

Review

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High-entropy alloy catalysts: high-throughput and machine learning-driven design

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Abstract

High-entropy alloy (HEA) catalysts have recently attracted worldwide research interest due to their promising catalytic performance. Most current studies focus on designing HEA catalysts through trial-and-error methods. This produces scattered data and is not conducive to obtaining a fundamental understanding of the structure-property-performance relationships for HEA catalysts, thereby hindering their rational design. High-throughput (HT) techniques and machine learning (ML) methods show significant potential in generating, processing and analyzing databases with a vast amount of data, providing a new strategy for the further development of HEA catalysts. In this review, we summarize the recent literature on HT techniques for HEA synthesis, characterization and performance testing. We also review the ML models that are used to process and analyze existing databases to accelerate the discovery of HEA catalysts. Finally, the potential challenges and perspectives of HT techniques and ML models are presented to accelerate the discovery of new HEA catalysts and promote their development.

Keywords: High-entropy alloys, catalysts, high-throughput, machine learning, structure-activity relationship



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INTRODUCTION

The widespread prevalence of catalysts in industry determines human living standards, as more than 85% of chemicals are made through catalytic reactions^[1-3]. In their inception phase, with the growing understanding of catalysis, a series of catalysts with high catalytic efficiency have been developed and designed, resulting in the rapid development of catalysis^[4-6]. The development of catalysts has been mainly focused on the two strategies of simplification and complication. The former involves reducing the dimensions (i.e., 3D → 2D → 1D → 0D) or decreasing the particle size of the catalysts. Regarding complication, single-atom catalysts (SACs) take this strategy to the extreme and have become the most popular catalysts due to their high catalytic performance^[7,8]. The large-scale production of SACs, however, still presents an enormous challenge. Researchers have attempted to synthesize complex morphologies or structures to improve catalytic performance. Alloying has been proven to be one of the most effective methods by various research groups through the design of different alloy systems, including binary, ternary and quaternary alloys, with excellent catalytic performance^[9-12].

As opposed to SACs, which have the simplest active site, high-entropy alloy (HEA) catalysts have more complex active sites due to their large compositional space and diverse atomic arrangements^[13]. HEAs are composed of five or more elemental components in near-equimolar ratios and were first reported by Cantor and Yeh in 2004^[14,15]. Due to their rich compositional and configurational spaces, some novel HEAs with specific mechanical properties have been designed and synthesized by direct current magnetron co-sputtering^[16,17]. As shown in [Figure 1](#), significant exponential growth has been witnessed in the field of HEAs as the number of publications grew from 137 publications in 2011 to 2684 publications in 2021. During the development of HEAs in the last two decades, researchers have systematically investigated and summarized the properties of HEAs, including high entropy, sluggish diffusion, lattice distortion and cocktail effects^[18,19]. These properties have shown their potential to improve the stability and activity of HEA catalysts^[20-23]. Moreover, considering the high abundance and low cost of Fe, Co, Ni and Cu, Li *et al.* synthesized HEA Pt₁₈Ni₂₆Fe₁₅Co₁₄Cu₂₇ nanoparticles for catalyzing the hydrogen evolution reaction (HER) and methanol oxidation reaction (MOR)^[24]. This shows that utilizing HEAs as catalysts could provide a strategy to reduce their cost by alloying non-noble metals. Overall, HEA catalysts exhibit significant potential due to their huge compositional space^[13]. Although the research into HEA catalysts is gradually increasing, as shown in the inset of [Figure 1](#), the application of HEAs in catalysis was only ~3% based on publications (82/2684) in 2021.

Although HEAs have demonstrated excellent catalytic performance for various catalytic reactions, such as the HER, oxygen reduction reaction (ORR), oxygen evolution reaction (OER), CO₂ reduction reaction (CO₂ RR), MOR and ammonia decomposition reaction, their structure-property-performance relationships are still ambiguous due to the complex active sites of HEA catalysts. A HEA with a face-centered cubic (FCC) structure containing five elements should have $5^{10} = 9,765,625$ active sites on its (111) surface when only the top site (one atom) and the nearest neighbor atoms (nine atoms) are considered active centers, while it should be 5^{15} and 5^{18} - 5^{19} for bridge (two active atoms + 13 neighbor atoms) and hollow (hollow-FCC: three active atoms + 15 neighbor atoms; hollow-hexagonal-closed packed (HCP): three active atoms + 16 neighbor atoms) sites, respectively. Herein, symmetry was not considered for calculating the number of active sites since there is no symmetry on HEA surfaces when considering second nearest neighbor atoms. Such vast possibilities make it impossible for them to be investigated experimentally or computationally. Until now, the development of HEA catalysts has been sluggish and random, limited by the traditional “trial-and-error” methods or “directed research” approaches^[25]. The available data are limited for understanding the nature of HEA catalysts. Consequently, the greatest challenge is identifying the active center in order to explore the structure-property-performance relationships of HEA catalysts. Only by fully

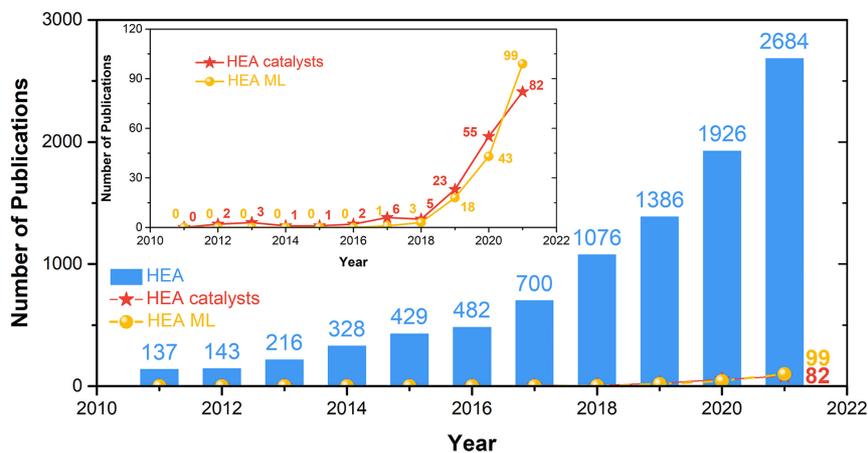


Figure 1. Number of publications on HEAs, HEA catalysts and HEA machine learning (ML) by 2021. The data are based on the Web of Science. HEA: high-entropy alloy.

understanding HEA catalysts can the rational design of HEA catalysts be implemented to make a breakthrough in catalysis.

To address the above challenge, more data are needed to analyze the structure-property-performance relationships of HEA catalysts. However, it is difficult to compile the single data points achieved by traditional “trial-and-error” methods and “directed research” approaches into an effective database due to the variance in synthesis methods, particle sizes, morphologies, and so on. Moreover, it is prohibitively time-consuming and costly to collect the data points one by one. To solve this issue, high-throughput (HT) techniques have been proposed to generate comprehensive databases with high efficiency^[26]. With the enhancement of computing power and the continuous improvement of theoretical calculation methods, HT theoretical calculations play an essential role in building databases as a result of their fast speed and low cost when compared with HT experiments. This does not mean that HT experiments are unnecessary, as they are essential to building a realistic database. To better understand HEA catalysts, a large database is a prerequisite and rational analysis of these data is indispensable. The diversity of HEA catalysts, however, leads to complex databases, which are difficult to analyze using only physical and chemical knowledge.

Fortunately, machine learning (ML) has received significant attention in recent years as a powerful tool for processing complex data^[27-30]. For example, to understand the structure-activity relationships of SACs for nitrogen fixation, different 2D materials supporting SACs and boron-doped graphene SACs were explored using HT density functional theory (DFT) calculations and ML^[28,29]. Deng *et al.* applied DFT calculations and ML to develop bi-atom catalysts for the ORR^[30]. These ML models provide new insights into atomic catalysts and help to speed up the design and discovery of new atomic catalysts. HEA catalysts are much more complex than atomic catalysts and the HT and ML methods show significant promise in exploring the structure-property-activity relationships of HEA catalysts, as evidenced by the works of Wan *et al.* and Roy *et al.*^[31,32]. These authors investigated IrPtRuRhAg and CuCoNiZnSn HEA catalysts by ML for the ORR and CO₂RR, respectively, providing rational guidance for the design of highly efficient HEA catalysts by altering the component elements and composition ratio. Unfortunately, the application of ML in the field of HEAs is not universal, as shown in Figure 1. We believe that ML could become the most powerful research tool for the study of HEAs.

In this review, we summarize the current HT techniques on the synthesis, characterization and performance testing used in the discovery of new HEA catalysts. Based on the achieved database, some ML models have been developed to analyze the structure-activity relationships and predict the catalytic activity of HEA catalysts for various catalytic reactions, such as the HER, OER/ORR, CO₂RR and ammonia decomposition or synthesis. After an overview of the application of HT techniques and ML models in HEA catalysts, we focus on the challenges, opportunities and prospects in the development of HEA catalysts.

HT RESEARCH FOR HEAS

HT techniques are essential for scientists to efficiently generate large databases and the subsequent information extraction^[33,34]. For HEAs with huge composition space, HT techniques can be used to great effect for the discovery and development of HEAs^[35]. HT techniques make HEA research more automatic, parallel and efficient^[36]. The application of HT techniques in the development of HEA catalysts is summarized from two aspects, namely, HT theoretical calculations and HT experimental approaches.

HT theoretical calculations

With the rapid development of high-performance computing and theoretical calculation methods, HT theoretical calculations have become more efficient in generating material databases compared to experiments^[37]. Both DFT and semi-empirical calculations of phase diagrams (CALPHAD) have shown to be viable and popular approaches for investigating the atomistic and thermodynamic mechanisms of existing HEA systems, as well as for new HEA catalyst design^[38].

The first step of a theoretical calculation is to build a HEA model, which is necessary for targeted and rapid HEA discovery and application^[39]. Currently, the most widely used method to build HEA structures is the special quasi-random structure (SQS) generation approach, which combines the cluster expansion technique and Monte Carlo algorithms through several codes (ICET, ATAT, MCSQS and Supercell)^[40-43]. The designed SQSs can model disordered alloys with atomic resolution and the radial distribution function of a random system is a quintessential concept for the generation of realistic random structures^[44]. Moreover, by combining artificial neural networks (ANNs) and evolutionary algorithms, our group proposed a neural evolution structure (NES) generation methodology for HEA structure generation [Figure 2]^[45]. According to pair distribution functions and atomic properties, a model is first trained on smaller unit cells and then the larger unit cell is generated by the inverse design approach. Compared to SQSs, the computational cost and time can be dramatically reduced (i.e., ~1000 times faster) for NESs and consequently large structures (over 400,00 atoms) can be generated in a few hours. Another advantage of NESs is that multiple structures with the same fractional composition can be generated by the same model.

Surface energies and work functions are directly related to adsorption performance and can therefore be used as descriptors of catalytic activity^[28,46,47]. Duong *et al.* calculated the surface energies and work functions of Co-Cr-Fe-Mn-Mo-Ni alloys using HT DFT calculations^[48]. In this work, low-order FCC alloys (up to quaternary) with different alloy compositions belonging to the senary Co-Cr-Fe-Mn-Mo-Ni system were considered in calculating their surface energies and work functions. The CALPHAD methodology was applied in the sub-regular solution model to analyze the populated first-principles data. CALPHAD models allow smooth interpolations across the alloy composition domains, as well as extrapolations to higher-order FCC HEAs, achieving more information with fewer data. In addition, considering the error of any models, Bayesian statistics were adopted to quantify, to some extent, the uncertainties of the CALPHAD models. Note that in this work, the achieved surface energies and work functions by HT calculations were used to rank the inherent corrosion-resistance potential of various equiatomic and near-equiatomic FCC HEAs belonging to the Co-Cr-Fe-Mn-Mo-Ni system, rather than their catalytic performance. We believe that HT

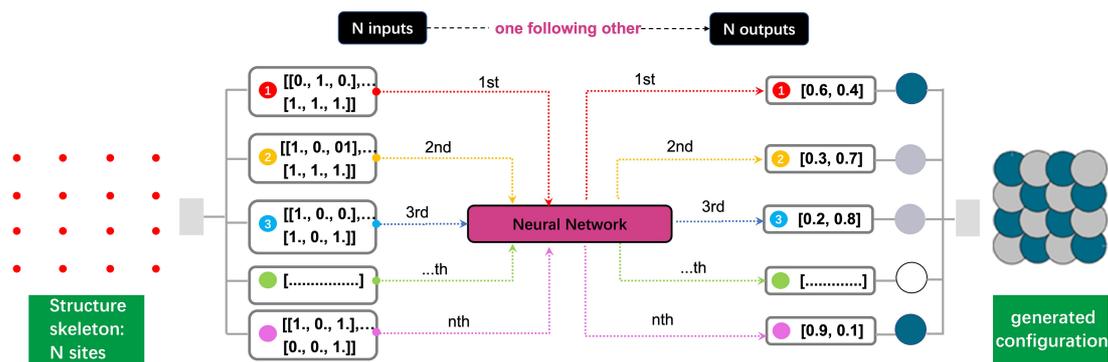


Figure 2. Roadmap of NES generation process. From left to right: an input structure (a template mesh with N sites), a representation of the N input arrays, the ANNs, a representation of the N output vectors, the atom type associated with each output vector and the generated configuration. Reproduced with permission^[45]. Copyright 2021, AIP Publishing. NES: neural evolution structure.

calculations of surface energies and work functions show significant potential for the application of HEAs in catalysis because these surface properties are closely related to surface catalysis.

The adsorption energy of a specific intermediate has been considered as a descriptor of the catalytic performance for the corresponding catalytic reaction. This can be utilized to deliver prominent results due to the Brønsted-Evans-Polanyi (BEP) relationship between activation barriers and reaction energies and the scaling relationship among the adsorption energies of some intermediates on catalyst surfaces^[49,50]. To reduce the calculation cost, it is necessary to only consider the adsorption energy of some important intermediates to evaluate the catalytic performance of the catalysts^[49]. Taking the CO₂RR as an example, various products can be produced at different potentials, while product distributions also depend on the solvent, promoter and other factors^[51,52]. Roy *et al.* considered that the intermediates of CO*, HCO*, H₂CO* and H₃CO* are more important for the formation of methanol from CO₂, even though several CO₂RR mechanistic pathways are possible for methanol formation^[53]. Moreover, the catalyst for the electrocatalytic CO₂RR should not be active for the HER and oxidized during the reaction process. Thus, except for CO*, HCO*, H₂CO* and H₃CO*, the adsorption energies of H* and O* were calculated by HT DFT calculations. The neighboring atoms of the active center on a certain catalyst play a vital role in determining the adsorption energy of a given intermediate adsorbed on the active center. In this work, the neighboring atoms were divided into three regions based on their varying effects on the adsorption energy. The first region is the atoms on the adsorption site; the second region includes the surface layer atoms around the adsorption site and the third region contains the subsurface layer atoms, which are the nearest neighbors of the adsorption sites, as shown in Figure 3A-C. A database with 474 data points in more than 40,000 possible microstructures was generated for the CO₂RR. Based on this database, the authors built an ML algorithm to predict the adsorption energies of the important intermediates on all possible adsorption sites. This work demonstrates an efficient approach to exploring catalysts with high catalytic performance, which can be extended to other reactions in the future.

For ammonia decomposition or synthesis, N*, NH*, NH₂* and NH₃* are important intermediates for adsorption energy calculations, as well as H*, since the HER is the most important side reaction for ammonia synthesis^[54,55]. Saidi *et al.* performed HT DFT calculations to determine the adsorption energies of H*, N*, NH*, NH₂* and NH₃* species on FeCoNiCuMo HEA surfaces^[56]. It was found that the most stable adsorption site changes from the HCP site to the bridge site and finally to the top site with increasing H atoms, as shown in Figure 3D. Based on these adsorption configurations, they established a database with

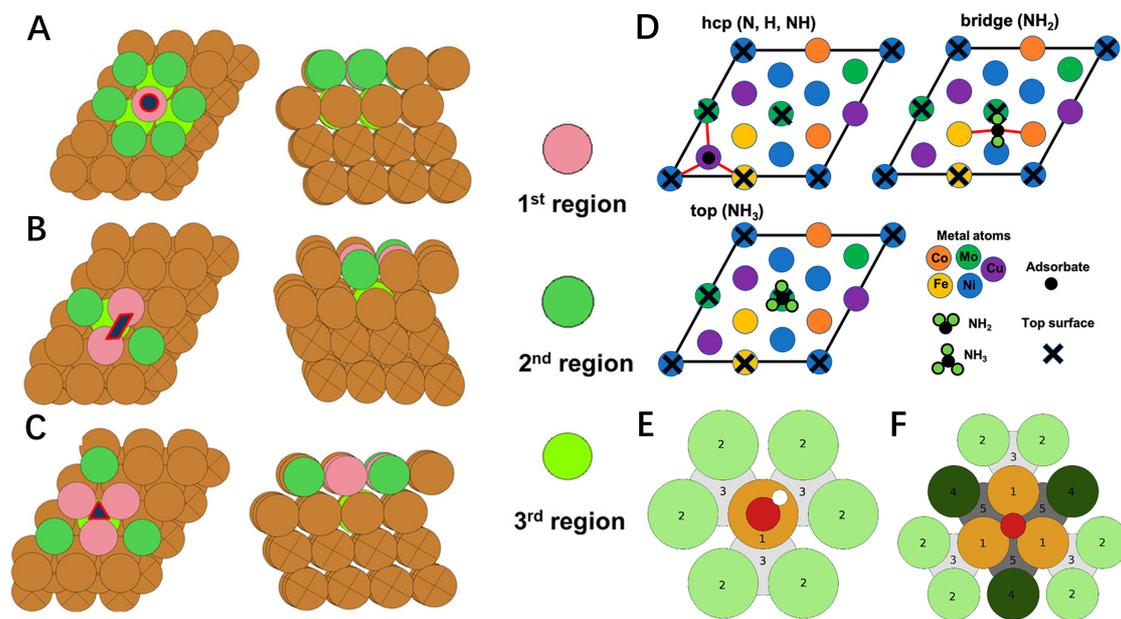


Figure 3. Top and side views of (A) on-top, (B) bridge and (C) hollow-HCP surface microstructures in a $3 \times 4 \times 4$ slab, where atoms with different colors denote different regions. (D) Top view for adsorption configurations for different chemical species on a representative HEA FCC (111) surface. Atoms belonging to the top surface layer are labeled with a cross. Bonds between adsorbates and metal atoms are shown as red lines. Parameterization of the surface configurations is achieved using nearest neighbors. (E) OH* on-top binding. Each colored zone represents a set of three parameters. Orange (1): binding site. Light green (2): surface neighbors singly coordinated to binding site. Light gray (3): subsurface neighbors singly coordinated to binding site. (F) O* FCC hollow site binding. Each colored zone represents a set of five parameters, except for zone 1, where the set is 35 parameters. Zones 1-3 as in (E) and additionally dark green (4): surface neighbors doubly coordinated to binding site. Dark gray (5): subsurface neighbors doubly coordinated to binding site. (A-C) Reproduced with permission^[53]. Copyright 2021, American Chemical Society. (D) Reproduced with permission^[56]. Copyright 2021, American Chemical Society. (E) and (F) Reproduced with permission^[60]. Copyright 2019, Elsevier. HCP: hexagonal-closed packed; FCC: face-centered cubic; HEA: high-entropy alloy.

1911 configurations. Combined with data analytics and ML, the scaling relationships among the binding energies of NH_x^* ($x = 0, 1, 2$ or 3) can still be seen in FeCoNiCuMo, as in the case of monometallic surfaces^[57,58]. These correlations indicate that the adsorption energy of N^* could be a good descriptor for ammonia decomposition and synthesis on these HEAs.

The adsorption of OH^* on Pt(111) is ~ 0.1 eV stronger than the optimum adsorption energy based on Sabatier volcano plots for catalyzing the ORR^[59]. Batchelor *et al.* presented HT DFT calculated adsorption energies of OH^* and O^* on the (111) surface of an IrPdPtRhRu HEA^[60]. As shown in Figure 3E and F, the most stable adsorption sites of OH^* and O^* are the top and FCC hollow adsorption sites, respectively. A model was established to link the local atomic arrangement around the adsorption sites to the adsorption energy, which can predict the adsorption energy values on all possible surface sites. Our group extended this model from the (111) surface to other Miller index surfaces, including the (100), (110), (211) and (532) of an equimolar FCC IrPdPtRhRu HEA^[61]. A total of 12 types of coordination environments were considered for HT DFT calculations of the adsorption energy of OH^* [Figure 4A], including more than 1,000 data points. According to these data points, we developed and trained a neural network model with high accuracy and universality. The ligand and coordination effects on the adsorption energy were analyzed sequentially by the feature importance based on our developed neural network model [Figure 4B]. Quantitatively, the adsorption energy was found to have a linear relationship with the total coordination number of nearest neighbors. More interestingly, the neural network model can be simplified to a simple linear scaling relationship with only a slight loss of accuracy. It has been demonstrated that these recently developed

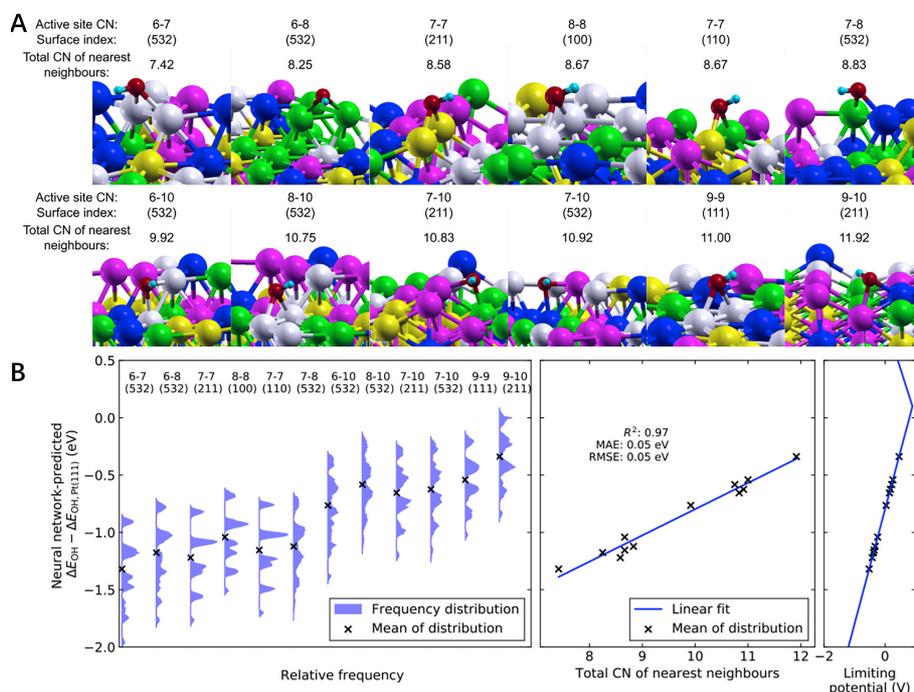


Figure 4. Adsorption energy affected by combined ligand and coordination effects. (A) Different coordination environments are ranked in increasing order of total coordination number of nearest neighbors, as defined in the text. (B) Frequency distribution of OH* adsorption energy for each coordination environment (left), whose mean values are found to correlate linearly with the total coordination number of nearest neighbors (middle). The coordination types are ordered the same manner in (A) and (B) and the energy values are horizontally aligned in (B). Reproduced with permission^[61]. Copyright 2020, Elsevier. MAE: mean absolute error; RMSE: root-mean-square error.

neural network techniques are powerful tools that can be used to overcome the huge chemical space of HEA catalysts.

HT experimental approach

HT theoretical calculations have shown significant potential in generating a database and predicting novel HEA catalysts with desirable catalytic performance. Most HT theoretical calculation techniques, however, utilize semi-empirical physical and chemical parameters without considering the complexity of such experiments^[62]. The synthesis, characterization and performance testing severely affect the reliability of theoretical calculation results. Therefore, the HT experimental approaches should also be explored to assess the structure, property and performance of HEA catalysts, which could further validate the HT theoretical calculation results^[63].

For example, Shukla *et al.* developed a solid-state gradient alloying method for the HT screening of HEA systems, in which a tapered section of a pure alloying element was retrofitted to the base alloy groove via milling and the subsequent friction stir processing of the assembled region achieved a continuous increase in alloy content of the additional element, as shown in Figure 5A-D^[64]. This prototype technique was applied to investigate the effect of the gradient variations of Cu on the phases (ϵ -hcp and γ -fcc) and the mechanical property response of a Fe₄₀Mn₂₀Co₂₀Cr₁₅Si₅ HEA. HEA samples with a compositional gradient can also be achieved by diffusion multiple technologies. Zhu *et al.* applied this technology to the HT synthesis of a Ti-based HEA^[65]. By combining the HT diffusion multiple technology and back propagation neural network, a Ti alloy was successfully designed and synthesized, which showed outstanding mechanical properties. HEA products with a continuous composition gradient achieved by HT experiments can be

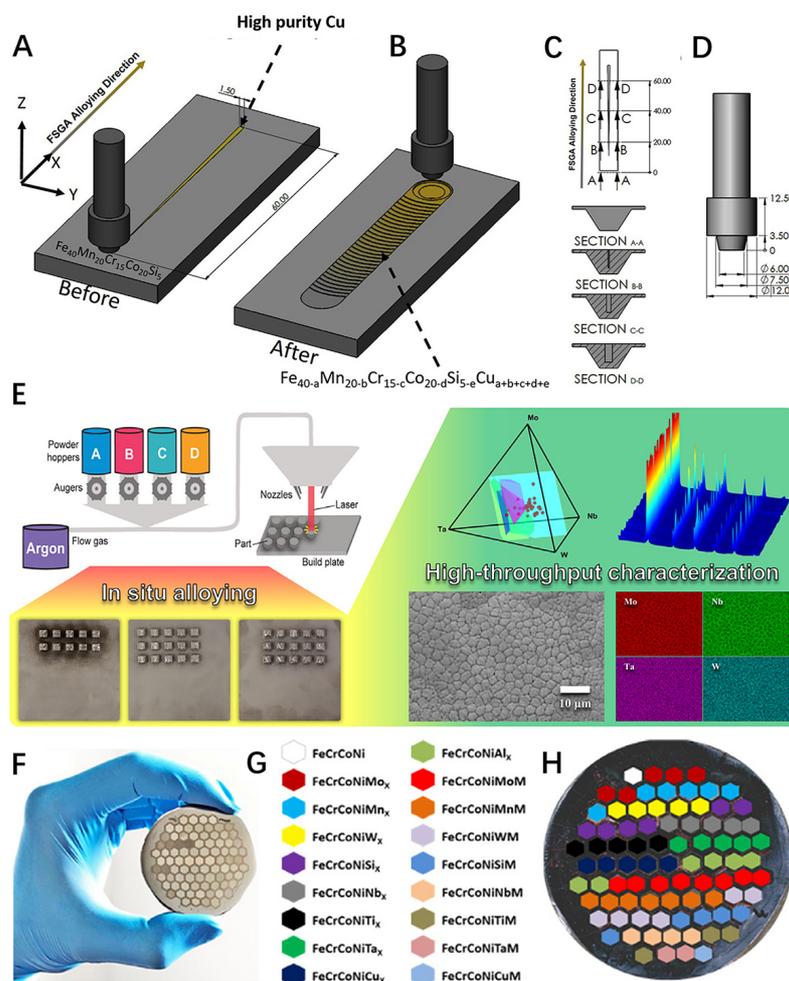


Figure 5. (A) Schematic of friction stir gradient alloying (FSGA) assembly showing tapered Cu section retrofitted in the groove created via CNC (Computer Numerical Control) milling on the base CS-HEA. The total length of the Cu section and groove is 60 mm, with at least 16 mm of base material on either side for tool plunging and retrieval (not drawn to scale). (B) Schematic of processed (alloyed) region after the FSGA process is completed. (C) Location of W-Re tool with respect to the tapered Cu plate during FSGA assembly and (D) design of the tool used in the current study. (E) Schematic of LENS MR-7 system, synthesized Mo-Nb-Ta-W arrays and corresponding HT characterization. (F, G) Design of honeycomb-structured HEAs. (A-D) Reproduced with permission^[64]. Copyright 2020, Elsevier. (E) Reproduced with permission^[67]. Copyright 2020, Elsevier. (F, G) Reproduced with permission^[72]. Copyright 2020, Elsevier. HEA: high-entropy alloy.

easily characterized by energy-dispersive X-ray spectroscopy (EDS) with scanning electron microscopy (SEM) and are thus suitable for building a database with fewer compositional variables to analyze the composition-activity relationships of HEA catalysts^[66].

Using an Optomec LENS MR-7, the arrays of different HEA compositions were produced by HT additive manufacturing in the form of directed energy deposition^[67]. Sample arrays of MoNbTaW HEAs were then characterized by SEM, EDS and X-ray diffraction (XRD). All characterizations were performed non-destructively with samples remaining on the build plate, thereby enabling future HT testing to be performed using the same build plate, as shown in Figure 5E. Pegues *et al.* employed HT additive manufacturing to synthesize a broad range of compositions in transition metal-based CoCrFeMnNi HEAs and assessed their microstructure and hardness as a function of alloy composition^[68]. As an additive manufacturing technique, laser metal deposition can be used to produce high-melting-point prototype

refractory HEAs^[69]. Li *et al.* rapidly fabricated continuously graded compositional libraries of $\text{Al}_x\text{CoCrFeNi}$ HEAs by a HT laser engineered net shaping process^[70,71]. XRD, SEM, scanning transmission electron microscopy (TEM) and nanoindentation were performed to analyze variations in the crystal structures, microstructures and mechanical properties of these HEAs. Currently, some limitations in the HT additive manufacturing method cannot be addressed. This methodology, however, has proven to be a high-efficiency tool in HEA development, as well as further validating HT computational approaches.

To accelerate the exploration of HEAs with targeted properties or performance, Zhao *et al.* developed the HT hot-isostatic-pressing-based micro-synthesis approach (HT-HIP-MSA), which can efficiently synthesize and characterize 85 combinatorial alloys in a 13-principal element alloying space, as shown in Figure 5F-H^[72]. Combined with theoretical computations, the HT-HIP-MSA can systematically and economically investigate the composition-structure-property relationships of HEAs. Moreover, the *in-situ* HT synthesis of FeCoNiCrCuAl_x was performed during TEM by Xu *et al.*, where they recorded the dynamic melting process of FeCoNiCrCu with Al and examined the composition of FeCoNiCrCuAl_x by EDS^[73]. This *in-situ* HT method provides a new strategy to produce HEA samples with high accuracy regarding composition.

Conventional methods for the synthesis of HEAs, such as arc melting, laser cladding, thermal spray, spark plasma sintering and ball milling, are too time-consuming for use in HT techniques for the accelerated discovery of HEAs. To address this issue, our group reported a radio frequency inductively coupled plasma (RF-ICP) method to synthesize HEAs in a rapid and HT fashion^[74]. The schematics of the experimental setup and HEA synthesis process using the RF-ICP system are presented in Figure 6A and B. As shown in Figure 6C and D, the time for HEA preparation was within 40 s and ~15 s were needed for cooling the sample. It was found that the porosity of the $\text{Cu}_{50}\text{Ni}_{50}$ binary alloy was significantly decreased by increasing the heating time from 9 to 21 s. More importantly, high-purity FeCoNi-based alloys ($\text{Fe}_{33}\text{Co}_{33}\text{Ni}_{34}$, $\text{Fe}_{25}\text{Co}_{25}\text{Ni}_{25}\text{Cu}_{25}$, $\text{Fe}_{20}\text{Co}_{20}\text{Ni}_{20}\text{Cu}_{20}\text{Al}_{20}$ and $\text{Fe}_{20}\text{Co}_{20}\text{Ni}_{20}\text{Cu}_{20}\text{Ti}_{20}$) were also successfully synthesized with a low level of defects. This methodology opens a new avenue to accelerate the compositional exploration of this multidimensional alloy space. We also believe data scientists and metallurgists would be inspired to explore new high-performance alloy systems with this methodology.

Huang *et al.* utilized eight or 28 sample holders of one electrode to simultaneously prepare eight or 28 different HEAs, respectively, in one batch of electrolysis under the same conditions^[75]. The prepared HEA systems at different locations of the one cathode will not contaminate each other due to the insolubilization of most transition metals and metal oxides in molten salts. Moreover, chemical solution deposition was applied to prepare a library of $[\text{Ca}_x(\text{Nb}_{1-y}\text{Ta}_y)_{1-x}]_{1-z}\text{Bi}_z\text{O}_8$ films with a total of 288 compositions^[76]. The surface and cross-section microstructures of these designed systems were characterized using field-emission SEM. A HT XRD system was applied to analyze the corresponding crystal structures using synchrotron radiation with a wavelength of 0.8 Å along with a 2D detector (PILATUS) at the SPring-8 facility. This work indicates that implementing HT conductivity measurements and HT XRD allowed us to increase the total experimental throughput for exploring HEA materials.

The HT techniques of computation, synthesis, processing, characterization and data analysis to accelerate the discovery of HEAs are well established. An integrated closed-loop process for HT HEA development, however, has been demonstrated infrequently. Vecchio *et al.* developed a HT rapid experimental alloy development (HT-READ) methodology, as shown in Figure 7^[77]. CALPHAD and ML model-based computational screening provided recommendations for composition selection and sample library design. The designed samples were synthesized, processed, characterized, tested and analyzed in an automated HT

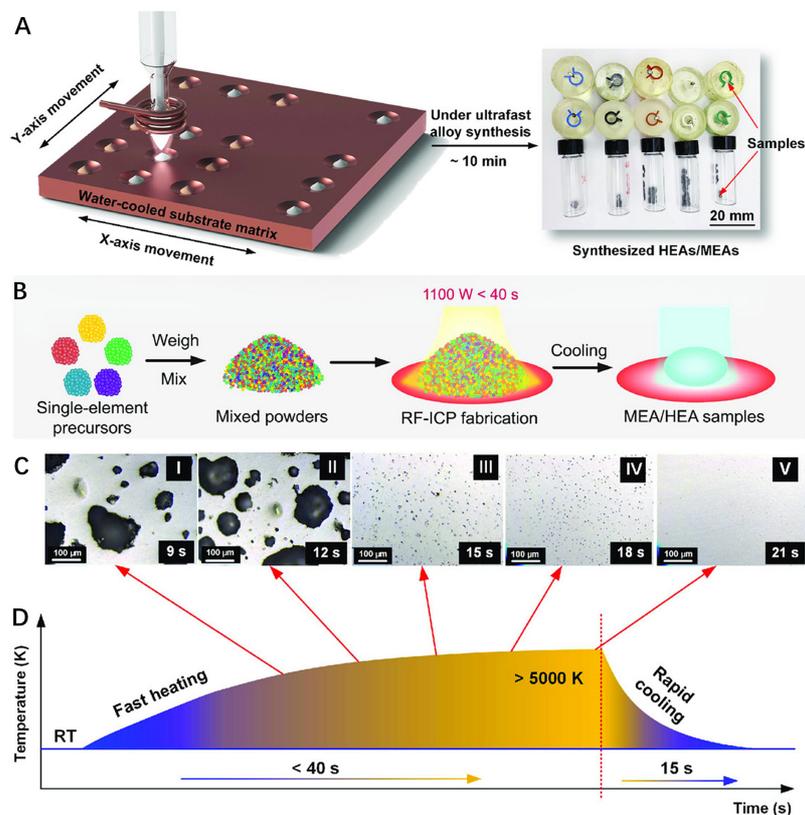


Figure 6. Schematic of (A) HT experimental setup for synthesizing alloys with a large composition space. (B) RF-ICP synthesis of HEAs using mixed pure metal powders. Fast RF-ICP synthesis of alloys using powder mixtures: (C) typical optical microscopy (OM) images showing the evolution of porosity in the $\text{Cu}_{50}\text{Ni}_{50}$ alloy under different heating times; (D) temperature profile of RF-ICP synthesis. Reproduced with permission^[74]. Copyright 2021, Wiley-VCH. RF-ICP: radio frequency inductively coupled plasma; MEA: medium-entropy alloy; HEA: high-entropy alloy.

fashion. The achieved knowledge was the key to improving the subsequent screening and alloy design. This work indicates that HT theoretical calculations, synthesis and characterization are no longer bottlenecks for the development of HEAs. Note that these methodologies were used to develop HEAs with optimum mechanical properties. It is still challenging to achieve desired catalytic performances in these HT synthesized HEA catalysts, as it is difficult to control the surface properties, morphology and particle size of the synthesized HEAs, which are crucial in the field of catalysis.

To extend the application of HEAs in the catalysis field, Yao *et al.* reported a HT synthesis technique for the compositional design and rapid thermal-shock treatment of ultrafine HEA nanoclusters (PtPdRhRuIrFeCoNi) with a homogeneous alloy structure, as shown in Figure 8A^[78]. In this process, carbon materials with surface defects were used as the supports to ensure the size uniformity for the different composition samples. The HT HEA catalyst systems were then rapidly tested by scanning droplet cell analysis [Figure 8B] for their electrochemical ORR. Their corresponding catalytic performance is displayed in Figure 8C and D, where the two best-performing HEA catalysts were quickly identified. This work indicates that the rapid synthesis and compositional exploration of HEAs by HT techniques are very efficient for exploring HEA catalysts.

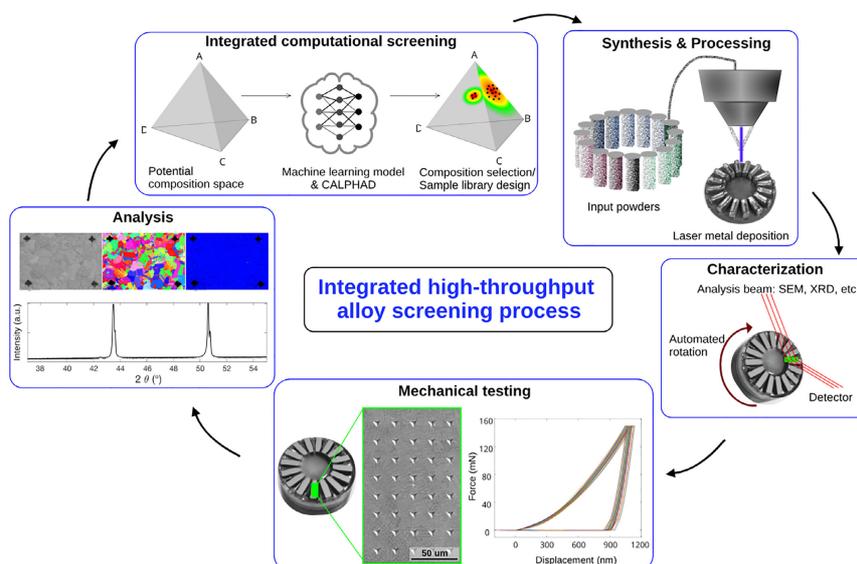


Figure 7. Illustration of the steps incorporated into the integrated HT-READ methodology. Clockwise from the top left, computational screening utilizing CALPHAD and the ML model provides recommendations for sample library compositions. The samples are then synthesized, processed, characterized, tested and analyzed in an automated HT fashion. New data are utilized to improve the subsequent screening and design. Reproduced with permission^[77]. Copyright 2021, Elsevier. CALPHAD: calculations of phase diagrams; SEM: scanning electron microscopy; XRD: X-ray diffraction; HT-READ: HT rapid experimental alloy development; HT: high-throughput.

The co-sputtering method has shown significant potential for the HT synthesis of HEAs due to the small grain size, few defects and low contamination in film samples^[74,79]. More importantly, this HT co-sputtering approach has been utilized successfully to synthesize HEA catalysts and build a library. The Ludwig group demonstrated a closed-loop, data-driven HT experimentation technique that iteratively combines DFT calculations, the combinatorial synthesis of material libraries and HT characterization [Figure 8E]^[80]. In this work, three Ag-Ir-Pd-Pt-Ru material libraries with large compositional spaces, centered around the predicted compositions, were prepared by combinatorial co-sputtering of the five elemental targets. The refined model, with the input derived from HT characterization data sets, could predict the activity of the exemplary Ag-Ir-Pd-Pt-Ru model system and further identify optimal HEA catalysts in an unprecedented manner. Furthermore, they extended this strategy to the HEA systems of Rh-Ir-Pd-Pt-Ru to unravel the composition-activity-stability relations in HEA electrocatalysts^[81].

ML MODELS FOR HEA CATALYSTS

ML is a powerful tool for accelerating catalyst discovery, as it can be used to build models with high accuracy, predict the catalytic performance of unknown catalysts and understand the structure-property-performance relationships, especially for HEA catalysts with huge compositional spaces^[82]. The key to successful ML models is to use suitable general descriptors, which can accurately and comprehensively represent the structural information of the catalysts. An effective descriptor can accelerate the development of ML models and uncover the fundamental physical nature of the catalytic process^[83]. In the review, we briefly summarize the ML models with the descriptors developed in the field of HEA catalysts, which mainly include the following four reactions of ammonia decomposition and synthesis, the ORR and the CO₂RR.

ML models of ammonia decomposition and synthesis on HEA catalysts

Based on HT calculations, a database with 1911 configurations was generated by Saidi *et al.*, where the adsorption energies of N* were within the range of -2.4-1.2 eV on CoMoFeNiCu HEAs^[56]. The large

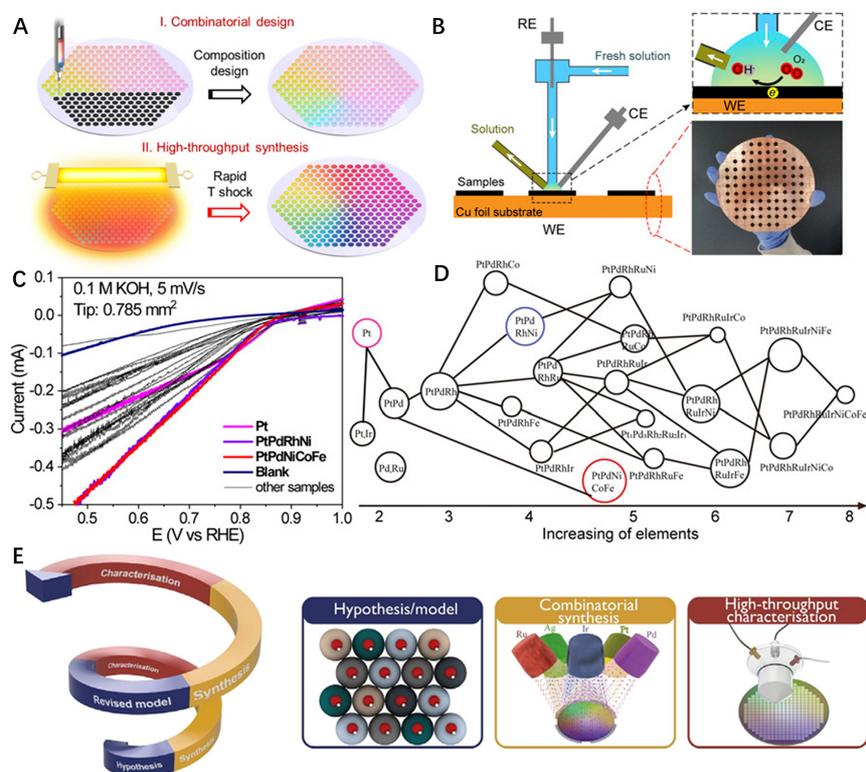


Figure 8. (A) Schematic illustration of combinatorial and HT synthesis of uniform MMNCs. (B) Scanning droplet cell setup and patterned samples on copper substrate. (C) Fast screening of PtPd-based MMNCs for catalytic ORR (22 compositions + one blank, 0.1 M KOH, 5 mV/s scan rate). (D) Compositional designs and their corresponding ORR performance are presented in a neural network diagram. The size of the circles represents the magnitude of the specific current at 0.45 V for ORR. (E) Data-driven discovery cycle combines prediction, combinatorial synthesis and HT characterization. (A-D) Reproduced with permission^[78]. Copyright 2019, Science. (E) Reproduced with permission^[80]. Copyright 2021, Wiley-VCH. CE: counter electrode; RE: reference electrode; WE: working electrode.

number of adsorption sites with different chemical environments results in the large variance in the adsorption energies of N*. This means that such a wide variation requires training the data set with a wide range of different systems, thereby increasing the training time for accurate ML. To consider the symmetry of active sites, the data set was extended by 10%-25%. Taking the HCP-hollow site as an example, little variation in the adsorption energy can be observed when changing the arrangement of three nearest-neighbor elements around the adsorption site. A ratio of 80%/20% for the training/testing data set was selected to build the convolutional neural network model. The high accuracy of the convolutional neural network model is demonstrated by a mean absolute error (MAE) of 0.05 eV per nitrogen atom. Furthermore, there is no systematic bias in these predictions, suggesting that the accuracy of the neural network model is absolutely comparable to the intrinsic accuracy of DFT calculations, which is used for the generation of the training data set. The catalytic activity of CoMoFeNiCu for ammonia decomposition and synthesis was optimized, where Co_xMo_yFe₁₀Ni₁₀Cu₁₀ with x:y ratios of 25:45-35:35 have a similar nitrogen binding energy to that of Ru (0001). Recent experimental results also indicated that Co₂₅Mo₄₅Fe₁₀Ni₁₀Cu₁₀ is a high-performance catalyst towards ammonia decomposition^[20].

ML models of ORR on HEA catalysts

The Rossmeisl group performed HT DFT calculations for the adsorption energies of OH* and O* on 871 and 998 different 2 × 2 unit cells of IrPdPtRhRu HEAs, respectively, as summarized in Figure 9A and B^[60].

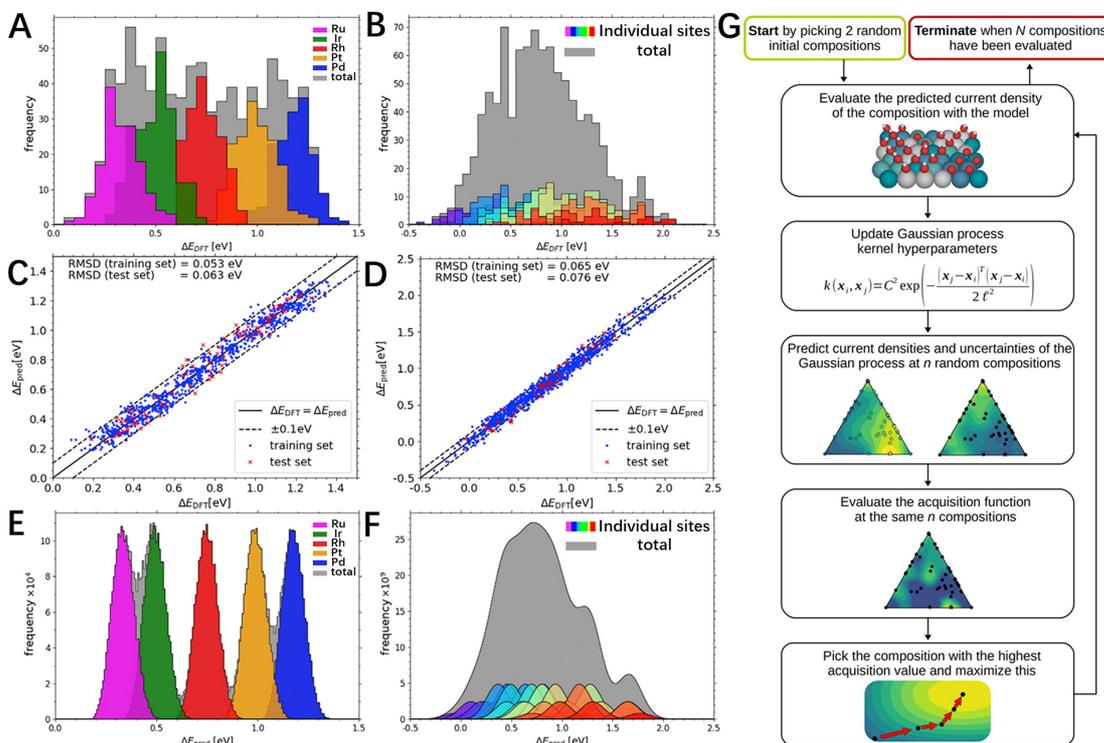


Figure 9. (A) OH* adsorption energies for 871 2×2 periodic unit cells. (B) O* adsorption energies for 998 2×2 periodic unit cells. (C) OH* adsorption. The linear model was trained on 871 symmetric 2×2 unit cells (blue dots) and tested on 76 asymmetric 3×4 unit cells (red crosses). The linear model used 15 parameters. (D) O* adsorption. The linear model was trained on 998 symmetric 2×2 unit cells (blue dots) and tested on 36 asymmetric 3×4 unit cells (red crosses). The linear model used 55 parameters. The dashed lines span the region ± 0.1 eV, where most of the data were seen to be contained. (E) OH* adsorption. Each color represents an individual on-top binding site as in (A). (F) O* adsorption. Each color represents an individual FCC hollow binding site, as shown in (B). (G) Workflow of Bayesian optimization algorithm. The algorithm was terminated after $n = 150$ samples to ensure sufficient evaluations for gauging the deviation in the number of samples needed for the discovery of the optimal compositions. For evaluation of the acquisition function, $n = 1000$ random compositions were sampled. (A-F) Reproduced with permission^[60]. Copyright 2019, Elsevier. (G) Reproduced with permission^[84]. Copyright 2021, Wiley-VCH. FCC: face-centered cubic.

According to these data, the authors trained a ML model using the ordinary least squares algorithm to predict the full span of available adsorption energies on the HEA (111) surface. As shown in Figure 9C and D, the high predictive accuracy is indicated by root-mean-square deviations of 0.063 and 0.076 eV for OH* and O*, respectively, when compared with their corresponding DFT calculated adsorption energies. The ML model was then used with the full span of available adsorption energies and the predicted distributions, as shown in Figure 9E and F. The fully spanned adsorption energies are highly consistent with the DFT-calculated distributions of adsorption energies. With the full distribution spanned out, the surface could be optimized to maximize the likelihood of finding specific binding sites with the desired adsorption energy. They also proposed a workflow of the Bayesian optimization algorithm, as shown in Figure 9G^[80,84]. The surrogate function was initiated by choosing two random compositions. The expected improvement acquisition function was the basis for the selection of the next composition for further study. The expected improvement comes from the current densities predicted by the surrogate function and the readily obtained uncertainties of the predictions, which is a standard choice as a natural starting point for the study. The catalytic activity of the selected composition was studied by the kinetic model and the Gaussian process posterior was updated with this new sample using Bayesian inference, as implemented in scikit-learn. In most cases, 150 iterations of optimization were sufficient to achieve locally

optimal compositions with high activity. Combining HT DFT calculations, ML, data-guided combinatorial synthesis and HT characterization, these works demonstrate an efficient methodology for HT closed-loop materials design in the rising field of HEA catalysts.

ML models of CO₂RR on HEA catalysts

The ever-increasing demand for global energy and the need to replace CO₂-emitting fossil fuels with renewable sources have driven interest in energy conversion and storage. In particular, the electrochemical reduction of CO₂ to chemical feedstocks is a hot topic due to its high correlation with both CO₂ removal and renewable energy generation. To accelerate catalyst discovery for the CO₂RR, Zhong *et al.* developed a ML-accelerated HT DFT framework and explored 12,229 surfaces and 228,969 adsorption sites on 244 copper-containing intermetallic crystals^[85]. This work illustrates the significance of computation and ML for exploring multi-metallic systems in experiments. By combining DFT with supervised ML, Pedersen *et al.* presented a strategy for the probabilistic and unbiased discovery of high-performance CO₂RR catalysts on disordered CoCuGaNiZn and AgAuCuPdPt HEAs^[86]. Gaussian process regressors were trained by hundreds of adsorption energy values of CO* (on-top site) and H* (hollow site) on (111) surfaces of CoCuGaNiZn and AgAuCuPdPt, achieved by DFT calculations, as illustrated in Figure 10A-F. The normally distributed errors of the Gaussian process regressors are similar to those of the cross-validations. As seen in Figure 10A-F, most predictions are within the dotted lines (± 0.1 eV deviation from the DFT values), which indicates that the Gaussian process regressors successfully capture the essential parts of the chemical environment of adsorption sites. The learning curves, which give the relation between the prediction error and the number of training samples, validate that the Gaussian process regressors have converged prediction error for the current number of adsorption energies achieved by DFT calculations. For a ML model, the input feature is of vital importance for the precision and universality of the model. More importantly, it is essential to understand the structure-activity relationships of HEA catalysts. To address this aspect, Roy *et al.* applied the permutation importance module as implemented in the scikit-learn library of Python to understand the contribution of every input feature towards the output, as depicted in Figure 10G^[53]. To determine the correlation between every input feature, a correlation matrix was generated, where the highly correlated features could be eliminated to decrease the dimensionality of the data set. Moreover, the correlation of each metal from every region with the corresponding adsorption energy is easily achieved and analyzed by the feature importance.

Descriptors in ML models of HEA catalysts

The key to constructing a ML model is designing effective descriptors, which is more important for HEA catalysts due to the complex active sites. The appropriate descriptors as input features for a ML model should be achieved directly from databases or by the simplest DFT calculations and include sufficient information on surface active sites. Some approaches, such as coordination atom fingerprints (CAFs), Coulomb matrices, the spectrum of London and Axilrod-Teller-Muto, elemental properties and SLATM (EP & SLATM), smooth overlap of atomic positions, Voronoi connectivity-based crystal graph, labeled site crystal graph (LSCG) and FCHL19, have recently been reported^[87-94]. Li *et al.* applied elemental groups and periods (GP) to replace atomic numbers in the FCHL19, LSCG, Atomic Number and Coordination Number (ANCN) and CAF representations to achieve an effective improvement for predicting adsorption energies on alloys^[95]. This strategy effectively enables ML models to learn from the periodic table. An improvement is achieved up to ~ 0.2 eV in adsorption energy MAE, compared to those obtained using ANCN, CAF, FCHL19 and LSCG. In particular, for the GP-LSCG representation, the MAE is 0.05 eV (near chemical accuracy) in predicting hydrogen adsorption and ~ 0.1 eV for other strong binding adsorbates (C*, N*, O* and S*). Although this work mainly focuses on bimetallic alloy systems, it has the potential to be extended to HEA catalysts, which has been verified by another research group, who proposed a transferable ML model by considering the intrinsic properties of substrates and adsorbates^[96]. Simply training the

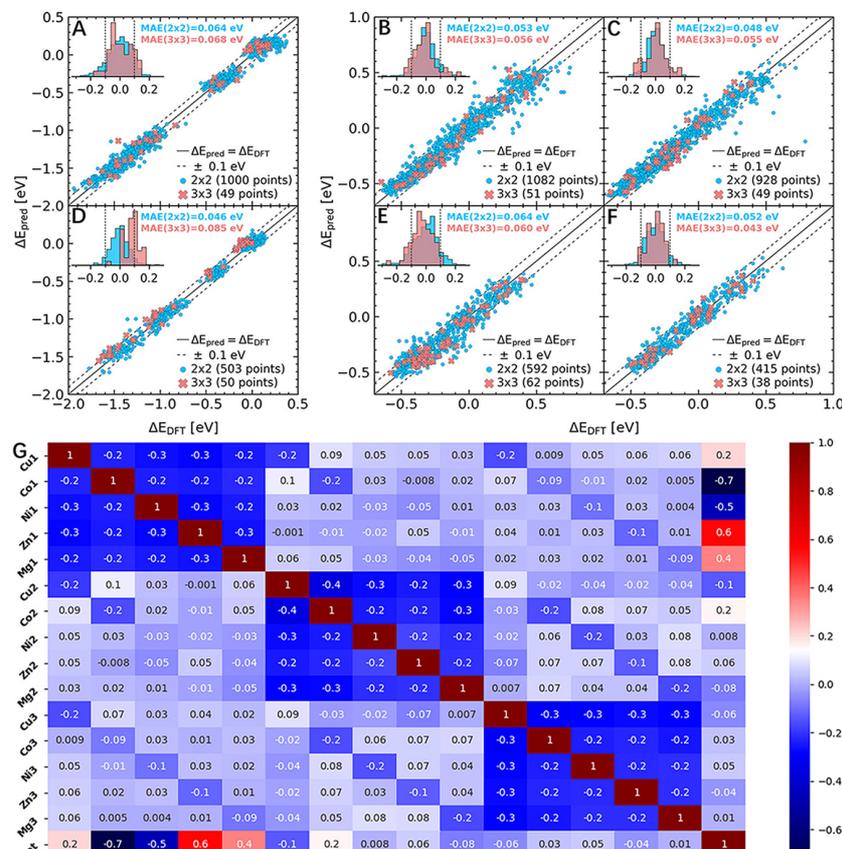


Figure 10. Plots showing GPR-predicted (ΔE_{pred}) versus DFT-calculated (ΔE_{DFT}) adsorption energies for (A-C) CoCuGaNiZn and (D-F) AgAuCuPdPt for (A, D) on-top CO, (B, E) FCC-hollow H and (C, F) HCP-hollow H. Blue and red indicate data for 2×2 and 3×3 atom slabs. MAEs are calculated as a fivefold cross-validation prediction error for the 2×2 and 3×3 slabs as the prediction error when training on the set of all 2×2 slabs. The insets show the distribution of the prediction errors in eV defined as $\Delta E_{\text{pred}} - \Delta E_{\text{DFT}}$. (G) correlation matrix of all the input features and output (target) adsorption energy for CO*, where 1, 2 and 3 represent the first, second and third regions of the microstructure, respectively. (A-F) Reproduced with permission^[86]. Copyright 2020, American Chemical Society. (G) Reproduced with permission^[53]. Copyright 2021, American Chemical Society. HCP: hexagonal-closed packed; FCC: face-centered cubic; MAE: mean absolute error.

properties of transition metals could predict the adsorption energies of single atom alloys, AB intermetallics and HEAs through understanding the relation between some descriptors (surface atom valence, electronegativity, coordination and adsorbate valence) and the adsorption energy. This transferable scheme could achieve new insights into the adsorption mechanism on HEA surfaces and the rapid design of HEA catalysts.

Although the current applications of ML in the field of HEA catalysts are limited to the reactions discussed above, some HEAs have shown excellent catalytic performance for some other reactions^[18,19,97]. For instance, Wang *et al.* developed a new class of structurally ordered PtRhFeNiCu HEAs as electrocatalysts for the ethanol oxidation reaction^[98]. Feng *et al.* synthesized ultrasmall HEA nanoparticles with an average diameter of 1.68 nm by a suitable and scalable synthetic strategy, which achieved an ultrahigh mass activity of $28.3 \text{ A mg}^{-1}_{\text{ noble metals}}$ at -0.05 V (vs. RHE) for the HER in $0.5 \text{ M H}_2\text{SO}_4$ solution^[99]. Thus, ML has excellent potential in the field of HEA catalysts, which will promote their rapid development.

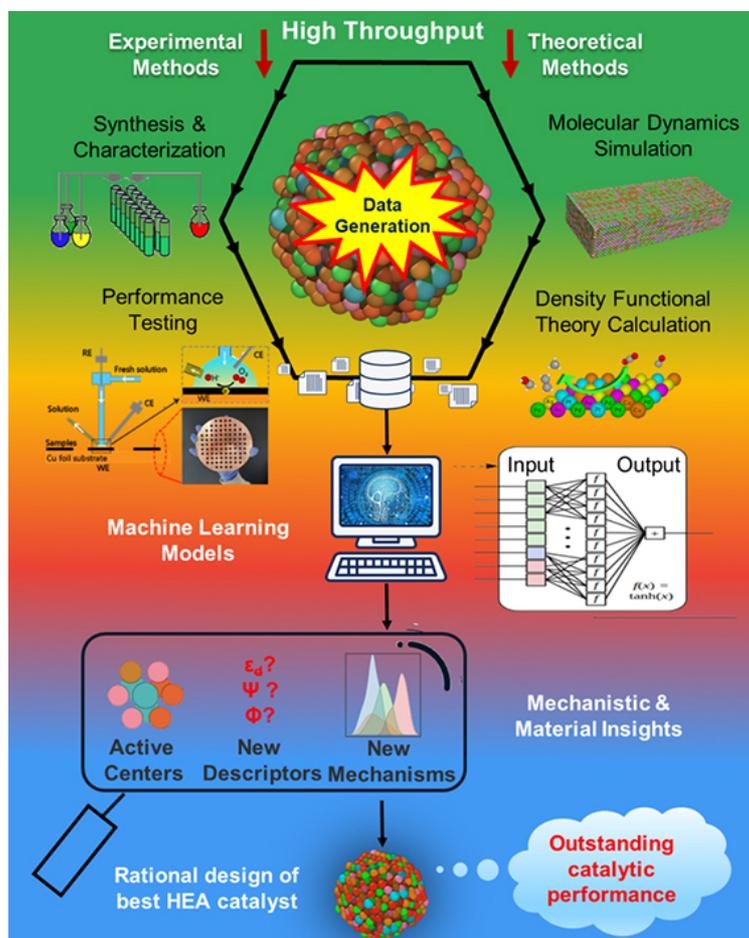


Figure 11. New research strategy for HEA catalysts. HT experimental and theoretical methods are used to generate comprehensive databases of HEA catalysts. Highly transferable and accurate ML models are explored to analyze databases and predict optimal HEA catalysts. New insights into active centers and new catalytic mechanisms and descriptors are expected to be developed on the basis of HT techniques and ML models. Finally, high-performance HEA catalysts will be rationally designed, promoting the development of catalysis. Reproduced with permission^[25]. Copyright 2022, Elsevier. CE: counter electrode; RE: reference electrode; WE: working electrode; HEA: high-entropy alloy; HT: high-throughput; ML: machine learning.

CHALLENGES

HEA catalyst research is in its infancy and some open questions for synthetic methods, catalytic reactions and mechanistic understandings should be addressed. As the number of components increases, the active centers of HEAs become much more complex compared to traditional alloy systems. Thus, the key study in the research of HEA catalysts is the identification of active centers. The development of HT techniques and ML models is an essential part of the accelerated research of HEA catalysts with a huge compositional space, as illustrated in Figure 11^[25]. However, more challenges are still unresolved for the development of HT techniques and ML models for HEA catalysts, as shown below:

- (1) The morphology and particle size of HEA catalysts should have obvious influences on the catalytic performance; however, they are difficult to control flexibly by current HT synthesis techniques;
- (2) Though many HT synthesis techniques have been successful, the synthesized HEAs are not easy to further test for catalytic performance;

(3) Current HT calculations are mainly focused on the adsorption of some important intermediates on HEA catalysts, rather than the most intuitive catalytic performance. This strategy effectively reduces the cost of computation; however, it only works when the BEP and scaling relationships between the adsorption energies of the relevant intermediates are still valid on the surface of the HEA catalysts;

(4) Although ML has a powerful prediction ability, the accuracy depends on the sufficiency of training data. The existing databases contain many valuable material data; unfortunately, there are still more data in the published literature that cannot be entered into databases and shared. Therefore, a more comprehensive and generic material information standard should be established to achieve data sharing among databases and to reduce obstacles in data acquisition;

(5) The key challenge in ML is the exploration of suitable, simple and general descriptors to accurately describe HEA catalysts, which are required to reasonably design catalysts and efficiently screen candidates;

(6) The prediction ability of ML models based on HT techniques has been proven to be powerful. Another important milestone is to uncover the structure-property-performance relationships of HEA catalysts and explore new mechanisms in the catalysis field, which deserves more attention in future research.

PERSPECTIVES

To address the above challenges, more HT synthesis strategies, as well as HT techniques for characterizing and testing catalytic performances, should be explored to build databases for HEA catalysts. The achieved databases should be more comprehensive, realistic, reliable and universal. In addition to the composition of HEA catalysts, we should also pay attention to other parameters (such as morphology, size, specific surface area and reaction environments), which are directly related to catalytic performances. For HT calculations, a more comprehensive list of catalytic performance parameters should be considered to evaluate the catalytic performance of HEA catalysts, rather than only calculating the adsorption energy of some important intermediates. Because both the BEP relationship and scaling relationship of adsorption energies are currently controversial phenomena for HEA catalysts, which need to be further verified. With their complex surface active sites, HEA catalysts have shown excellent catalytic activity for various catalytic reactions. The selectivity of HEA catalysts, however, is also a concern as these complex active centers might catalyze multiple reactions, which is another challenge for the development of HEA catalysts. To make full use of the existing data (both experimental and calculation data) in published studies and databases, and avoid the batch effect, workflows with natural language processing techniques should be explored to achieve effective communication between humans and computers with human languages and integrate these useful data into a comprehensive database. Finally, more effective descriptors should be proposed, as the accuracy and universality of the designed ML models are determined by the descriptors adopted. An effective descriptor can accelerate our understanding of HEA catalysts and help us to discover new catalytic mechanisms, thus promoting the development of HEA catalysts. We hope that this review will help researchers to better understand the significance of HT techniques and ML models for the development of HEA catalysts.

DECLARATIONS

Authors' contributions

Made substantial contributions to conception and design of this review, writing and editing: Chen L, Singh CV, Zou Y

Made substantial contributions to collation of literatures, figures preparation, and writing: Chen L, Chen Z, Singh CV, Zou Y

Performed data analysis, discussion and writing review: Chen L, Chen Z, Yao X, Su B, Chen W, Pang X, Kim KS, Singh CV, Zou Y

Performed data acquisition and interpretation: Chen L, Zou Y

Availability of data and materials

Not applicable.

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Conflicts of interest

All authors declared that there are no conflicts of interest.

Ethical approval and consent to participate

Not applicable.

Consent for publication

Not applicable.

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