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Promoting the reversibility of electrolytic MnO₂-Zn battery with high areal capacity by VOSO₄ mediator

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Abstract

Electrolytic MnO_2 -Zn batteries possess high energy density due to the high reduction potential and capacity of the cathode Mn^{2+}/MnO_2 . However, the low reversibility of the Mn^{2+}/MnO_2 conversion results in a limited lifespan. In this study, we propose the utilization of VOSO₄ as a redox mediator in the MnO_2 -Zn battery to facilitate the dissolution of MnO_2 . Through various techniques such as electrochemical measurements, *ex-situ* UV-visible spectroscopy, X-ray diffraction, and scanning electron microscopes, we validate the interaction between VO^{2+} and MnO_2 , which effectively mitigates the accumulation of MnO_2 . The introduction of the redox mediator results in exceptional redox reversibility and outstanding cycling stability of the $MnO_2/VOSO_4$ -Zn battery at high areal capacities, with 900 cycles at 5 mAh cm⁻² and 500 cycles at 10 mAh cm⁻². Notably, even in the flow battery device, the battery exhibits a stable cycling performance over 300 cycles at 20 mAh cm⁻². These research findings shed light on the potential large-scale application of electrolytic MnO_2 -Zn batteries.

Keywords: Electrolytic MnO₂-Zn batteries, high areal capacity, redox mediators, VOSO₄, cycling stability

INTRODUCTION

Lithium-ion batteries have emerged as the leading power source due to their exceptional energy density^[1,2]. However, the flammability and toxicity of organic electrolytes have raised significant safety concerns,

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restricting their widespread use in energy storage applications^[3,4]. In light of these challenges, aqueous zinc (Zn)-ion batteries have gained considerable attention. Zn, with its low reduction potential (-0.763 V *vs*. SHE) [Equation (1)], abundant reserves, and affordable price, stands out as an attractive option. Furthermore, the use of aqueous electrolytes ensures inherent safety and environmental friendliness^[5-9].

In the realm of Zn-ion batteries, MnO_2 has traditionally served as a cathode material. It exhibits a theoretical capacity of 308 mAh g⁻¹ by allowing the intercalation of Zn²⁺ and/or H⁺ ions while reducing Mn⁴⁺ to $Mn^{3+[10,11]}$. Recently, a novel electrolytic mechanism of MnO_2 [Equation (2)] has been reported. The two-electron transfer reaction involves the deposition and dissolution of MnO_2 , resulting in an impressive theoretical capacity of 616 mAh g^{-1[12-18]}. Notably, the MnO_2 -Zn battery benefits from the high reduction potential of Mn^{2+}/MnO_2 (1.228 V), enabling it to achieve a high voltage output of 1.991 V^[19-28].

$$\operatorname{Zn}^{2+} + 2e^{-} \to \operatorname{Zn}$$
 $E = -0.763 \text{ V} \text{ vs. SHE}$ (1)

$$Mn^{2+} + 2H_2O \rightarrow MnO_2 + 4H^+ + 2e^ E = 1.228 V vs. SHE$$
 (2)

However, electrolytic MnO_2 -Zn batteries exhibit a limited cycling lifespan due to the incomplete dissolution of deposited MnO_2 . This issue is exacerbated by the accumulation of MnO_2 particles, which can detach from the current collector and block ion transportation channels in the separator. Consequently, substantial polarization and rapid capacity degradation occur in the battery^[29,30].

To address this prominent issue, various ions have been introduced into the electrolyte of electrolytic MnO_2 -Zn batteries to enhance their electrochemical performance^[31-36]. For instance, the incorporation of Ni²⁺ ions serves as a catalyst for the Mn^{2+}/MnO_2 reaction kinetics, enabling charge and discharge at a rate of 50 C while maintaining a lifespan of over 450 cycles at an areal capacity of 1 mAh cm^{-2[31]}. Additionally, Al³⁺ ions play a vital role in promoting MnO_2 dissolution by creating oxygen vacancies during deposition, leading to improved cycling stability over 2,000 cycles at an areal capacity of 2 mAh cm^{-2[36]}. Another approach involves the utilization of redox mediators such as I⁻ and Br⁻ ions^[30,33], which interact with MnO_2 to prevent its accumulation. This mechanism has shown promising results, extending the lifespan of electrolytic MnO_2 -Zn batteries to over 100 cycles at an areal capacity of 5 mAh cm⁻². Nevertheless, achieving enhanced reversibility and cyclic stability, particularly at high areal capacities, remains a significant challenge.

In this study, we employ VOSO₄ as a redox mediator to enhance the dissolution of MnO_2 in the electrolytic MnO_2 -Zn battery. VOSO₄ is commonly utilized as an active component in the cathode of vanadium-based flow batteries due to its favorable reversibility and stability^[37,38], Given that the reduction potential of the VO^{2+}/VO_2^+ reaction is slightly lower than that of the Mn^{2+}/MnO_2 reaction (1.01 V *vs.* 1.228 V), VOSO₄ is a suitable choice as a redox mediator for the Mn^{2+}/MnO_2 reaction. Through electrochemical measurements, *ex-situ* UV-visible (UV-vis) spectroscopy, X-ray diffraction (XRD), and scanning electron microscopes (SEM), we validate the interaction between VO^{2+} and MnO_2 , which effectively mitigates the accumulation of MnO_2 . By employing this redox mediator, the $MnO_2/VOSO_4$ -Zn battery exhibits exceptional redox reversibility and excellent cycling stability, even at high areal capacities. Specifically, it achieves 900 cycles at 5 mAh cm⁻² and 500 cycles at 10 mAh cm⁻². Notably, even in the flow battery device, the battery demonstrates stable cycling performance for over 300 cycles at 20 mAh cm⁻².

EXPERIMENTAL

Materials

Manganese sulfate monohydrate ($MnSO_4 \cdot H_2O$, 99%), zinc sulfate heptahydrate ($ZnSO_4 \cdot 7H_2O$, 99%), and sodium sulfate (Na_2SO_4 , 99%) were all received from Aladdin, Shanghai. Vanadium (IV) sulfate oxide hydrate ($VOSO_4 \cdot xH_2O$, 99.9%) was received from Macklin. Sulfuric acid (H_2SO_4 , 99%) was sourced from the Guangzhou Chemical Reagent Factory. Carbon felt (5 mm in thickness) was purchased from CeTech Co. Ltd. Zinc foil (0.2 mm in thickness) was purchased from Tianjin Avixin Chemical Technology Co., Ltd. Cation exchange membrane Nafion 117 was acquired from Dupont.

Cell assembly

In both static and flow batteries, carbon felts were applied as current collectors for the cathodes, and zinc foils were applied as the anodes. To prevent the shuttle effect of VO_2^{+}/VO_2^{+} , the anode and cathode were separated by a cation exchange membrane (Nafion 117) in a battery. Anolytes consisted of 1 M ZnSO₄ and 0.5 M H₂SO₄. Catholytes of E-Mn (0.5 M MnSO₄ + 1 M Na₂SO₄ + 0.5 M H₂SO₄) and E-MnVO (0.5 M MnSO₄ + 0.05 M VOSO₄ + 1 M Na₂SO₄ + 0.5 M H₂SO₄) and E-MnVO (0.5 M MnSO₄ + 0.05 M VOSO₄ + 1 M Na₂SO₄ + 0.5 M H₂SO₄) were used for the MnO₂-Zn and MnO₂/VOSO₄-Zn batteries, respectively. In a typical static battery, the anode and cathode were 1 cm × 1 cm in size. The anolyte and catholyte were both 5 mL. In the case of flow batteries, 25 mL of catholytes and anolytes were stored separately in two tanks. Electrolyte flow was driven by a peristaltic pump at a flow rate of approximately 40 mL min⁻¹.

Material characterization

UV-vis spectra were acquired using a UV-VIS Spectrophotometer (Shimadzu, UV-2600). SEM was carried out using a TESCAN GAIA3 instrument. XRD analysis was performed utilizing a PANalytical Empyrean instrument with Cu K α radiation. X-ray photoelectron spectroscopy (XPS) measurements were conducted using a Thermos Scientific K-Alpha instrument.

Electrochemical measurements

In a typical three-electrode setup, the cathode employed carbon felt as a current collector, while a platinum plate served as the counter electrode, and a saturated calomel electrode (SCE) functioned as the reference electrode. Cyclic voltammetry (CV) and linear sweep voltammetry (LSV) were conducted using an electrochemical workstation (Gamry Interface1000). The CV was performed at a scanning rate of 0.5 mV s⁻¹, while the LSV was conducted at a scanning rate of 1 mV s⁻¹. Electrochemical impedance spectroscopy was performed on a Chenhua electrochemical workstation (CHI660e) across a frequency range of 10^5 - 10^{-2} Hz. Galvanostatic charge/discharge tests were carried out using a Neware battery analyzer. In these tests, the battery was charged up to a specific capacity and discharged until reaching a cutoff voltage of 1.0 V.

RESULTS AND DISCUSSION

The electrochemical performance of the Mn^{2+}/MnO_2 was first studied using a three-electrode system. The electrolyte consisted of 0.5 M MnSO₄, 1 M Na₂SO₄, and 0.5 M H₂SO₄ (referred to as E-Mn). During the galvanostatic discharge/charge measurements [Figure 1A], E-Mn exhibits a cyclic life of less than 200 cycles when charged to 5 mAh and discharged to 0.2 V *vs.* SCE at an applied current density of 10 mA cm⁻². The limited cyclic life can be attributed to incomplete MnO₂ dissolution [Figure 2A]. As revealed in the SEM image [Supplementary Figure 1], a thick layer of MnO₂ remains on the carbon fiber after 30 cycles. Therefore, enhancing MnO₂ dissolution is imperative to extend the cyclic life of Mn²⁺/MnO₂.

The electrolyte, denoted as E-VO, consists of 0.05 M VOSO₄, 1 M Na₂SO₄, and 0.5 M H₂SO₄. E-VO displays a pair of peaks at 0.75/0.71 V vs. SCE on the CV curve, corresponding to the redox reaction of VO²⁺/VO₂⁺ [Figure 1B]. When charged/discharged in the range of 0.2-1.1 V at a current density of 10 mA cm⁻², E-VO



Figure 1. (A) Charge/discharge profiles, (B) CV curves of E-Mn, E-VO, and E-MnVO at 0.5 mV s⁻¹ and (C) cyclic performance of E-Mn and E-MnVO at 5 mAh cm⁻² and 10 mA cm⁻². (D) Optical images, (E) UV-vis spectra, and (F) discharge profiles of VOSO₄ solution, MnO₂ suspension, and a mixture of VOSO₄ and MnO₂.



Figure 2. Schematic illustration of the charge/discharge process of the (A) MnO₂-Zn and (B) MnO₂/VOSO₄-Zn batteries.

exhibits a stable discharge capacity of ~0.42 mAh cm⁻² over 4,000 cycles [Supplementary Figure 2]. Comparatively, the VO_2^+/VO_2^+ redox couple demonstrates a lower redox potential and improved cyclic

stability than Mn^{2+}/MnO_2 , indicating that $VOSO_4$ is a suitable redox mediator for the conversion reaction of Mn^{2+}/MnO_2 .

The cyclic performance of E-Mn at different VOSO₄ concentrations was evaluated [Supplementary Figure 3]. The E-Mn system with a VOSO₄ concentration of 0.02 M displays a relatively low coulombic efficiency and a short lifespan. While an addition of 0.1 M VOSO, improves the coulombic efficiency and cyclic stability, it results in a relatively low energy density. Consequently, a concentration of 0.05 M VOSO, was determined to be the optimal choice for the system, termed E-MnVO. The initial charge curve of E-MnVO exhibits a slope followed by a plateau at 1.13 V [Figure 1A], corresponding to the oxidation of VO^{2+} prior to that of Mn^{2+} . The oxidation of VO^{2+} contributes a capacity of ~0.5 mAh cm⁻², consistent with the charge capacity delivered by E-VO [Supplementary Figure 2]. Initially, E-MnVO displays low discharge capacities. After a few cycles of activation, the discharge capacity of E-MnVO gradually increases and stabilizes. This phenomenon may be attributed to the initial cycles where an excessive amount of $VO_{2^+}^{2^+}$ ions cannot be fully oxidized into $VO_{2^+}^{-1}$ ions during the charging process. These excess VO²⁺ ions react with MnO₂, consequently depleting MnO₂, but without contributing to the discharge capacity. As a result, the discharge capacity is relatively lower during the initial cycles. However, after some cycles, the oxidation of VO²⁺ ions contribute more capacity in the charge process as indicated by the lowered charge plateau, which may be because residual MnO, trap Mn²⁺ ions in the electrolyte. The remaining VO²⁺ ions are reduced in the electrolyte after the charge process, therefore leading to an increase of the discharge capacity. The discharge curve of the 100th cycle [Figure 1A] presents two plateaus, indicating the successive electrochemical reductions of MnO₂ (1.06 V) and VO₂⁺ (0.69 V). This cycle delivers a capacity of 4.75 mAh cm⁻² with a coulombic efficiency of 95%. Moreover, the charge/discharge platforms of E-MnVO maintain stability throughout cycling, with a cyclic life exceeding 1,000 cycles [Figure 1A]. In contrast, the charge/discharge platforms of the E-Mn gradually increase over cycling, suggesting a decrease in pH value due to the accumulation of MnO₂. The capacity of E-Mn remains stable for only 200 cycles [Figure 1C]. SEM images [Supplementary Figure 4] demonstrate the accumulation of thick layers of MnO₂ on or exfoliating from the carbon fibers in E-Mn after 200 cycles. However, with the addition of $VOSO_4$, the accumulation of MnO₂ on the carbon felt significantly decreases after 200 cycles.

To elucidate the underlying mechanism of the VOSO₄ mediator, we conducted an experiment by adding MnO_2 powder into the blue VOSO₄ solution. As a result, the precipitate dissolved [Figure 1D], leading to a change in solution color to light yellow. Accordingly, the distinct peak on the UV-vis spectrum of VOSO₄, corresponding to VO²⁺ (~770 nm)^[37], disappeared upon the addition of MnO_2 , suggesting a reaction between VOSO₄ and MnO_2 [Figure 1E]. Furthermore, we examined the discharge curves of the VOSO₄ solution, MnO_2 suspension, and a mixture of VOSO₄ and MnO_2 . Individually, VOSO₄ and MnO_2 exhibited negligible capacities [Figure 1F], attributed to the reductive state of VO²⁺ and the inadequate contact between MnO_2 suspended in the electrolyte and the carbon felt. However, when the mixture was formed, it became transparent and displayed a plateau at 0.70 V, indicating MnO_2 oxidizes VO²⁺ to VO₂⁺, which can subsequently undergo electrochemical reduction.

Therefore, the working principle of the VOSO₄ mediator to improve the dissolution of MnO₂ is depicted in Figure 2B. During the charging progress, the oxidation of VO²⁺ precedes the deposition of MnO₂ on the carbon felt, generating VO₂⁺ [Equation (3)]. Conversely, during discharge, MnO₂ dissolves into Mn²⁺ prior to the reduction of VO₂⁺. The reduction product VO²⁺ then reacts with the residual MnO₂, leading to the formation of VO₂⁺ and Mn²⁺ [Equation (4)]. Moreover, the resulting VO₂⁺ is subjected to another round of electrochemical reduction [Equation (3)], steadily providing a continuous supply of VO²⁺ to react with MnO₂ until all the remaining MnO₂ is consumed. As a result, this process promotes the reversibility of the

MnO₂ deposition and dissolution.

$$VO^{2+} + H_2O \leftrightarrow VO_2^+ + 2H^+ + e^-$$
(3)

$$MnO_2 + 2VO^{2+} \rightarrow Mn^{2+} + 2VO_2^{+}$$
(4)

To investigate the impact of the VOSO₄ mediator in the MnO₂-Zn full cell, we introduced VOSO₄ into the catholyte of a two-electrode configuration [Supplementary Figure 5]. First, we assessed the capacity contribution and stability of 0.05 M VOSO4 in the MnO2-Zn full cell. The VOSO4-Zn batteries exhibit excellent cyclic stability, retaining a reversible capacity of 0.3 mAh cm⁻² even after 3,000 cycles [Supplementary Figure 6]. The MnO₂/VOSO₄-Zn battery, charged to 1 mAh cm⁻² at 10 mA cm⁻², exhibits stable charge/discharge performance for over 3000 cycles, surpassing the cyclic life of the MnO,-Zn battery [Figure 3A and Supplementary Figure 7]. When the charging capacity increases to 5 mAh cm⁻², the MnO₂-Zn battery exhibits a high discharge plateau at 1.95 V and coulombic efficiency of 95% at 10 mA cm⁻² in the first cycle. However, after 150 cycles, the discharge plateau declines to 1.69 V, and the gap between the charge and discharge plateaus widens to 0.76 V [Figure 3B]. This decline in performance can be attributed to the accumulation of undissolved MnO₂, leading to serious polarization and high charge transfer resistance [Supplementary Figure 8], which results in a substantial loss in capacity. In comparison, the use of the VOSO₄ mediator reduces the discharge voltage. However, the MnO₂/VOSO₄-Zn battery demonstrates remarkable stability in charge/discharge cycles, exceeding 900 cycles and maintaining a high coulombic efficiency of approximately 98% [Figure 3C and D]. On the other hand, the energy efficiency of the MnO₂-Zn battery gradually declines over cycling, averaging around 60%. In contrast, the MnO₂/VOSO₄-Zn battery displays a stable energy efficiency of approximately 75% throughout cycling [Supplementary Figure 9].

Figure 3E and F compares the rate capability of the MnO_2 -Zn battery and $MnO_2/VOSO_4$ -Zn battery, ranging from 5 to 50 mA cm⁻². As a current density of 50 mA cm⁻², the $MnO_2/VOSO_4$ -Zn battery exhibits a discharge plateau at 1.77 V, maintaining a reversible capacity of 3.50 mAh cm⁻² and an energy density of 5.42 mWh cm⁻². In contrast, the MnO_2 -Zn cell displays a lower discharge voltage of 1.67 V, with a reversible capacity of 3.09 mAh and an energy density of 4.63 mWh cm⁻². Additionally, even when charged to 10 mAh cm⁻² at 1 C [Figure 3G], the $MnO_2/VOSO_4$ -Zn battery exhibits a high coulombic efficiency of 97 % and an energy efficiency of 79% after 500 cycles [Supplementary Figure 10]. On the other hand, the MnO_2 -Zn battery fails to deliver capacity after 100 cycles. These comparisons of the charge/discharge performance clearly highlight the effectiveness of $VOSO_4$ as a mediator in enhancing the cyclic stability of the MnO_2 -Zn battery.

To investigate the influence of VOSO₄ on the deposition and dissolution behavior of Mn^{2+} , extensive characterization was performed on the cathodes of the $MnO_2/VOSO_4$ -Zn and MnO_2 -Zn batteries using XRD, SEM, and XPS after charge and discharge cycles. Following the 20th charging at 5 mAh cm⁻² and 10 mA cm⁻², diffraction peaks corresponding to ϵ -MnO₂ (JCPDS #30-0820) can be observed in the XRD pattern of the cathode from the $MnO_2/VOSO_4$ -Zn battery [Figure 4A]. The broad peaks with low intensity indicate low crystallinity of the deposited ϵ -MnO₂. SEM images [Figure 4B and C] reveal a dense layer of MnO₂ layer composed of nanoparticles. XPS analysis of the charged cathode shows peaks at 84.16 and 89.26 eV in the Mn 3s region, corresponding to Mn^{4+} [Figure 4D]. Peaks at 529.9 and 531.3 eV in the O1s region correspond to the Mn-O-Mn and Mn-OH bonds, respectively [Figure 4E]^[12]. Notably, the energy splitting of Mn⁴⁺ peaks in the charged cathode of the MnO₂/VOSO₄-Zn battery is higher than that in the MnO₂-Zn battery (5.10 eV *vs.* 4.99 eV) [Supplementary Figure 11], suggesting the formation of lower-



Figure 3. Electrochemical performance of the MnO_2 -Zn and $MnO_2/VOSO_4$ -Zn batteries: (A) Cycling performance at 1 mAh cm⁻², (B and C) charge and discharge curves, and (D) cyclic performance at 5 mAh cm⁻² and 10 mA cm⁻², (E and F) discharge curves at 5 mAh cm⁻² and various current densities, and (G) cyclic performance at 10 mAh cm⁻² and 10 mA cm⁻².



Figure 4. Characterizations of the cathode in the $MnO_2/VOSO_4$ -Zn battery in the 20th cycle: (A) XRD patterns at different states, (B) SEM and (C) EDS elemental mapping images after charge, (F) SEM image after discharge, XPS spectra in the region of (D) Mn 3s and (E) O 1s.

valence Mn during the deposition^[33,39]. The VOSO₄ mediator introduces more defects in MnO₂, thereby promoting its dissolution. After subsequent discharge to 1.0 V, the peaks corresponding to ε -MnO₂ disappear in the XRD pattern [Figure 4A], and the surface of the carbon fiber appears bare and smooth in the SEM image [Figure 4F, Supplementary Figure 12]. Additionally, the peaks associated with Mn⁴⁺, Mn-O-Mn, and Mn-OH disappear in the XPS spectrum, indicating good reversibility of MnO₂ deposition and dissolution. After 100 cycles, the ε -phase MnO₂ peaks are nearly absent in the XRD pattern of the cathode [Figure 5A], and only a few residual MnO₂ particles are observed in the SEM image [Figure 5B and C]. These SEM and XRD measurements conclusively confirmed that MnO₂ can be completely dissolved by the VOSO₄ mediator during cycling.

Regarding the cathode of the MnO_2 -Zn battery, the XRD pattern consistently exhibits increasingly intense diffraction peaks corresponding to MnO_2 after 20 cycles [Supplementary Figure 13] and 100 cycles [Figure 5D]. SEM images reveal a remaining MnO_2 layer on the carbon fiber with a thickness of approximately 6 µm after 20 cycles [Supplementary Figures 14 and 15], which further grows to 14 µm after 100 cycles [Figure 5E and F]. Furthermore, some MnO_2 particles even detach from the carbon felt, obstructing ion transport channels on the membrane [Figure 6A]. The accumulation and detachment of MnO_2 ultimately lead to the failure of the MnO_2 -Zn battery.

Figure 6A provides a comparison of the Nafion membranes from the $MnO_2/VOSO_4$ -Zn and MnO_2 -Zn batteries after 100 cycles. The membrane in the $MnO_2/VOSO_4$ -Zn battery remains transparent, while the membrane in the MnO_2 -Zn battery turns black due to the migration of MnO_2 . However, the blackened membrane regains transparency after immersion in E-MnVO. The MnO_2 -Zn battery experienced failure after 218 cycles. When $VOSO_4$ is added to the catholyte of the failed MnO_2 -Zn battery, the battery exhibits a discharge capacity of 31 mAh cm⁻². Because a significant amount of MnO_2 accumulated on the carbon felt or suspended within the catholyte reacts with VO^{2+} ions, generating VO_2^+ ions, which contribute discharge capacity through the electrochemical reduction. After $VOSO_4$ completely transforms the accumulated MnO_2 into Mn^{2+} , the battery is successfully revived and demonstrates stable charge/discharge behavior in subsequent cycles [Figure 6B and C]. Conversely, replacing the zinc foil in the failed MnO_2 -Zn battery does not resolve the abnormal charge/discharge performance [Supplementary Figure 16]. This comparison indicates that the failure of the MnO_2 -Zn battery is primarily attributed to the accumulation of undissolved MnO_2 rather than the degradation of the zinc anode. Hence, the use of the $VOSO_4$ mediator proves effective in extending the lifespan of the MnO_2 -Zn battery.

Additionally, we introduced the VOSO₄ mediator into the catholyte of the MnO_2 -Zn flow battery [Supplementary Figure 17], resulting in the achievement of even higher area capacity. As depicted in Figure 6D and E, when the charge capacity is increased to 20 mAh cm⁻², the flow battery exhibits stable cyclic performance with a coulombic efficiency of approximately 97% and an energy density of 30.32 mWh cm⁻² after 300 cycles. Remarkably, even at 30 mAh cm⁻², the flow battery achieves 90 stable cycles with excellent reversibility [Supplementary Figure 18]. Furthermore, the summarized results in Figure 6F and Supplementary Table 1 illustrate that our assembled $MnO_2/VOSO_4$ -Zn battery displays outstanding performance in terms of areal capacity, reversibility, and cycle life compared with the reported electrolytic MnO_2 -based batteries. To upscale the battery, a carbon felt measuring 64 cm² (8 cm × 8 cm) was utilized as the current collector for the cathode, resulting in the assembly of a larger-scale $MnO_2/VOSO_4$ -Zn battery demonstrates good cyclic stability, maintaining approximately 500 mAh and 751 mWh over 20 cycles, highlighting the potential of the $MnO_2/VOSO_4$ -Zn battery for practical energy storage applications.



Figure 5. Comparisons of the cathodes in the $MnO_2/VOSO_4$ -Zn and MnO_2 -Zn batteries after cycling: (A) XRD patterns, SEM images after (B) 10 and (C) 100 cycles of the cathode in the $MnO_2/VOSO_4$ -Zn battery. (D) XRD patterns, SEM images after (E) 10 and (F) 100 cycles of the cathodes in the MnO_2 -Zn battery.



Figure 6. (A) Comparisons of Nafion membranes obtained from the cycled $MnO_2/VOSO_4$ -Zn and MnO_2 -Zn batteries before and after immersion in E-MnVO. (B) Discharge curves and (C) cyclic performance of the failed MnO_2 -Zn battery after addition of $VOSO_4$ at 5 mAh cm⁻² and 10 mA cm⁻². (D) Charge/discharge curves and (E) cyclic performance of $MnO_2/VOSO_4$ -Zn flow battery at 20 mAh cm⁻² and 10 mA cm⁻². (F) Comparison of the electrochemical performance of reported electrolytic MnO_2 -based batteries.

CONCLUSIONS

In conclusion, our study demonstrates significant improvements in the reversibility and cycling stability of MnO_2 -Zn batteries through the utilization of redox mediator $VOSO_4$. The oxidation reaction involving VO^{2+} effectively interacts with residual MnO_2 , facilitating its dissolution. As a result, when this mediator

strategy is adopted in the MnO_2 -Zn battery, it exhibits an impressive long cycle life of 500 cycles, even at a high area capacity of 10 mAh cm⁻², while maintaining a coulombic efficiency of approximately 97%. Furthermore, the $MnO_2/VOSO_4$ -Zn flow battery demonstrates excellent cycle stability, maintaining performance over 300 cycles at 20 mAh cm⁻². These findings provide valuable insights into the potential large-scale application of electrolytic MnO_2 -Zn batteries.

DECLARATIONS

Authors' contributions

Investigation, data curation, writing - original draft: Xu Y Formal analysis: Huang W, Liu J, Hu R, Ouyang L Resources, supervision, writing - review & editing, project administration: Yang L Resources, project administration: Zhu M

Availability of data and materials

The data supporting our findings can be found in the Supplementary Material.

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Conflicts of interest

All authors declared that there are no conflicts of interest.

Ethical approval and consent to participate

Not applicable.

Consent for publication

Not applicable.

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