance. Sivakumar *et al.* compared the performance of two PCDTBT:P3HT:PC<sub>61</sub>BM and PCPDTBT:P3HT:PC<sub>61</sub>BM blends from the perspective of crystalline behavior, optical properties and film morphology<sup>[107]</sup>. They found that the P3HT fibers in the PCPDTBT:PC<sub>61</sub>BM blend provided more pathways for charge transport, contributing to the enhanced device performance. For the PCDTBT:PC<sub>61</sub>BM blend, the addition of P3HT





**Figure 7.** (A) XRD patterns of P3HT and Si-PCPDTBT neat films and blends with various P3HT contents. Reproduced with permission. Copyright, 2014 Elsevier<sup>[105]</sup>. (B) J-V characteristic curves of PTB7:PC<sub>71</sub>BM with different P3HT contents. Reproduced with permission. Copyright, Royal Society of Chemistry<sup>[106]</sup>. (C) J-V characteristic of PCDTBT:PC<sub>61</sub>BM with different P3HT contents. Reproduced with permission. Copyright, Elsevier<sup>[107]</sup>. (D) TEM images of PM6:Y6 binary and PM6:Y6:PDCBT-2F:N2200 ternary blends in CF and CB. Reproduced with permission. Copyright, 2022 American Chemical Society<sup>[108]</sup>. (E) AFM images of PBDB-T:ITIC as a function of PDCBT and P3HT content. Reproduced with permission. Copyright, 2019 American Chemical Society<sup>[109]</sup>.

had no obvious positive effect on the absorption spectra and the decreased  $V_{oc}$  and  $J_{sc}$  led to a decreased PCE [Figure 7C].

Based on the understanding that the crystalline state affecting the miscibility of the components might be related to the solvent type used in device fabrication, Yao *et al.* incorporated PDCBT-2F and N2200, acting as the additives, into the PM6:Y6 model system with chloroform (CF) and chlorobenzene (CB) as the processing solvents<sup>[108]</sup>. They found that the efficiency of the two systems processed with CF had an obvious increase. Different from PM6:Y6:N2200, the PCE of PDCBT-2F:PM6:Y6 processed with CB decreased rapidly due to the mismatched miscibility. As shown in Figure 7D, the excessive phase separation of PDCBT-2F:PM6:Y6 processed with CB led to a decreased PCE, while the incorporation of N2200 barely changed the morphology of the PM6:Y6 films, indicating that the crystallization and miscibility of Y6-based OSCs were related to the preaggregation resulting from the solvent type.

To further investigate the interaction between the third component and the blend from the perspective of miscibility, Yi and coworkers<sup>[109]</sup> selected two materials with similar chemical structures but different side chains, namely, P3HT and PDCBT, to act as the third component of the PBDB-T:ITIC blend. It was discovered that the miscibility between the third component and the acceptor, instead of the interaction



between P3HT/PDCBT and the polymer donor, dominated the phase separation. The Flory-Huggins interaction parameters of the three PDCBT:PBDB-T, PBDB-T:ITIC and P3HT:PDCBT blends were calculated according to the melting point depression method to evaluate the interaction of different blends. Compared with P3HT:ITIC, PDCBT:ITIC exhibited poor miscibility. Benefiting from the proper phase separation and enhanced crystallization, the device performance based on PDCBT:PBDB-T:ITIC OSCs improved from 9.40% to 10.97% [Figure 7E]. The PCE of P3HT:PBDB-T:ITIC decreased rapidly with increasing P3HT weight ratio. Combined with the previous studies from the same authors, a conclusion was reached that the mechanisms affecting the interaction between the third component and the control blend of OSCs based on fullerene and non-fullerene acceptors are not the same.

It seems that the incorporation of the third component as an additive generally leads to an increased  $J_{sc}$  resulting from the complementary absorption between the third component and host materials but a decreased FF originating from the poor compatibility. The film morphology of the ternary OSCs can be well controlled by manipulating the weight ratio of the additives and an appropriate post-treatment. The miscibility induced by the third component type or the preaggregation in different solvents provides guidelines for the selection of guest materials.

## CONCLUSIONS AND OUTLOOK

The design of PT and its derivatives boosts the efficiency of PT-based OSCs to over 17%. Ternary mixing is an effective strategy for further improving the efficiency and stability of PT-based OSCs. Applying approaches employed in fullerene OSCs may be an effective route to achieving highly efficient and thermally stable OSCs for PT:nonfullerene. Insulating polymers, conjugated small molecules and donor polymers are the most widely used guest materials. Notably, some studies have demonstrated that insulating polymers not only can effectively enhance the device performance, thermal stability<sup>[110]</sup> and mechanical properties of OSCs, but can also be conveniently applied in different systems<sup>[111]</sup>. Therefore, more emphasis should be devoted to the insulating polymers involved in ternary OSCs. Despite several principles based on absorption, energy levels, miscibility and the glass transition temperature having been proposed, specific and universal criteria should be perfected for selecting appropriate third components efficiently. Finally, we outline herein five possible directions that deserve further attention.

### (1) Thermally stable PT-based ternary solar cells

Excellent thermal stability is a prerequisite for the practical application of OSCs. Several studies have demonstrated that the morphology of blend systems containing appropriate third components, such as photovoltaic materials or insulating polymers, with good compatibility with host materials can be stabilized under continuous heating, contributing to the well-retained efficiency. As a small change in side chains can make a remarkable difference to its properties, PT can behave differently when the side chain is varied, for instance, odd or even numbered carbon chains. Blending two different PTs<sup>[112,113]</sup> is very promising for achieving thermally stable and high-efficiency solar cells. Furthermore, applying high- $T_g$  insulators<sup>[114]</sup> might be another good option.

### (2) Mechanically robust PT-based all-polymer ternary solar cells

Flexible and stretchable OSCs with high efficiency and excellent mechanical stability are considered as the most promising powering sources for wearable electronic devices. Li *et al.* recently demonstrated over 7% efficiency in all-polymer solar cells based on P3HT<sup>[115]</sup>. Thus, constructing ternary blends might be an efficient means to obtain over 10% efficiency in this kind of solar cell. Moreover, the cells might be

mechanically stretchable due to the entanglement of PT and polymer acceptors<sup>[2,116-118]</sup>. In addition, combining ternary blends with flexible substrates, such as flexible conductive textiles, is an effective strategy for achieving wearable electronic devices.

### (3) High-efficiency PT/quantum dot hybrid solar cells

A recent study<sup>[119]</sup> has shown the significant potential of polythiophenes in achieving high-performance hybrid organic and quantum dot solar cells<sup>[120,124]</sup>. Thus, designing ternary blends comprising PT as hole transporting layers might promote the power conversion efficiency of hybrid solar cells greatly.

### (4) Low-cost OSCs in real applications

Despite the scalability of PT and its derivatives, the high cost of acceptor molecules with highly fused and large aromatic groups seriously hinders the commercial application of OSCs. Therefore, the design and synthesis of acceptors with simple chemical structures and good reproducibility, such as non-fused ring acceptors based on oligothiophenes and derivatives, is urgently required.

### (5) Large-area OSCs based on PT

Studies have demonstrated that the insulators involved in ternary blends with varied film thicknesses can achieve comparable performance, which is favorable for the large-scale application of OSCs. Nonetheless, the relatively low efficiency is still the main challenge OSCs face. The ternary mixing of PT derivatives and insulator polymers may be a feasible method to fabricate high-efficiency large-area OSCs via scalable coating techniques such as blade-coating, slot-die coating, and roll-to-roll printing.

## DECLARATIONS

### Authors' contributions

Made substantial contributions to conception and design of the study: Ye L

Performed literature survey and wrote the first draft: Qi Q

Performed manuscript revisions, as well as provided administrative and technical support: Ye L, Ke H

### Availability of data and materials

Not applicable.

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### Conflicts of interest

All authors declared that there are no conflicts of interest.

### Ethical approval and consent to participate

Not applicable.

## Consent for publication

Not applicable.

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