

Review

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Catalyzed carbon-based materials for CO₂-battery utilization

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Abstract

Increasing atmospheric CO₂ levels and global carbon neutrality goals have driven interest in technologies that both mitigate CO₂ emissions and provide sustainable energy storage solutions. Metal-carbon dioxide (M-CO₂) batteries offer significant promise due to their high energy density and potential to utilize atmospheric CO₂. A key challenge in advancing M-CO₂ batteries lies in optimizing CO₂-breathing cathodes, which are essential for CO₂ adsorption, diffusion, and conversion. Carbon-based cathodes play a critical role in facilitating CO₂ redox for M-CO₂ batteries, owing to their cost-effectiveness, high conductivity, tunable microstructure, and porosity. However, there is a lack of current systematic understanding of the relationship between the structure, composition, and catalytic properties of carbon-based cathodes, as well as their impact on the overall efficiency, stability, and durability of M-CO₂ batteries. In this review, we will give an insightful review and analysis of recent advances in various carbon-based materials, including commercial carbons, single-atom catalysts, transition metal/carbon composites, metal-organic frameworks, *etc.*, focusing on their structure-function-property relationships. A comprehensive understanding of the pivotal role played by carbon-based materials and their optimization strategies in M-CO₂ batteries will be provided. Moreover, future perspectives and research suggestions for carbon-based materials are presented to advance the development and innovation of M-CO₂ batteries.

Keywords: Carbon-based materials, electrocatalysts, CO₂ utilization, CO₂ batteries, CO₂ reduction reaction



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INTRODUCTION

The global rise in atmospheric carbon dioxide (CO₂) levels has become an urgent environmental concern, necessitating sustainable carbon management strategies. Many countries have committed to carbon neutrality, setting ambitious targets to achieve net-zero emissions by mid-century^[1-3]. For instance, the European Union aims for carbon neutrality by 2050, China by 2060, and the United States pledges a 50%-52% reduction in greenhouse gas emissions by 2030 compared to 2005 levels. Achieving these targets requires clean energy technologies, such as solar and wind power^[4-6]. However, the intermittent nature of these energy sources, combined with growing energy demands, highlights the need for advanced energy storage systems to ensure a stable power supply.

Metal-carbon dioxide (M-CO₂) batteries represent a promising solution, utilizing CO₂ as a reactant for energy storage and offering high energy densities - up to 1,876 Wh/kg for Li-CO₂ and 1,125 Wh/kg for Na-CO₂ batteries^[7-9]. These batteries address both CO₂ reduction and energy storage challenges, making them revolutionary for sustainable energy applications. Efficient CO₂-breathing cathodes are critical to their performance, facilitating CO₂ adsorption, diffusion, and conversion. Gas diffusion electrodes (GDEs), such as carbon paper or cloth, are commonly used to transport CO₂ to reaction sites, with electrocatalysts driving the conversion processes during charging and discharging^[10-12]. Carbon-based materials are the leading candidates for M-CO₂ battery cathodes due to their low cost, high electrical conductivity, large surface area, and structural tunability. These materials can be engineered through doping, surface functionalization, and structural modifications to enhance their catalytic properties. However, challenges remain: commercial carbon materials, such as Ketjen Black (KB) and Super P, exhibit limited catalytic activity due to their inert nature and lack of active sites, as well as insufficient structural robustness to accommodate discharge products^[13,14]; Single-atom catalysts (SACs), despite their high theoretical activity, are prone to forming agglomerated nanoparticles under operational conditions, leading to a loss of catalytic efficiency^[15,16]; Transition metal/carbon composites often suffer from poor chemical and thermal stability, especially under high voltages and long-term cycling^[17-19]; Similarly, metal-organic frameworks (MOFs), though offering high porosity and tunable structures, are susceptible to structural collapse or deactivation during catalytic processes, particularly in harsh electrochemical environments^[20,21]. Addressing these limitations requires a deeper understanding of the interplay between structural and compositional properties and catalytic performance. The pore structure and surface area of carbon materials significantly influence CO₂ adsorption and diffusion^[22], while functional groups or dopants can enhance catalytic activity by altering electronic structures. Robust carbon materials capable of withstanding high voltages, current densities, and long-term cycling are a major focus of research^[7,23,24]. Similarly, interactions between carbon and transition metals in composite materials can enhance catalytic efficiency but require precise control of composition and nanostructures.

Despite progress in the field, a comprehensive analysis of the functional applications of carbon-based materials in M-CO₂ batteries remains lacking. This review aims to bridge that gap, providing a detailed examination of recent advancements in carbon-based cathode materials, their catalytic mechanisms, and functional applications. Strategies for enhancing catalytic efficiency, including innovative material designs and advanced fabrication techniques, will also be explored. By summarizing the relationships between material properties and battery performance, this review seeks to guide future research efforts and accelerate the commercialization of M-CO₂ batteries as a sustainable energy technology.

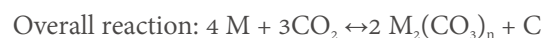
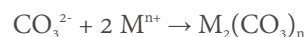
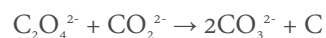
FUNDAMENTAL UNDERSTANDING OF M-CO₂ BATTERIES

Battery reaction mechanism

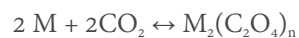
Electrochemistry of nonaqueous M-CO₂ batteries

A typical nonaqueous M-CO₂ battery consists of three main components: a metal anode, a porous CO₂-breathing cathode, and a separator containing a metal-ion-conducting electrolyte^[25]. Figure 1A provides a schematic illustration of this configuration. Unlike the reversible insertion of Li⁺ or Na⁺ ions between the electrodes seen in traditional Li-ion or Na-ion batteries, the energy storage and release mechanism in M-CO₂ batteries is based on conversion reactions^[26-29]. During discharge, the metal anode is oxidized, forming metal ions (Mⁿ⁺, where n represents the oxidation state of metal ions), which then migrate through the organic electrolyte to the cathode. The movement direction of Mⁿ⁺ is indicated by the green arrow. Concurrently, CO₂ is reduced at the cathode, and the resulting reduction products react with the migrating metal ions to generate the final discharge product. This is referred to as the CO₂ reduction reaction (CO₂RR). During charging, the reactions reverse. When the battery is charged, the reverse occurs: discharge products are decomposed, releasing Mⁿ⁺ and CO₂, and the Mⁿ⁺ re-moves from the cathode to the anode (in the direction of the orange arrows), with metal being deposited back onto the anode and CO₂ evolving at the cathode. This process is commonly referred to as the CO₂ evolution reaction (CO₂ER)^[30].

In nonaqueous M-CO₂ batteries, the CO₂RR can proceed *via* either a two-electron or a four-electron transfer pathway, resulting in distinct discharge products^[7]. The two-electron transfer pathway typically leads to the formation of metal oxalate [M₂(C₂O₄)_n], while the four-electron transfer process yields a combination of metal carbonate and carbon as the discharge products. The mechanism involving the four-electron transfer pathway is now widely recognized, and the corresponding reaction equations can be summarized as follows: corresponding reactions can be expressed as follows:



Experimentally, significant progress has been achieved in understanding the electrochemistry of the four-electron transfer pathway in M-CO₂ batteries, and several physicochemical factors influencing the reaction mechanisms have been studied^[31-33]. When studying intermediate discharge products, the intermediate M₂(C₂O₄)_n is usually further converted to metal carbonate due to its poor thermal stability. Interestingly, it was discovered that under specific conditions, the intermediate product C₂O₄²⁻ can be stabilized, implying that CO₂RR proceeds via the two-electron transfer pathway, and the corresponding reaction process is as follows:



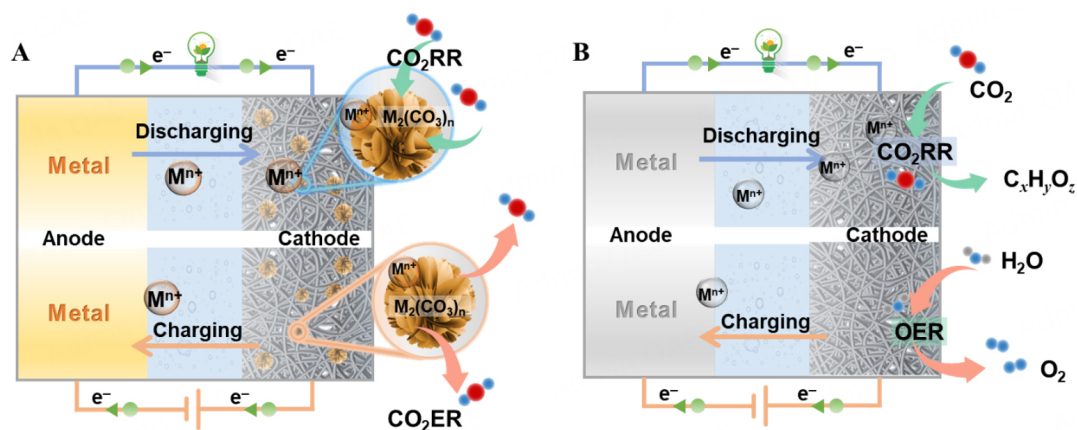


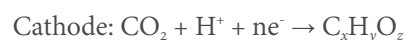
Figure 1. Schematic illustration of (A) Nonaqueous and (B) Aqueous M-CO₂ battery configuration.

For example, when using a molybdenum carbide (Mo₂C) catalyst, it was found that Mo₂C stabilizes the amorphous intermediate discharge product Li₂C₂O₄ through the delocalized electrons generated by Mo-O coupling^[34,35]. Zhou *et al.* demonstrated that Mo-O chemical bonds form between Mo₂C and C₂O₄²⁻ during discharge processes^[36]. This interaction allows delocalized electrons from Mo²⁺ and Mo³⁺ in Mo₂C to transfer to the electronegative oxygen atoms in C₂O₄²⁻, thereby stabilizing Li₂C₂O₄ as a discharge product. Notably, Li₂C₂O₄ decomposes thermodynamically more easily during charging ($E^\circ = 3.01$ V vs. Li/Li⁺) compared to Li₂CO₃ ($E^\circ = 3.82$ V vs. Li/Li⁺), reducing the charge potential to approximately 3.4 V. In contrast, catalysts such as carbon nanotubes (CNTs) lack this stabilization capability, causing C₂O₄²⁻ to disproportionate into CO₃²⁻ and carbon, leading to the formation of Li₂CO₃, which is thermodynamically stable but highly insulating and difficult to decompose below 4.0 V. Furthermore, experimental results, including a precipitation test with CaCl₂ and H₃PO₄, confirm the presence of C₂O₄²⁻ in Mo₂C-based systems but not in CNT-based systems. Density functional theory (DFT) calculations demonstrated the superior adsorption energy of Mo₂C for Li and CO₂, as well as its metallic conductivity and ability to stabilize Li₂C₂O₄. Charge density difference analyses further revealed significant electron transfer from Mo₂C to Li₂C₂O₄, unlike CNTs, which lack this capability. As a result, Li₂C₂O₄ forms as a stable discharge product with Mo₂C, while Li₂CO₃ is the primary discharge product for CNTs. Currently, few catalysts, including Mo₂C and MoN, have shown the ability to stabilize M₂(C₂O₄)_n. Enhancing interactions between Li⁺ ions and electrolyte molecules also promotes the formation of stable solution-phase C₂O₄²⁻ while suppressing the direct reduction of CO₂ to Li₂CO₃^[37]. Thermodynamically, M₂C₂O₄ decomposes more readily than M₂CO₃, reducing charging voltage, facilitating reversibility, and enabling stable M-CO₂ battery operation^[38-40]. DFT studies on Ru and Au catalysts in Li-CO₂ batteries further support this. Li₂C₂O₄ emerges as a kinetically favorable intermediate, with Li₂CO₃ as the thermodynamically stable product. Ru outperforms Au due to lower Gibbs free energy for nucleation and reduced activation barriers for rate-limiting steps, enabling efficient conversion of Li₂C₂O₄ into Li₂CO₃ and carbon along distinct reaction pathways^[41].

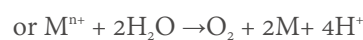
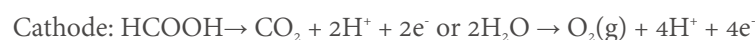
Electrochemistry of aqueous M-CO₂ batteries

Aqueous M-CO₂ batteries are gaining particular attention due to their feature of simultaneously reducing CO₂ to value-added carbonaceous products and energy storage. As illustrated in Figure 1B, A typical aqueous M-CO₂ battery consists of a metal anode, a catalytically active cathode, and a separator (e.g., solid electrolyte for Li/Na-CO₂, or bipolar membrane for Zn/Al/Mg-CO₂ batteries) for separating the anolyte and catholyte. The main function of the anode chamber is to dissolve and deposit metals at the anode for energy release and storage during discharge and charging, respectively, similar to nonaqueous M-CO₂ batteries. The cathode chamber is similar to that of the electrolyzer of direct electrocatalytic CO₂ reduction, and thus

offers unique advantages for CO₂ conversion in the CO₂RR process based on the proton-coupled electron transfer mechanism that produces a wide range of carbon-containing products (C_xH_yO_z, such as HCOOH, CO, CH₄, CH₃OH, *etc.*), and the corresponding discharge reaction is as follows:



During the charging phase of aqueous M-CO₂ batteries, depending on the discharge product, the reaction may involve reversible conversion of the discharge product (mainly for HCOOH) or oxidation of H₂O [oxygen evolution reaction (OER)] (mainly for other difficult to oxidize products such as CO₂ or CH₄, *etc.*). The charging reaction can be expressed as follows:



It is worth mentioning that (i) the fundamental difference between aqueous and nonaqueous M-CO₂ batteries lies in their reaction mechanisms, which are strongly influenced by the electrolyte environment. In nonaqueous M-CO₂ batteries, the reaction mechanism primarily involves the reduction of CO₂ to form solid discharge products such as M₂(CO₃)_n + C or M₂(C₂O₄)_n. These solid products are decomposed reversibly during charging, releasing CO₂ again. In contrast, in aqueous M-CO₂ batteries, the reaction is more similar to direct electrocatalytic CO₂RR to produce carbon-containing products (C_xH_yO_z) due to the aqueous medium. Upon charging, oxidation of HCOOH or H₂O occurs depending on the type of discharge product.

(ii) Nonaqueous systems, typically composed of metal salts and solvents such as tetraethylene glycol dimethyl ether (TEGDME), leverage the low volatility, wide electrochemical window, and high chemical stability of ether-based electrolytes, making TEGDME a representative solvent widely used in nonaqueous M-CO₂ batteries^[7,24]. In contrast, aqueous M-CO₂ batteries, such as Zn-CO₂ batteries, feature an alkaline anolyte [e.g., KOH mixed with Zn(Ac)₂] to promote Zn redox reactions and a near-neutral catholyte [e.g., KHCO₃ solution or KHCO₃ mixed with Zn(Ac)₂] to favor CO₂RR, as alkaline conditions conflict with CO₂RR pathways^[42]. The two electrolytes are separated by a bipolar membrane to maintain their respective chemical environments, ensuring optimal reaction conditions at both electrodes.

(iii) The electrochemical performance of both nonaqueous and aqueous M-CO₂ batteries is heavily influenced by the properties of the cathode catalyst materials. In nonaqueous M-CO₂ batteries, M₂(CO₃)_n + C is widely recognized as the main discharge product. However, it is found that under the influence of some special catalysts, the intermediate M₂(C₂O₄)_n does not decompose further and can be stabilized while forming the final product. For example, when using a molybdenum carbide (Mo₂C) catalyst, it was found that Mo₂C stabilizes the amorphous intermediate discharge product Li₂C₂O₄ through the

delocalized electrons generated by Mo-O coupling^[34]. Only a few reported catalysts (e.g., Mo₂C and MoN) have been able to stabilize the production of M₂(C₂O₄)_n. Therefore, catalysts that promote the stabilization of solid M₂(C₂O₄)_n shall be preferred due to their high catalytic efficiency, and ability to stabilize intermediates. In addition, it is essential to develop catalysts that promote the reversible generation and decomposition of discharge products, such as M₂(CO₃)_n + C. These catalysts can reduce charging polarization, thereby improving energy efficiency and prolonging battery life. For aqueous M-CO₂ batteries, catalysts that exhibit high selectivity and efficiency for CO₂ reduction to specific products are preferred. Examples include metal-based catalysts (e.g., Au and Ag) that can produce CO, and Sn-based catalysts that can produce HCOOH. Moreover, this catalyst needs to have OER catalytic activity to facilitate the charging reaction and reduce energy consumption.

Cathode engineering in CO₂-batteries

Fundamental challenges of cathode in M-CO₂ batteries

CO₂-breathing cathodes are the primary site of CO₂ capture, conversion, product formation, decomposition or release, involving multi-phase interfaces and complex reactions^[18,43-46]. Typically, sufficiently efficient electron and mass (e.g., Mⁿ⁺ and CO₂) transfers and the abundance of catalytically active sites at the three-phase interface of the CO₂-breathing cathode are required to ensure that CO₂ is converted efficiently, so that the high performance of M-CO₂ batteries can be realized. Therefore, significant challenges arise and need to be addressed before their practical application [Figure 2].

(i) Limited CO₂ availability and utilization: ensuring a continuous and sufficient supply of CO₂ to the cathode is critical for the sustained operation of M-CO₂ batteries. However, the cathode structure (e.g., surface area, porosity, wettability, etc.) may be suboptimal, with uneven distribution of surface reaction sites, low reactivity, or a catalyst surface that is easily covered by products, making it difficult for CO₂ to adequately reach the reaction sites or to overflow before it has been reacted, leading to a lower efficiency of CO₂ utilization. Catalyst surfaces may also become blocked by discharge products, reducing access to active reaction sites. In aqueous systems, the hydrogen evolution reaction (HER) competes with CO₂RR^[47,48], consuming energy and further decreasing CO₂ conversion efficiency due to the favorable reduction of H₂O to H₂ gas.

(ii) Sluggish reaction kinetics of CO₂RR: CO₂ is chemically stable, with a strong C=O bond (750 kJ mol⁻¹) necessitating high overpotentials for activation. The CO₂RR process involves multi-step electron transfers and the formation, dissolution, and adsorption of intermediates. Suboptimal interactions between the cathode material and these intermediates often lead to kinetic bottlenecks. Ideal cathode materials should effectively adsorb CO₂ and intermediates, facilitating their reduction while maintaining optimal binding strength.

(iii) Clogging of CO₂-breathing cathodes: in aqueous M-CO₂ batteries, the discharge products are dissolved in the electrolyte. However, in nonaqueous M-CO₂ batteries, insoluble carbonate discharge products (e.g., Li₂CO₃ and Na₂CO₃) precipitate on the cathode surface or within its pores, clogging channels and blocking CO₂ and ion diffusion. The limited solubility and low diffusivity of CO₂ in nonaqueous electrolytes exacerbate uneven product deposition, concentrating discharge products near the CO₂ source. This spatial imbalance leads to increased charge transfer resistance, higher overpotentials, and significant capacity decline over time, ultimately reducing battery lifespan and practical viability.

(iv) Cathode corrosion and degradation: high charging voltages required to decompose insulating carbonate products can cause undesirable side reactions, including electrolyte decomposition and cathode corrosion.

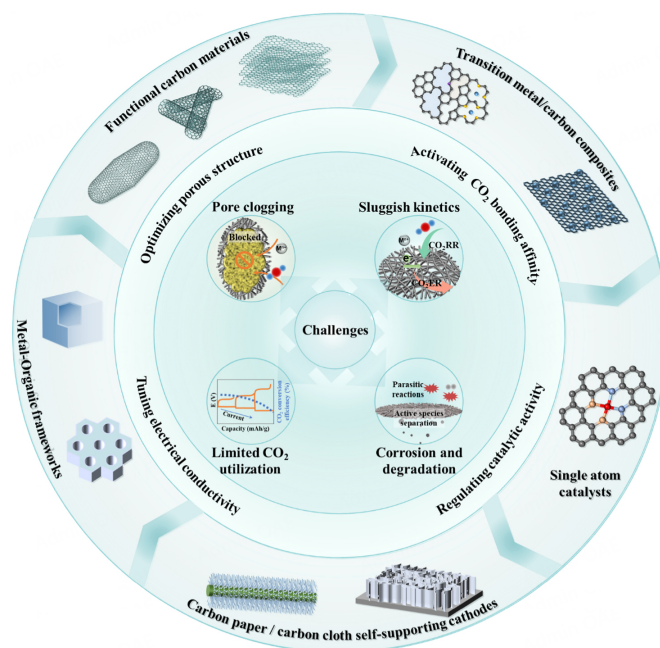


Figure 2. Challenges, structural design principles and typical carbon-based materials of CO₂-breathing cathodes for M-CO₂ batteries.

Carbon-based materials may corrode due to reactive oxygen species, while metal cathodes can form passivating oxides or hydroxides in the presence of H₂O, O₂ and CO₂. Repeated redox cycling can induce stress accumulation, microstructural changes, or crack formation, invalidating active sites and limiting performance. At the nanoscale, local electric fields and structural remodeling can further degrade the catalyst, diminishing system efficiency and durability.

Structural design of cathode in M-CO₂ batteries

To clearly address the aforementioned challenges, a well-designed CO₂-breathing cathode should fulfill several key functions, including facilitating the diffusion of Mⁿ⁺ ions and CO₂, catalyzing the formation and decomposition of the discharge products, providing sufficient storage space for these products, and guiding their morphological evolution. To meet these functional requirements, several critical performance metrics must be optimized: (i) High electrical conductivity and an optimized porous structure are key for CO₂ cathodes. Conductivity ensures fast electron transfer, reducing internal resistance and boosting power output. A well-designed porous structure helps diffuse Mn⁺ ions and CO₂ and stores discharge products. Mesopores (2-50 nm) can facilitate CO₂ and ion transport, while macropores (> 50 nm) provide ample space for the storage of insulating discharge products such as Li₂CO₃ or Na₂CO₃, reducing impedance and enhancing cycle performance^[49]. One study shows that combining micro- and mesopores increases surface area (2,003 vs. 1,813 m²g⁻¹), enhancing ion and gas movement. However, excessive activation reagents can collapse the structure, reducing surface area. A balance between pore size and volume is crucial^[50]. Therefore, a balance between pore size and volume is desirable; (ii) Favorable CO₂ bonding affinity. An ideal CO₂ cathode should possess a high binding affinity for CO₂ to facilitate its adsorption and activation, which helps reduce the energy barriers associated with CO₂RR, improving overall reaction kinetics and battery performance; and (iii) Excellent catalytic activity. To achieve high electrochemical performance, catalytic cathodes should possess abundant and accessible active sites for CO₂RR and CO₂ER, to lower the overpotential during both charge and discharge cycles, and to improve the overall efficiency of the battery. Whereas in aqueous M-CO₂ batteries, depending on their charging/discharging mechanism, effective CO₂RR/CO₂ER or CO₂RR/OER bifunctional electrocatalysts are required on the CO₂ cathode.

Importance and strategies of carbon-based cathode for M-CO₂ batteries

As mentioned earlier, the remarkable electrical conductivity, low mass density, affordability, significant specific surface area, customizable porous structure, and macroscopic morphology contribute to the extensive investigation of carbon materials in M-CO₂ batteries. For example, Super P and KB with inherent high conductivity have been widely recognized as conductive agents, which are essential for electron transfer in M-CO₂ batteries^[14,45,51]. In addition, carbon materials (e.g., CNTs and graphene) have a tunable porous structure allowing for optimized pore size distribution and a large specific surface area, which contributes to the efficient adsorption of CO₂, thus ensuring the efficient diffusion of CO₂ and Mⁿ⁺ ions and providing space for the storage of discharge products. Additionally, carbon paper and carbon cloth serve as current collectors or gas diffusion layers, providing a conductive framework that allows for the efficient electron transfer from the external circuit to the active sites, while facilitating the diffusion of CO₂ into the catalyst and enhancing the supply of CO₂ at the active sites^[52-54]. Consequently, the presence of carbon materials is indispensable in M-CO₂ batteries.

However, pure carbon materials have limited catalytic activity in promoting CO₂RR and CO₂ER processes, which may lead to higher overpotentials. Carbon materials alone may not be sufficient to achieve the high catalytic activity required for practical M-CO₂ batteries^[55-57]. To overcome this limitation, various integrated strategies have been reported: (a) Nanostructure engineering to design conductive networks with hierarchical porous structures to alleviate pore clogging and promote sustained Mⁿ⁺ and CO₂ transport; (b) constructing of CO₂ cathodes with sufficient electrocatalytic activity, e.g., doping (e.g., heteroatom doping such as N, S, B, *etc.*), metal/carbon composites, single atom (SA) catalysts, MOFs or, to enhance their CO₂ bonding affinity and catalytic activity; and (c) designing self-supporting cathodes for surface protection of the carbon layer to mitigate the degradation of the carbon cathode.

CATALYZED CARBON-BASED MATERIALS IN CO₂-BATTERIES APPLICATION

In this section, our primary focus will be on exploring the application of carbon-based materials in M-CO₂ batteries, with an emphasis on their inherent advantages in achieving high-performance M-CO₂ batteries. The main objective is to gain a comprehensive understanding of how carbon materials with diverse microstructures, surface properties, and compositions contribute significantly to enhancing the electrochemical performance of M-CO₂ batteries. Furthermore, potential future research directions for improving M-CO₂ battery performance using carbon-based materials will be proposed.

Commercial carbon materials

Ketjen black and super P

Commercial carbon materials, such as KB and Super P, are widely used as cathode materials for M-CO₂ batteries due to their favorable electronic conductivity, large surface area, chemical stability, low cost, and scalable manufacturing processes. While Super P was among the first carbon materials used as a cathode in M-CO₂ batteries, the electrolyte plays a critical role in determining overall battery performance, even when the electrode materials remain the same. For instance, Ci *et al.* found that Super P had negligible discharge capacity in ionic liquid (IL) electrolytes^[58]. In contrast, Yang *et al.* demonstrated significantly improved discharge capacity (6,062 mAh g⁻¹) in a Li-CO₂ battery using Super P with an ether-based electrolyte. This capacity further increased to 8,229 mAh g⁻¹ with the addition of Ru metal (Ru@Super P)^[59]. Similarly, Guo *et al.* developed a porous KB-supported Ru nanoparticle (Ru@KB) composite that enhances the decomposition of Na₂CO₃, lowering charge voltage and improving cycling performance in Na-CO₂ batteries^[60]. With the Ru@KB cathode, the batteries achieved a high discharge capacity of 11,537 mAh g⁻¹ and a Coulombic efficiency of 94.1%, and ran for over 130 cycles with stable performance^[60]. The disparity in performance is attributed to the electrolyte's effect on the stability of intermediate discharge products and

the formation of final products. Early studies on Na-CO₂ batteries with Super P cathodes in different electrolytes confirmed this trend. However, their electrochemical performance is limited by inherent drawbacks, including low conductivity, small pore volume, and limited active sites.

Carbon nanotubes

Commercial CNTs, featuring a porous, cross-linked structure with 5 nm wall thickness and 0.34 nm interlayer spacing, are integral to M-CO₂ batteries. The outer CNT walls facilitate nucleation and anchoring of M₂(CO₃)_n products, while inner walls enhance conductivity and electron transfer. Xiao *et al.* examined the discharge and charge behavior of Li-CO₂ batteries with CNT electrodes under varying current densities [Figure 3A]^[61]. Voltage profiles revealed two stages: an initial low-voltage plateau decreasing during charging and a sharp rise to completion [Figure 3B]. |dV/dQ| analysis identified three phases: rapid polarization (L0), characteristic peaks (L1), and stabilization (L2). Lower currents prolonged L1 but increased crystalline Li₂CO₃ formation, raising resistance and irreversibility. Pure CNT cathodes often exhibit high charging voltages and electrolyte decomposition^[62]. However, multi-walled CNTs (MWCNTs) have demonstrated improved performance, such as a Li-CO₂ battery with Au nanoparticle-supported CNTs achieving 6,399 mAh g⁻¹ at 100 mA g⁻¹^[63]. Further advancements will be discussed in the next chapter.

Graphene

Graphene, with its excellent conductivity, large surface area, and high stability, is an ideal cathode material for Li-CO₂ batteries^[64,65]. Its structure enables efficient oxygen diffusion, ample discharge product storage, and abundant active sites. Zhang *et al.* developed ultrathin, porous graphene nanosheets with enhanced electrolyte wettability and CO₂ diffusion, achieving discharge capacities of 14,722 and 6,600 mAh g⁻¹, with a stable 2.77 V plateau at 50 mA g⁻¹^[66]. However, the efficiency and reaction kinetics of graphene-based cathodes remain moderate. Strategies such as heteroatom doping, defect engineering, and compositing with metals or non-metals are being explored to further boost the electrochemical performance of graphene in M-CO₂ batteries.

Defective carbon

Topological defects in carbon structures, such as pentagon and heptagon rings, create highly active sites for CO₂RR in Li-CO₂ batteries. Ye *et al.* developed defect-rich graphene (TDG) by removing nitrogen dopants from N-doped graphene at high temperatures, introducing topological defects [Figure 3C]^[67]. TDG-1000 exhibited a high surface area (410 m² g⁻¹) and porous structure, enhancing electrolyte and CO₂ contact, Li-ion transport, and product absorption. As an air cathode, TDG-1000 achieved a discharge capacity exceeding 69,000 mAh g⁻¹ at 0.5 A g⁻¹, low overpotential (1.87 V) at 2 A g⁻¹, and stable cycling for 600 cycles at 1 A g⁻¹ [Figure 3D and E]. Both DFT and experiments highlighted that topological defects outperform conventional N-doped sites in Li-CO₂ batteries^[67].

Heteroatom doped carbon

Doping involves introducing non-metal atoms or different metal ions into the carbon-based structure to modify its properties, including structure, electrochemical behavior, and electrocatalytic activity. Heteroatom doping adjusts the Fermi level and electronic structure of the carbon-based material, influencing adsorption energy, adsorption modes, and material interface properties, thereby enhancing gas reduction kinetics. Additionally, heteroatom doping stabilizes the structure, creates abundant active sites, and improves the inherent conductivity of the carbon-based material, which facilitates the decomposition of M-CO₂ discharge products. Consequently, doping with heteroatoms such as B, N, S, P, and F is a proven strategy for enhancing the catalytic activity of carbon-based material in M-CO₂ batteries.

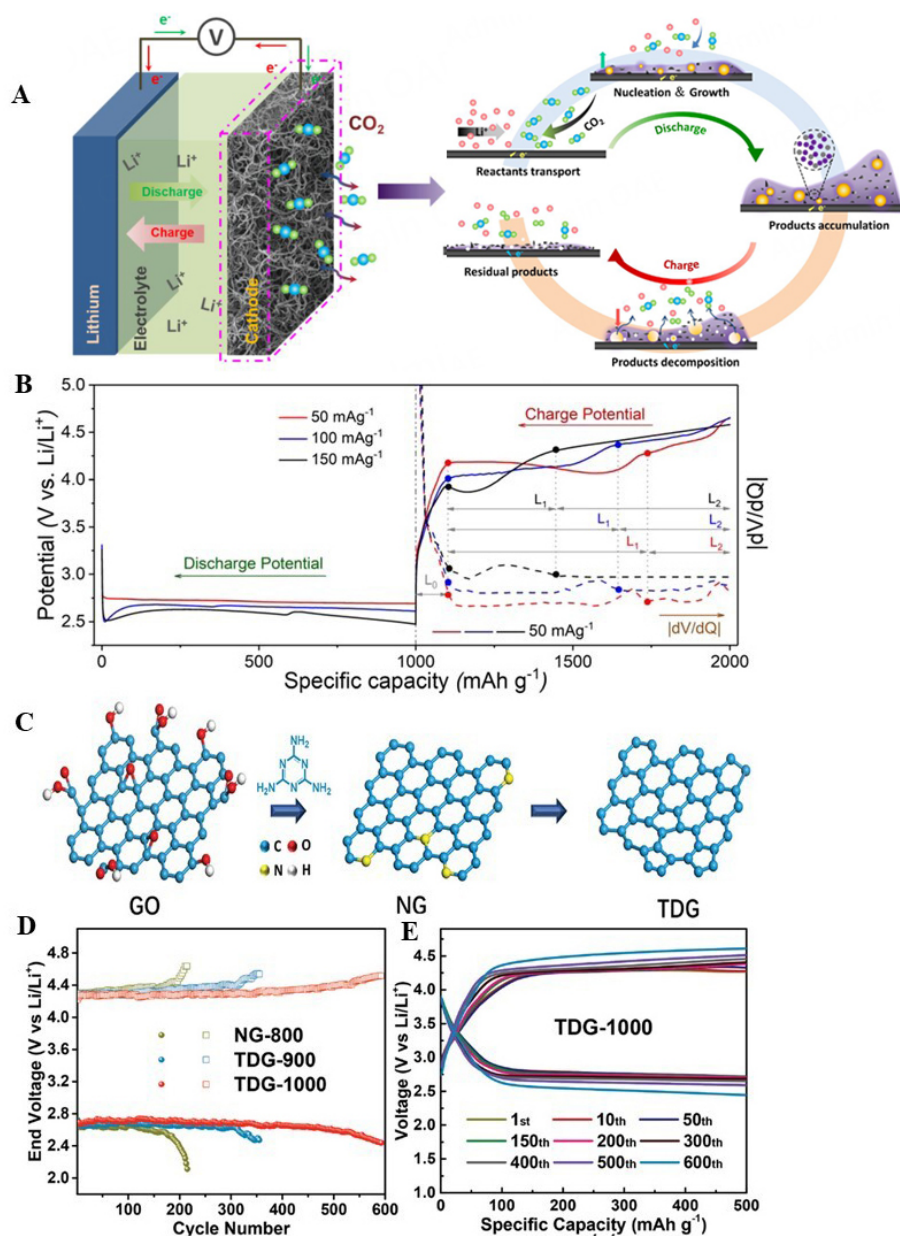


Figure 3. (A and B) Discharge/charge mechanisms and performances of Li-CO₂ batteries with CNT electrodes^[61]. (C) Preparation of NG and TDG and (D and E) long-term cycling performances and stability^[67]. These figures are reproduced with permission from American Chemical Society^[61] and Wiley^[67], respectively.

N-doped carbon

Nitrogen doping significantly enhances the CO₂RR in carbon-based materials by disrupting charge neutrality and redistributing electrons, imparting unique catalytic properties. The synthesis methods and precursor materials play a pivotal role in modulating heteroatom dopants. Common approaches for N-doped carbon catalysts include: (i) etching carbon with nitrogen sources such as NH₃; (ii) pyrolytic conversion of nitrogen-rich precursors such as MOFs or polymers; and (iii) *in situ* growth from nitrogen-containing gaseous species^[68-70]. Among nitrogen configurations, pyridinic-N and pyrrolic-N exhibit superior catalytic performance in CO₂RR and CO₂ER compared to graphitic-N, due to stronger interactions with CO₂ and Li that enhance adsorption and reaction kinetics^[71-73]. These active nitrogen species contribute

to electronic affinity, wettability, and catalytic activity, while graphitic-N improves electrical conductivity, facilitating charge transport^[74]. Thus, optimizing catalysts with a high content of pyridinic- and pyrrolic-N alongside good conductivity is essential for enhanced performance^[75]. Chen *et al.* activated nitrogen (N)-doped graphene with CO₂, achieving 72.65% pyridinic- and pyrrolic-N content, which enabled reversible Li₂CO₃ transformation in Li-CO₂ batteries^[76]. This resulted in a low voltage gap of 2.13 V at 1,200 mA g⁻¹ and excellent cycling stability over 170 cycles at 500 mA g⁻¹. Additionally, K-CO₂ batteries with N-doped CNT (NCNT) cathodes and KF-rich solid electrolyte interphase (SEI)-protected K anodes achieved 9,436 mAh g⁻¹, capacity, a 0.81 V overpotential gap at 50 mA g⁻¹, and 450 cycles. Fiber-shaped batteries also demonstrated stable power output under bending conditions, highlighting their flexibility and potential for next-generation energy storage^[77]. These advancements emphasize nitrogen doping as a crucial strategy for optimizing M-CO₂ battery performance.

N, S co-doped carbon

Carbon and sulfur (S), both p-block elements, share similar electronic structures with p-orbital outer electrons. However, the higher electronegativity of sulfur enables stronger electron attraction and interactions with atoms and functional groups, making S-doped carbon nanomaterials highly attractive for catalysis. Sulfur also acts as a bridge, inducing coupling effects that form robust carbon-based composites^[78-80]. Recently, introducing N and S heteroatoms in carbon catalysts has emerged as a strategy to modulate p-band centers, optimize orbital hybridization, and accelerate CO₂RR and CO₂ER kinetics^[81-84]. Wang *et al.* demonstrated the vertical growth of an N, S co-doped porous carbon (NS-PC) network on self-supporting carbon paper using salt-template and vacuum-sealed sulfurization methods, guided by DFT calculations^[85]. This two-step synthesis reduced formation energy and yielded a high concentration of dopants [Figure 4A and B]. The study revealed that the NS-PC exhibited a moderate p-band center, resulting in superior CO₂RR and CO₂ER kinetics. The N, S co-doped graphene (NS-G) catalyst, with its moderate p-band center, exhibited the highest CO₂ER catalytic activity [Figure 4C]. Weak adsorption causes premature Na₂CO₃ desorption, while overly strong adsorption increases the energy barrier for Na-O bond dissociation [Figure 4D]. This balance enabled highly reversible formation and decomposition of Na₂CO₃ and carbon, resulting in excellent cycling stability over 1,000 h at 10 μA cm⁻², a low 1.04 V voltage gap, and 71.2% energy efficiency. Moreover, N, S-doped CNTs combined with flexible gel electrolytes have demonstrated superior performance in quasi-solid, flexible planar Li-CO₂ batteries. These systems omit traditional separators, enhancing the reversible formation and decomposition of Na₂CO₃ for improved efficiency and stability^[86].

N, B co-doped carbon

Additionally, N and boron (B) co-doped porous graphene has been investigated as a bifunctional metal-free cathode catalyst for rechargeable Li-CO₂ batteries^[87]. For example, Qie *et al.* developed a novel B, N co-doped porous graphene (BN-hG), which serves as an efficient bifunctional cathode catalyst for rechargeable Li-CO₂ batteries^[87]. Due to the unique porous nanostructure and high catalytic activity of the BN-hG catalyst, the newly developed Li-CO₂ battery exhibits low polarization, excellent rate performance, and exceptional long-term cycling stability, achieving 200 cycles at a current density of 1.0 A g⁻¹. Recently, Kaur *et al.* developed a tubular B, N co-doped nanotubes-like material (C-BN@600), as shown in Figure 4E and F, from an IL and MOF composite^[88]. The C-BN@600 catalyst efficiently reduces CO₂ to methanol with a 74% Faradaic efficiency and a production rate of 2,665 μg h⁻¹ mg⁻¹, driven by the synergistic effects of B and N co-doping that enhance catalytic sites. DFT calculations revealed that B and N play distinct roles: B acts as an electron-withdrawing site, and N as an electron-donating site, together improving the adsorption of intermediates necessary for methanol formation, Figure 4G^[88]. When used in a Zn-CO₂ battery, the system delivered a power density of 5.42 mW cm⁻² and an energy density of 330 Wh kg⁻¹, while operating stably for over 12 days and 800 cycles [Figure 4H].

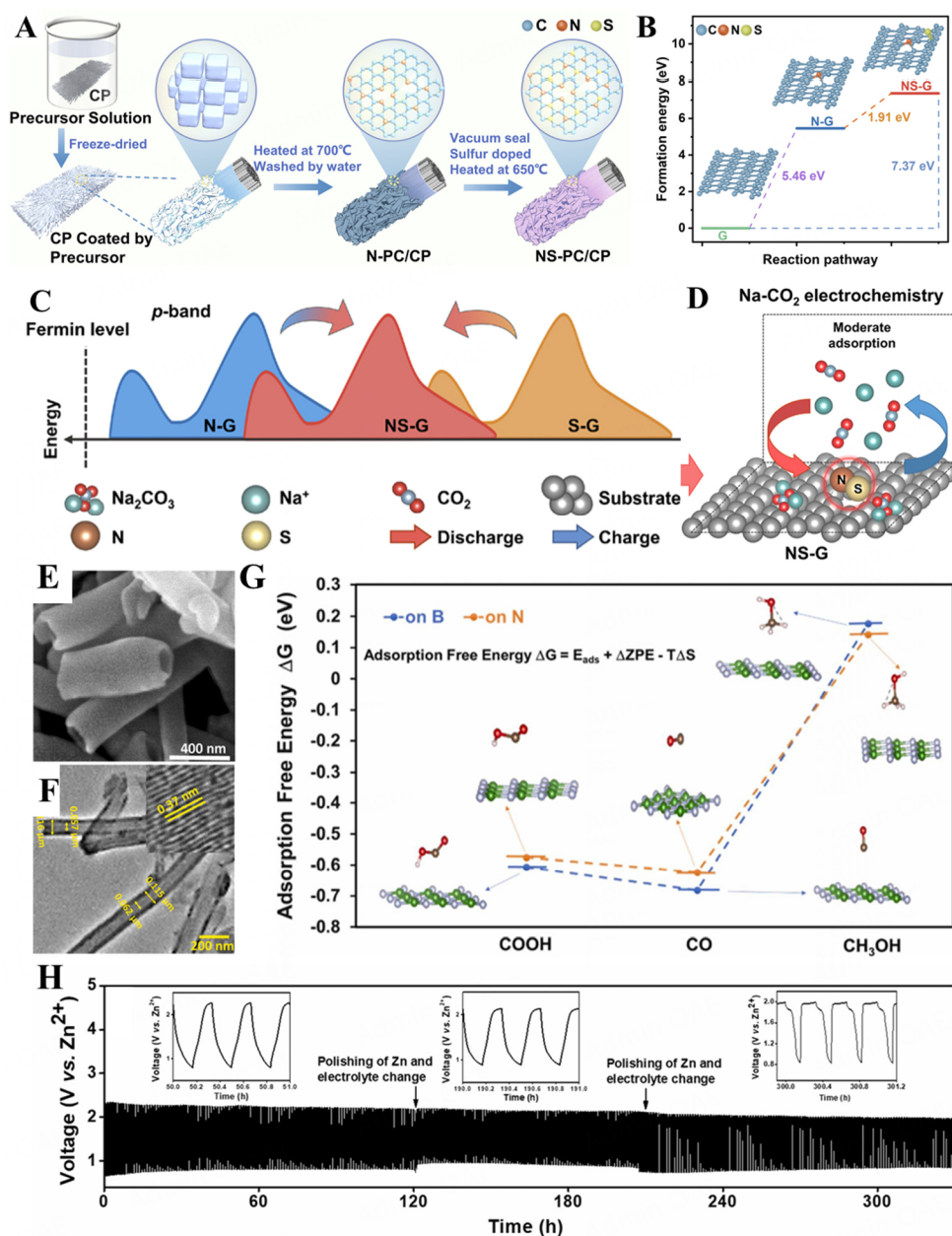


Figure 4. (A) Synthesis of NS-PC/CP, (B) Formation energies of NS-G. (C) Illustration of p-band center design and (D) the CO₂RR kinetics on NS-G catalysts^[85]. (E) FE-SEM and (F) HR-TEM images of C-BN@600. (G) Reaction coordinates for CO₂ reduction on BN^[88]. (H) Electrochemical performance. These figures are reproduced with permission from Elsevier^[85,88].

Transition metal/carbon composites

Based on recent advancements in M-CO₂ batteries, transition metal/carbon composites, encompassing both noble and non-noble metal/carbon composites, have emerged as promising catalysts, offering significant improvements in performance and cost-efficiency. Noble metal/carbon composites, such as Pt/C or Pd/C, benefit from the exceptional catalytic activity and stability of noble metals, while the carbon matrix maximizes the exposure of active sites, enhancing efficiency^[18,89,90]. The high cost of noble metals necessitates

alternatives such as non-noble metal/carbon composites (e.g., Fe, Co, and Ni), which offer comparable catalytic performance, better stability under harsh conditions, and lower costs. These materials leverage tunable electronic structures and strong metal-carbon interactions. However, challenges remain in enhancing non-noble metal efficiency, developing scalable synthesis methods, and ensuring long-term stability. Addressing these issues is vital for advancing transition metal/carbon composites in M-CO₂ battery technologies.

Noble metal/carbon composites

Ru catalysts offer a cost advantage among platinum group metals, being the least expensive. Their highly tunable electronic structure and unfilled 4d⁷ orbitals enable superior electrocatalytic activity for CO₂RR^[40,91,92]. For instance, Qiao *et al.* demonstrated that in Li-CO₂ batteries, Ru enables both Li₂CO₃ and carbon to undergo reversible decomposition ($2\text{Li}_2\text{CO}_3 + \text{C} \rightarrow 2\text{CO}_2 + 4\text{Li}^+ + 4\text{e}^-$; $E^\circ = 2.8 \text{ V vs. Li/Li}^+$), overcoming the 3.82 V thermodynamic limit observed with Au catalysts ($2\text{Li}_2\text{CO}_3 \rightarrow 2\text{CO}_2 + \text{O}_2 + 4\text{Li}^+ + 4\text{e}^-$, $E^\circ = 3.82 \text{ V vs. Li/Li}^+$), where carbon is nearly impossible to decompose^[93]. This is attributed to the catalytic activity of specific Ru crystal facets, which lower reaction barriers and promote reversibility. While the precise catalytic mechanisms of Ru remain partially unclear, its superior performance and economic feasibility have made it a focus of research. Ru AC + SA@NCB integrates Ru clusters (Ru AC) and Ru-N_x sites, improving carbon desorption, preventing poisoning, and accelerating discharge voltage^[94]. This synergy reduces energy barriers, achieving high rates with low overpotentials. In M-CO₂ batteries, Ru catalysts are often combined with or supported by carbon materials^[7,95,96]. Atomically dispersed Ru anchored on Co₃O₄ nanosheets supported by carbon cloth (SA Ru-Co₃O₄/CC) forms bifunctional catalysts^[97]. This combination provides high electrical conductivity, large surface areas, and well-structured porosity, which are critical for enhancing CO₂ diffusion, ion transport, and uniform Ru distribution. These properties maximize catalytic efficiency while functional groups on the carbon surface interact with Ru, further stabilizing active catalytic sites and enhancing the catalyst's performance. In addition to the excellent catalytic activity of metallic Ru, its oxide, RuO₂, also demonstrates significant catalytic capabilities for reversible M-CO₂ batteries. In our previous work, as depicted in [Figure 5A](#), we developed RuO₂ nanoparticles coated on carbon paper (RuCP) for Na-CO₂ batteries^[98]. Hydrous RuO_xN_yS_z was deposited onto pretreated carbon paper and annealed in nitrogen, producing highly dispersed RuO₂ nanoparticles within a 3D carbon matrix. This design enhances structural stability, promotes sodium nucleation, and improves CO₂RR and CO₂ER kinetics^[98].

Other noble metals, such as Pt, Ir, Au, and Pd, along with their oxides, have also shown promise as effective bifunctional catalysts for M-CO₂ batteries, warranting further exploration^[99-101]. Chen *et al.* advanced this further by engineering a porous Pt-(111)@CC catalyst using electro-joule heating. Its (111) crystal orientation improved CO₂ conversion kinetics and catalytic activity [[Figure 5B and C](#)]^[99]. The catalyst exhibited an exceptionally low overpotential (0.45 V) and high stability over 200 cycles, doubling area-specific capacity. A stacked Li-CO₂ pouch cell incorporating this catalyst demonstrated stable cycling at 20 μA cm⁻² with a low overpotential (0.5 V) and 82.2% energy efficiency, as shown in [Figure 5D-F](#). Compared to prior designs, this porous Pt-based cell showcased superior stability and scalability, highlighting its potential for large-scale applications^[99]. A Pt-N-doped polypyrrole (NPPy)-CNT composite by synthesizing NCNTs (NPPy-CNTs) and depositing Pt nanoparticles via magnetron sputtering was developed by Chen *et al.*^[102]. This cathode catalyst delivered a high discharge capacity of 29,614 mAh g⁻¹ with a low overpotential (0.75 V) and stable cycling over 30 cycles. N-doping enhanced conductivity and catalytic activity by creating defects and active sites, while uniform Pt dispersion facilitated Li₂CO₃ decomposition during charging, boosting performance and reversibility^[102]. Zhang *et al.* designed a Pt/FeNC catalyst for Li-CO₂/O₂ batteries by dispersing 2.4 nm Pt nanoparticles onto porous FeNC microcubic supports^[103]. This

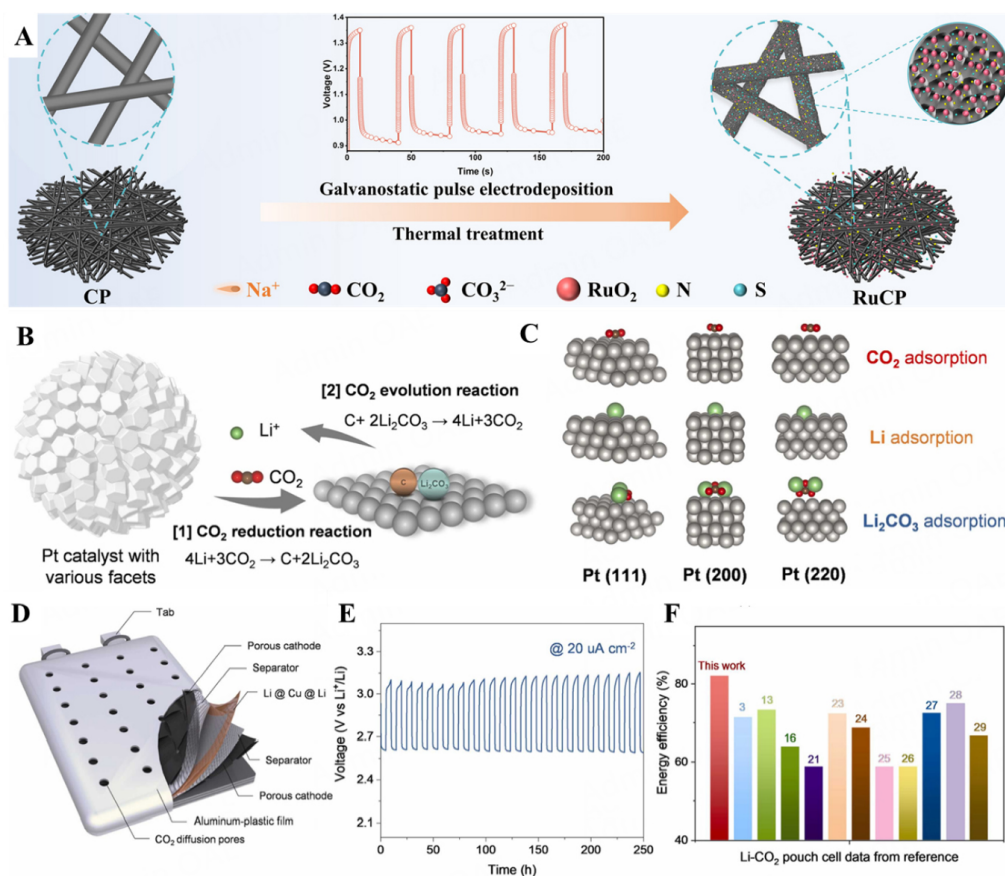


Figure 5. (A) Fabrication of RuCP^[98]. (B) CO₂ conversion in Pt-based Li-CO₂ batteries. (C) Adsorption behavior of CO₂, Li and Li₂CO₃ on different orientations of Pt surface^[99]. (D) Stacked Li-CO₂ pouch cell structure. (E) Cycling performance. (F) Energy efficiency comparison with other work^[99]. These figures are reproduced with permission from John Wiley & Sons Australia^[98] and Elsevier^[99], respectively.

catalyst achieved a low overpotential (0.54 V) and stable cycling over 142 cycles, outperforming FeNC and NC catalysts. The synergy between FeNC supports and Pt active sites constrained Li₂CO₃ particle size, reducing decomposition potential and enhancing performance^[103].

Iridium (Ir) and its oxides have shown excellent catalytic performance in CO₂RR, reducing overpotentials, increasing discharge capacity, and influencing discharge product formation. For example, Xing *et al.* synthesized ultrathin Ir nanosheets supported on N-doped carbon nanofibers (Ir NSs-CNFs), which stabilized intermediate products such as Li₂C₂O₄ during discharge^[104]. This stabilization delayed the formation of plate-like Li₂CO₃, enabling easier decomposition of discharge products at lower charging voltages and improving the cycling performance of Li-CO₂ batteries. IrO₂, in particular, is highly stable and active in oxygen evolution and CO₂-related reactions. IrO₂/carbon fibers derived from kapok (IrO₂/CK) were developed using simple carbonization and hydrothermal processes^[105]. These cathodes demonstrated excellent CO₂ catalytic activity, enhanced electron transfer, and efficient space for Li₂CO₃ deposition, enabling Li-CO₂ batteries to operate for over 4,000 h. Another example is NCNTs modified with ultrafine IrO₂ nanoparticles (IrO₂-N/CNT)^[106]. Batteries using these cathodes achieved high capacities (4,634 mAh g⁻¹), low overpotentials (3.95/1.34 V), and extended cycling lives (over 2,500 h or 316 cycles). These studies highlight the potential of IrO₂ as a key material for advancing efficient and long-lasting Li-CO₂ batteries. Beyond Ir, noble metals such as Pd have shown promise, with Pd-coated nanoporous gold

(NPG@Pd) cathodes for Al-CO₂ batteries achieving high energy efficiency (up to 87.7%) and small discharge-charge voltage gaps (as low as 0.091 V)^[107]. Similarly, PdO has shown potential in enhancing CO₂ reduction efficiencies, particularly in hybrid or composite configurations^[108].

Non-noble metal/carbon composites

Non-noble metals such as Ni, Co, Mo, Cu, Fe, and Mn, along with their composites, are among the most promising catalysts due to their abundance, cost-effectiveness, environmental benefits, and, importantly, their tunable structures and multivalence states. These metals can form a wide variety of oxides, sulfides, nitrides, and phosphides with distinct crystal structures, leading to excellent electrochemical catalytic performance. Numerous nanomaterials, including nanoparticles, nanosheets, and nanorods, have been developed for catalysts such as MoS₂, Mo₂C, WSe₂, and VS₂, all of which demonstrate high catalytic activity^[109,110].

A major challenge in using MoS₂ as a CO₂RR catalyst is its highest catalytic activity being confined to edge sites, coupled with limited electron transfer due to its semiconducting nature. To address this, heterostructures generating localized electric fields can enhance intermediate interactions and overall performance. Naik *et al.* developed a NiFe₂O₄/MoS₂/MWCNTs heterostructure using a two-step hydrothermal process for Li-CO₂ batteries under simulated Martian conditions^[111]. NiFe₂O₄, a p-type inverse spinel oxide, forms heterojunctions with MoS₂, modifying its electronic properties and enhancing CO₂RR through interfacial interactions. MWCNTs ensure efficient electron flow, reducing charge transfer resistance. This cathode achieved a high discharge capacity of 26,533.5 mAh g⁻¹ with strong cycling stability, demonstrating its potential for high-performance M-CO₂ batteries, as shown in [Figure 6A](#) and [B](#)^[111]. Peng *et al.* synthesized Cu₃P/C nanocomposites through phosphatization of a copper-based MOF precursor for use in Zn-CO₂ batteries, achieving a strong performance with an open-circuit voltage of 1.5 V^[112].

Transition metal catalysts coordinated with nitrogen on carbon supports, particularly those involving Fe, Co, Ni, Cu, and Bi, have shown great promise for CO₂ reduction in M-CO₂ batteries. For example, ultrathin Cu-N₂/GN nanosheets achieved a peak power density of 0.6 mW cm⁻² in Zn-CO₂ batteries, while Fe₁NC/S₁-1000 catalysts with atomic Fe-N₃ sites reached 0.5 mW cm⁻²^[113]. However, low power densities, with CO as the primary CO₂RR product, remain a challenge. To address this, Bi clusters (BiC/HCS) on hollow carbon spheres achieved a peak power density of 7.2 mW cm⁻², exceeding 200 recharge cycles and demonstrating 68.9% energy efficiency for formate production, paving the way for practical advancements^[110]. Carbon-based transition metal oxides and sulfides in composite materials offer excellent catalytic properties, making them ideal for M-CO₂ battery applications. Liu *et al.* demonstrated the pivotal role of octahedral Co sites in Co₃O₄, where low eg orbital filling enhanced CO₂-catalyst bonding, leading to a low overpotential of 1.0 V and a 600-h cycle life for Li-CO₂ batteries^[114]. Co/Co₉S₈ nanoparticles anchored on S, N co-doped porous carbon (Co/Co₉S₈@SNHC) exhibited superior catalytic activity in Na-CO₂ batteries, achieving an areal capacity of 18.9 mAh cm⁻² at 0.5 mA cm⁻²^[115]. Xu *et al.* developed Fe-Cu-N-C catalysts with synergistic Fe/Fe₃C nanocrystals, Fe-N_x, and Cu-N_x sites, achieving a low voltage gap of 0.44 V and over 1,550 cycles, showcasing excellent durability and performance^[75].

Manganese-based catalysts excel in CO₂RR and CO₂ER due to their diverse crystal structures, valence states, and electronic configurations^[116-118]. Liu *et al.* developed a 3D nanofiber framework with dual Mn active sites (MOC) supported on N-doped carbon nanofibers (MOC@NCNF) via electrospinning^[119]. They observed *in situ* electrochemical reconstruction between Mn(II) and Mn(III) during cycling [[Figure 6C](#)]. During charging, Mn(II) oxidizes to Mn(III), generating abundant Mn(III) active sites, which revert to Mn(II) during discharge. DFT calculations revealed that Mn(II) lowers energy barriers for CO₂RR intermediates,

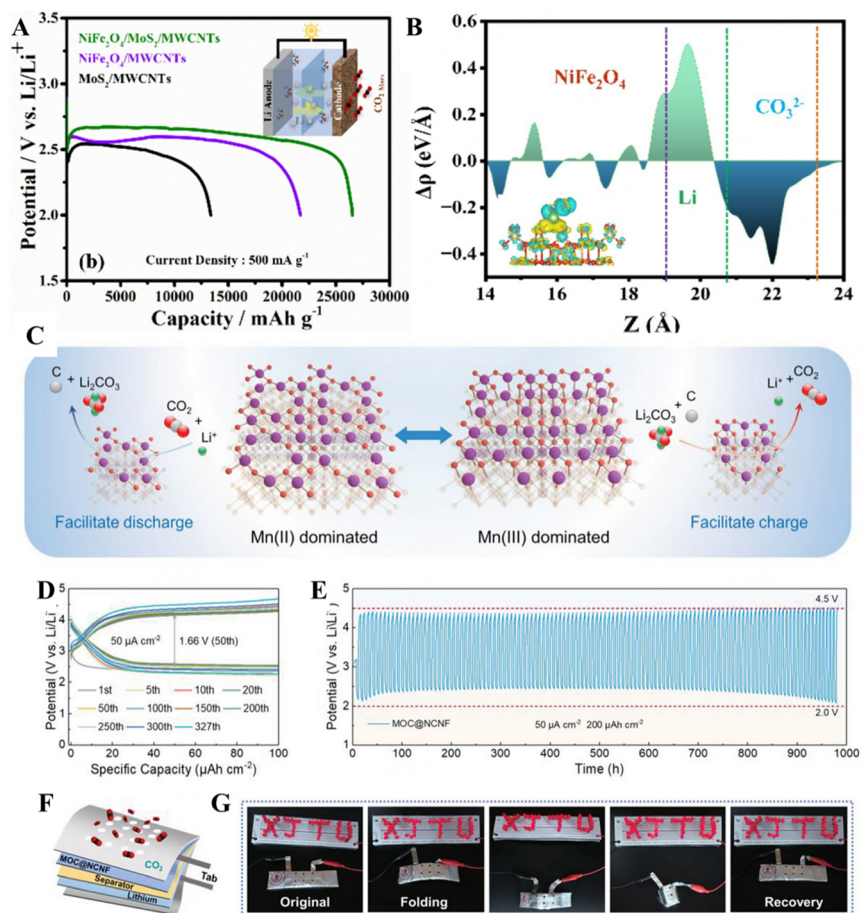


Figure 6. (A) Discharge profiles for Li- CO_2 Mars batteries. (B) 1-D charge density profile for Li_2CO_3 adsorption on the $\text{NiFe}_2\text{O}_4/\text{MoS}_2$. (C) MOC@NCNF electrochemical reconstruction mechanisms. (D) Discharge and charge profiles, and (E) Cycle performance of MOC@NCNF. (F) Schematic of flexible Li- CO_2 battery. (G) LEDs powered by flexible Li- CO_2 batteries at various bending states. These figures are reproduced with permission from Elsevier^[111] and Wiley^[119], respectively.

while Mn(III) activates $\text{Li}_2\text{C}_2\text{O}_4$ during CO_2 ER, facilitating Li-O bond cleavage. Li- CO_2 batteries with MOC@NCNF cathodes achieved high discharge capacity ($10.31 \text{ mAh cm}^{-2}$), energy efficiency (64.94%), and prolonged cycle life (327 cycles, 1,308 h) at $50 \mu\text{A cm}^{-2}$ [Figure 6D and E]. The flexible MOC@NCNF electrode also supports wearable electronics, maintaining performance under bending and folding [Figure 6F and G].

Single-atom catalysts

SACs, particularly metal-NC catalysts with single metal atoms such as Cu, Ni, Fe, Ru, and Co, have gained significant attention due to their high atomic utilization and strong interfacial electronic interactions^[120-123]. Theoretically, the metal dispersion in SACs approaches 100%, with unique metal coordination environments providing maximum atomic efficiency and distinct catalytic properties^[118,124-126].

Well-defined and uniformly dispersed structures offer significant potential for enhancing catalytic activity and selectivity in M- CO_2 batteries. Rho *et al.* demonstrated that noble metal catalysts, such as Ru and Ir, achieve lower overpotentials at the single-atom scale, highlighting improved catalytic efficiency with reduced particle size^[100]. Miao *et al.* developed a Ni- N_4 -SAC catalyst by annealing nickel nitrate and polyacrylonitrile (PAN) within a zeolitic framework, achieving Ni clusters (0.5 and 0.8 nm) alongside Ni- N_4

sites^[123]. Synergistic interactions between Ni clusters and Ni-N₄ sites enhanced CO₂ activation, as shown in [Figure 7A](#) and [B](#), achieving a peak power density of 11.7 mW cm⁻² in Zn-CO₂ batteries with stability over 1,200 cycles^[123]. A Cu-N₄ SAC (Cu/NCNF) for Li-CO₂ batteries was synthesized, which demonstrated high capacity (14,084 mAh g⁻¹), low polarization (1.29 V), and 133-cycle stability^[127]. The Cu-N₄ centers promoted CO₂ adsorption and discharge product decomposition, highlighting the impact of SACs on efficiency and durability.

The coordination environment of metal SAs (SACs) offers enhanced design flexibility compared to nanoparticles, enabling superior catalytic performance. For example, Zhu *et al.* developed a Cd-N₄ SAC (Cd SAs/NC) anchored on N-doped carbon by impregnating carbon black with cadmium nitrate and PAN, followed by calcination at 500-700 °C [[Figure 7C](#)]^[128]. X-ray absorption fine structure (XAFS) analysis confirmed atomic dispersion, with Cd in the +2 oxidation state and a Cd-N bond length of 2.24 Å [[Figure 7D](#) and [E](#)]. Cd SAs/NC-600 demonstrated outstanding CO₂RR and CO₂ER performance, achieving a record-high discharge capacity of 160,045 mAh g⁻¹ at 500 mA g⁻¹, ultra-low overpotential of 1.74 V at 2 A g⁻¹, and stable performance over 1,680 cycles, as depicted in [Figure 7F](#) and [G](#). DFT calculations revealed the atomic dispersion of Cd coordinated with four pyridinic nitrogen atoms, contributing to the catalyst's high activity^[128]. Li *et al.* optimized Pd-SACs by anchoring Pd₁-O₃C₁ sites on oxidized mesoporous carbon (Pd₁-O-CB)^[129]. This design led to excellent CO selectivity (98.7% at 0.47 V) and high current density (280.4 mA cm⁻²). Pd₁-O-CB demonstrated over 48 h of stability and a peak power density of 1.72 mW cm⁻² in a Zn-CO₂ battery.

SACs offer unique advantages, including maximized metal utilization, tunable catalytic sites, and cost-effectiveness, making them highly promising for M-CO₂ battery applications. However, challenges such as particle aggregation, dissolution, and the optimization of SAC loading on carbon carriers must be addressed. Future research should focus on enhancing SAC stability, preventing aggregation through advanced carrier designs, and elucidating the relationship between SAC loading and battery performance to unlock their full potential.

Metal-organic frameworks

MOFs, with tunable functionality and structural versatility, are widely studied for CO₂ capture, separation, and catalytic conversion. Tailored pore sizes and functional groups enhance CO₂ adsorption, while MOF nanostructures support uniform Li₂CO₃ nucleation and deposition^[130]. Li *et al.* studied various MOFs, with Mn₂(dobdc) achieving a discharge capacity of 18,022 mAh g⁻¹ at 50 mA g⁻¹, while Mn(HCOO)₂ maintaining a stable ~4 V charge potential at 200 mA g⁻¹^[131]. Additionally, ultrafine MnO nanoparticles in a N-doped carbon framework (MnO@NC-G) enabled fast electron transfer and rapid mass diffusion, achieving 25,021 mAh g⁻¹ at 50 mA g⁻¹ with stable cycling over 10 cycles^[132]. Mn(II)-based MOFs demonstrate high porosity, low charge potential, and excellent performance, even at high current densities.

In conventional Li-CO₂ systems, Li₂C₂O₄ disproportionates into Li₂CO₃, causing energy loss and voltage drops. Recent advancements using solid redox mediators (RMs), such as a Cu(II)-based MOF with benzene-1,3,5-tricarboxylate (BTC), address this issue [[Figure 8A](#) and [B](#)]^[133]. With a high surface area (1,650 m² g⁻¹) and 9 Å pore size, the MOF enhances CO₂ adsorption and catalytic activity. Paired with CNTs, the RM-based cathode achieves 3 S cm⁻¹ conductivity, reducing polarization and stabilizing Li₂C₂O₄. This increases discharge voltage (2.8 V), lowers charge voltage (3.7 V), and enables stable cycling for over 400 cycles [[Figure 8C](#) and [D](#)]. DFT calculations confirm RM(II)-BTC's strong adsorption of Li₂C₂O₄, preventing its disproportionation to Li₂CO₃.

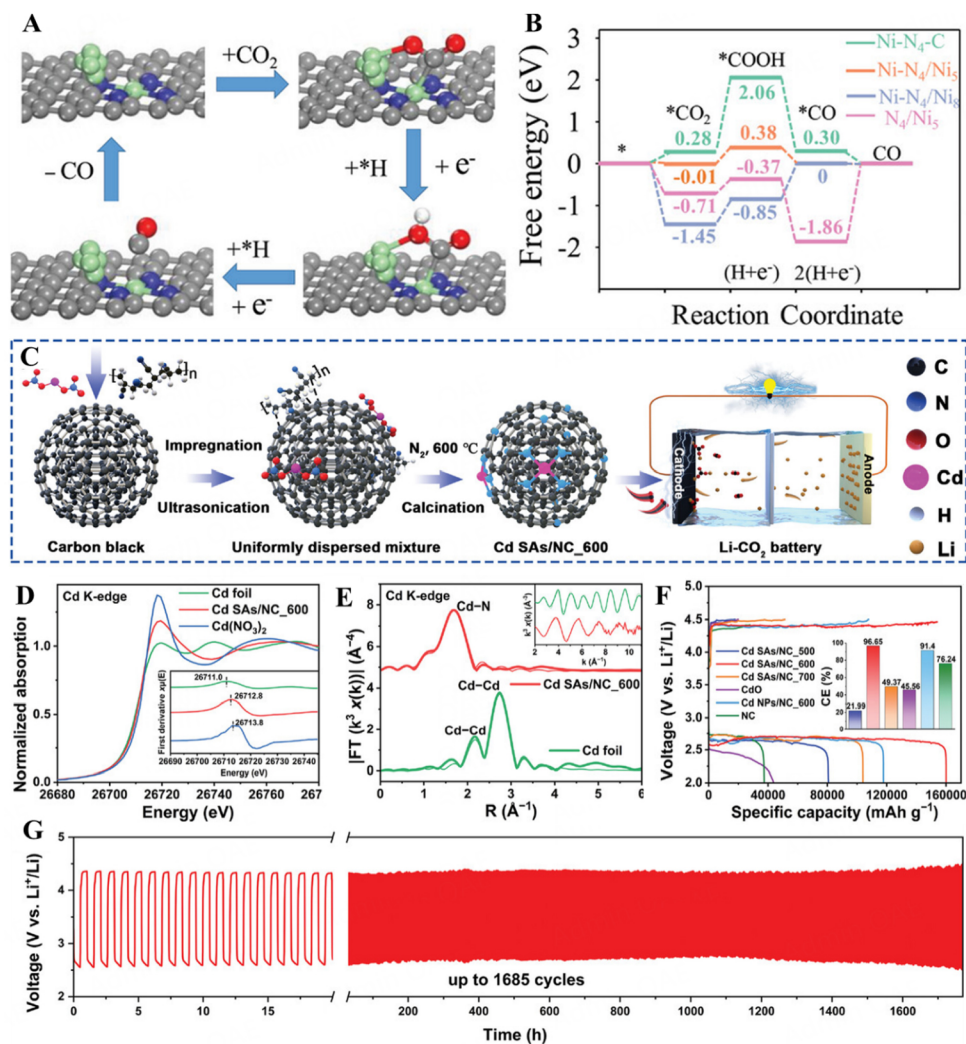


Figure 7. (A) Proposed CO₂RR pathways using Ni-N₄/Ni₅. (B) Free-energy diagram of CO₂RR to CO^[123]. (C) Preparation of Cd SAs/NC. (D) Cd K-edge XANES spectra and first derivatives (inset) of Cd foil, Cd SAs/NC-600, and Cd(NO₃)₂. (E) Cd K-edge EXAFS spectra in the R-space and k-space (inset) of Cd foil and Cd SAs/NC-600. (F) Full charge-discharge profiles. (G) Cycling performance of Cd SAs/NC-600^[128]. These figures are reproduced with permission from Wiley^[123,128].

Covalent organic frameworks (COFs), similar to MOFs, are crystalline porous materials with large surface areas, tunable structures, and ease of functionalization^[134,135]. Their thermal and chemical stability makes them ideal for gas storage, adsorption, and catalysis^[136]. Metalloporphyrin-based COFs enhance CO₂ capture, mass transfer, and ion migration, enabling precise CO₂RR critical for electrocatalysis^[136]. For example, hydrazine-linked COFs [e.g., benzene-1,3,5-tricarboxaldehyde (Tf)-2,5-dipropoxyterephthalohydrazide (DHZOPr)] improve CO₂ adsorption and Li⁺ migration at Ru/CNT cathodes, boosting Li-CO₂ battery performance^[137]. Zhang *et al.* developed TTCOF-Mn, a porphyrin-based COF catalyst featuring Mn(II) porphyrin sites, uniform microporous channels, and abundant catalytic sites [Figure 8E-G]^[138]. TTCOF-Mn achieved a low potential gap (1.07 V at 100 mA g⁻¹) and stable cycling for 180 cycles at 300 mA g⁻¹. Simulations revealed superior CO₂ adsorption (-0.02 eV free energy) and efficient four-electron CO₂ conversion of Mn-TAPP sites, corroborated by DFT analysis.

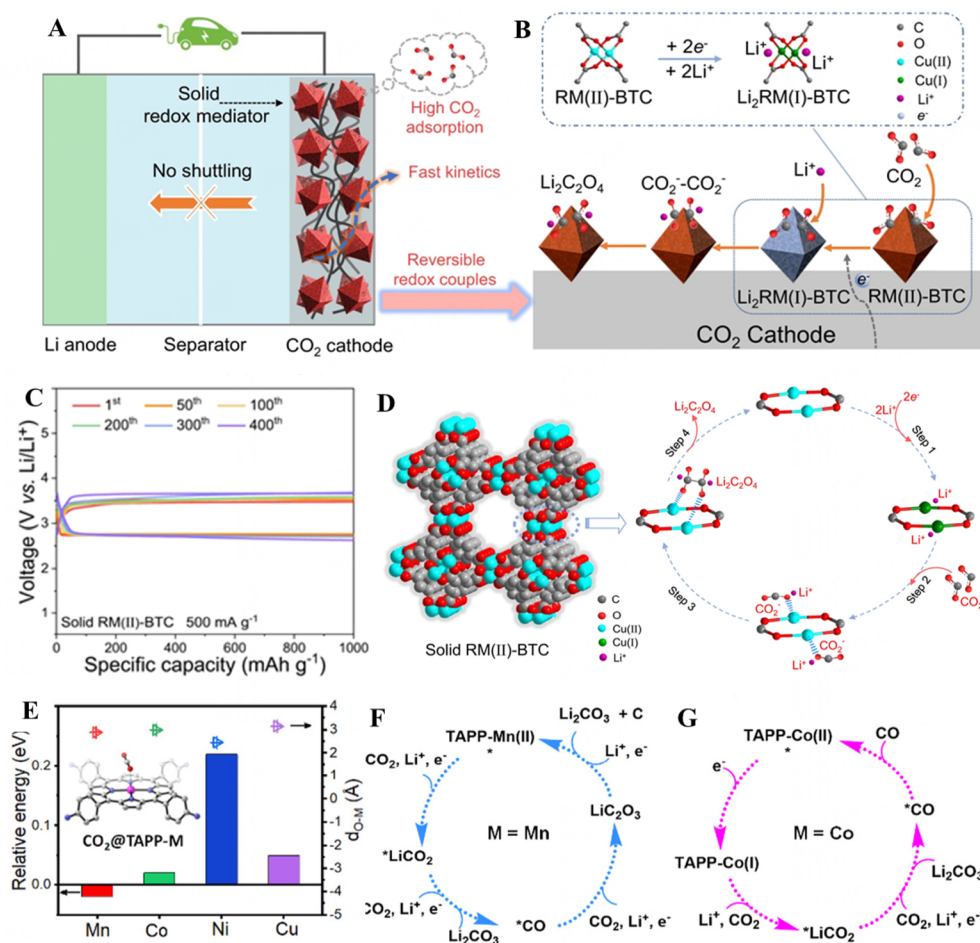


Figure 8. (A) Schematic of a Li-CO₂ battery with a solid RM(II)-BTC. (B) RM(II)-BTC reaction mechanism, showing reduction, CO₂ reaction, and Li₂C₂O₄ formation. (C) Cycling performance. (D) 3D space-filling diagram and reaction schematic for Li₂C₂O₄ formation^[133]. (E) DFT-calculation energy profiles of CO₂ adsorption on TAPP-M (M = Mn, Co, Ni, Cu) molecules. (F) Four-electron pathway at the TAPP-Mn site. (G) Two-electron pathway at the TAPP-Co site^[138]. These figures are reproduced with permission from Nature Publishing Group^[133] and American Chemical Society^[138], respectively.

Crystalline organic frameworks, including MOFs, COFs, and hydrogen-bonded organic frameworks (HOFs)^[139], feature multifunctional porous networks. HOFs, self-assembled via hydrogen bonding, are cost-effective but often lack structural stability^[140-142]. Strategies such as π - π stacking and double hydrogen bonds have improved HOF stability, enabling their use in Li-CO₂ batteries^[143-146]. Cheng *et al.* developed a high-performance HOF-based cathode by combining ultra-stable HOF-FJU-1 with Ru@CNT^[147]. HOF-FJU-1, synthesized from 3,3',6,6'-tetracyano-9,9'-bicarbazole, features π - π interactions, as shown in [Figure 9A](#) and [B](#), and high resistance to solvents, temperature, and acids/bases. This cathode delivers high capacity (24,245 mAh g⁻¹ at 100 mA g⁻¹), ultralow overpotential (1.09 V), and stable cycling over 1,800 h at 400 mA g⁻¹, even at 5 A g⁻¹, demonstrating excellent stability and promising applications for high-rate, durable Li-CO₂ batteries [[Figure 9C](#) and [D](#)].

MOFs, COFs, and HOFs exhibit unique advantages in M-CO₂ batteries due to their high tunability, large surface area, and diverse functional sites. These properties enhance CO₂ adsorption, facilitate efficient CO₂RR and CO₂ER, and support the uniform distribution of discharge products. MOFs, in particular, have demonstrated excellent electrochemical performance in terms of capacity, discharge voltage, and stability

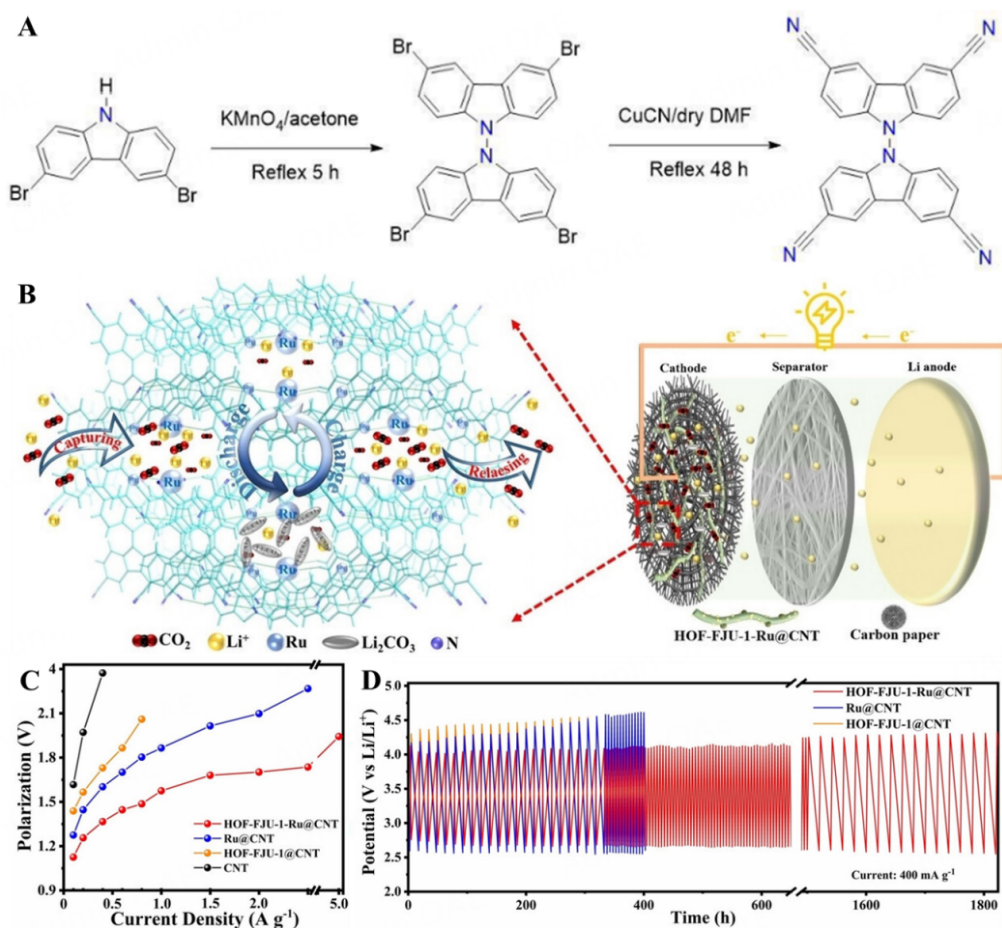


Figure 9. (A) Schematic of tetracyano-bicarbazole monomer synthesis. (B) Diagram of a $\text{Li}-\text{CO}_2$ battery. (C) Overpotentials at different current densities. (D) Battery lifespan at 400 mA g^{-1} .^[147] This figure is reproduced with permission from Wiley^[147].

when combined with materials such as Ru nanoparticles or CNTs. However, the long-term stability of MOFs, especially HOFs, under extreme electrochemical conditions remains a significant challenge. Future research should focus on improving the stability of these materials, particularly by hybridizing them with stable, conductive supports and exploring novel structural modifications to enhance their robustness. Furthermore, further investigation into optimizing pore size and improving the interaction between metal centers and CO_2 intermediates will be crucial to advancing MOF-based cathodes for high-performance, durable $\text{M}-\text{CO}_2$ batteries.

Other carbon materials (self-supporting cathodes)

Conventional $\text{M}-\text{CO}_2$ batteries face limitations due to powder-based electrodes requiring polymer binders and rigid 2D current collectors, which reduce flexibility, conductivity, and cycle stability, making them unsuitable for wearable devices^[74,115,148,149]. Additionally, traditional carbon nanomaterials often show limited catalytic activity. Freestanding electrodes address these issues by eliminating binders and rigid collectors, offering flexibility, uniform catalyst distribution, efficient electron transport, and reduced material agglomeration. These properties enhance catalytic activity, stability, and electrochemical performance, making them ideal for wearable energy storage. For instance, Xiao *et al.* developed a binder-free, freestanding N-doped 3D CNT/graphene cathode, achieving high specific capacity ($17,534.1 \text{ mAh g}^{-1}$), low overpotential (1.13 V), and excellent cycle stability over 180 cycles^[150].

Developing flexible, lightweight, and robust fibrous cathodes with high catalyst loading, efficient electron transport, and strong catalyst-current collector bonding is a critical challenge^[151]. Traditional methods often suffer from poor catalyst dispersion and weak interfacial bonding due to binder reliance, limiting performance and durability^[152]. Chen *et al.* addressed this by designing a flexible, stretchable, waterproof, and fireproof fiber-shaped Li-CO₂ battery with a “spring-like” architecture^[153]. The carbon fiber hybrid bundle (CFB)@NCNT-Mo₂N cathode was fabricated via chemical vapor deposition (CVD) of porous NCNTs on carbon fibers, followed by atomic layer deposition (ALD) of Mo₂N. Combined with a gel polymer electrolyte (GPE) and a CNT-based anode, the design achieved 14,250 Wh kg⁻¹ energy density, a 0.87 V voltage gap, and 525-cycle stability. This battery maintained performance under extreme conditions, offering a promising solution for wearable Li-CO₂ batteries in challenging environments. Investigating four-electron reaction pathways in Li-CO₂ batteries is crucial due to the slow kinetics of Li₂CO₃ formation. Hu *et al.* developed a binder-free, self-supporting Mo₂C nanowire (Mo₂C-NWs) electrode to enhance the four-electron Li₂C₂O₄ pathway^[154]. This design significantly improved cycle life, rate performance, and capacity. *In-situ* Fourier transform infrared (FTIR) spectroscopy confirmed the formation of Li₂C₂O₄ and Li₂CO₃ at varying discharge depths [Figure 10A] revealing a mechanism to regulate product formation on the Mo₂C cathode. Electrochemical testing showed the Mo₂C-NWs cathode achieved an ultra-low overpotential of 0.35 V, significantly outperforming CNT (2.60 V) cathodes [Figure 10B]. Additionally, the lower charge potential of the Mo₂C cathode (3.31 V) minimized overpotentials, improving cycle life and battery efficiency. The Mo₂C-NWs-based battery demonstrated exceptional performance, surpassing 165 cycles at 0.1 mA cm⁻² [Figure 10C]. As portable energy storage devices evolve, the high performance of liquid batteries has inspired the development of flexible batteries. A flexible Li-CO₂ battery was created using an integrated CNT/MoO₃ composite, eliminating the need for binders or metal current collectors^[155]. This battery demonstrated a high discharge capacity of 121.06 mAh cm⁻², a low charge voltage under 3.8 V, and 300 cycles at 0.25 mA cm⁻², maintaining stable performance under bending and twisting. Additionally, a Mo₂N-ZrO₂@NCNF bifunctional heterostructure catalyst was developed, enhancing CO₂ conversion and reversible Li₂C₂O₄ formation^[156]. The synthesis involves reacting ZrCl₄ with two organic ligands in a N,N-dimethylformamide (DMF)/H₂O mixture to form D-UiO-66-NH₂ [Figure 10D]. This design improves cycling stability and energy efficiency, offering a promising strategy for advanced energy storage devices.

The positive electrode of a Li-CO₂ battery typically consists of a mixed electrocatalyst, conductive carbon material, an organic binder, and a current collector^[157,158]. Traditional preparation methods are time- and energy-intensive, and the inclusion of insulating binders can increase internal resistance, thereby diminishing the performance of M-CO₂ batteries^[3,159]. In contrast, a binder-free, self-supporting cathode catalyst enhances electrode performance by integrating high electrochemical efficiency and stretchability^[153,160]. This design supports the development of advanced energy storage systems for high-performance, multifunctional wearable electronic devices.

CONCLUSION AND FUTURE PROSPECT

This review provides a comprehensive examination of the applications of various carbon-based materials and their enhancement strategies in M-CO₂ batteries. It emphasizes the intricate relationships between carbon material structure, composition, and electrochemical performance. Key findings are systematically summarized in Table 1, which includes detailed comparisons of battery types, operating atmospheres, cathode materials, electrolytes, voltage gaps (with applied current), discharge capacities (relative to current density), and cyclability (cycle life, cut-off conditions, and current density). This detailed analysis offers valuable insights into optimizing carbon-based materials for enhanced battery performance and durability. However, there are still technical challenges such as insufficient electrocatalytic activity, insufficient catalyst stability/durability, insufficient catalyst design, selection and performance optimization strategies, and

Table 1. A performance comparison of carbon-based cathode material for different Metal-CO₂ batteries

Battery type	Atmosphere	Cathode materials	Electrolyte	Voltage gap, applied current	Discharge capacity/current density	Cyclability (cycle life/cut-off condition/current density)	Ref.
Li-CO ₂	Pure CO ₂	Conductive carbon	1 M LiTFSI/[bmim]Tf ₂ N	-	2,500 mAh g ⁻¹	-	[161]
Li-CO ₂	Pure CO ₂	CNT	1 M LiTFSI/TEGDME	-	1,000 mAh g ⁻¹ 150 mA g ⁻¹	-	[61]
Li-CO ₂	Pure CO ₂	TDG	1 M LiTFSI/DMSO-0.3 M LiNO ₃	1.87 V 2 A g ⁻¹	69,000 mAh g ⁻¹ 500 mA g ⁻¹	600 cycles with a cut-off capacity of 500 mAh g ⁻¹ at 1 A g ⁻¹	[67]
Li-CO ₂	Pure CO ₂	Porous Pt@carbon cloth	1 M LiTFSI/TEGDME	0.45 V 20 μA cm ⁻²	5.81 mAh cm ⁻² 20 μA cm ⁻²	Over 200 cycles at 40 μA cm ⁻²	[99]
Li-CO ₂	Pure CO ₂	MOC@NCNF	-	1.45 V 20 μA cm ⁻²	10.31 mAh cm ⁻² 20 μA cm ⁻²	Over 327 cycles with a fixed capacity of 100 μAh cm ⁻² at 50 μA cm ⁻²	[119]
Li-CO ₂	Simulated martian atmosphere	NiFe ₂ O ₄ /MoS ₂ /MWCNTs	1 M LiTFSI/DMSO-0.3 M LiNO ₃	1.72 V 500 mA g ⁻¹	26,533.5 mAh g ⁻¹ 500 mA g ⁻¹	Over 195 cycles with a cut-off capacity of 500 mAh g ⁻¹ at 500 mA g ⁻¹	[111]
Li-CO ₂	Pure CO ₂	Cd SAs/NC	1.2 M LiTFSI/DMSO-0.3 M LiNO ₃	1.31 V 200 mA g ⁻¹	160,045 mAh g ⁻¹ 500 mA g ⁻¹	1,685 cycles with a capacity of 500 mAh g ⁻¹ at 1 A g ⁻¹	[162]
Li-CO ₂	Pure CO ₂	Solid RM(II)-BTC/CNTs	1 M LiTFSI/TEGDME	0.9 V 100 mA g ⁻¹	9,040 mAh g ⁻¹ 100 mA g ⁻¹	400 cycles with a curtailing capacity of 1,000 mAh g ⁻¹ at 500 mA g ⁻¹	[133]
Li-CO ₂	Pure CO ₂	TTCOF-Mn	1 M LiTFSI/TEGDME	1.07 V 100 mA g ⁻¹	13,018 mAh g ⁻¹ 100 mA g ⁻¹	180 cycles with a fixed capacity of 1,000 mAh g ⁻¹ at 300 mA g ⁻¹	[138]
Li-CO ₂	Pure CO ₂	HOF-FJU-1-Ru@CNT	1 M LiTFSI/TEGDME	1.09 V 100 mA g ⁻¹	24,245.3 mAh g ⁻¹ 100 mA g ⁻¹	350 cycles with a capacity of 1,000 mAh g ⁻¹ at 400 mA g ⁻¹	[147]
Li-CO ₂	Pure CO ₂	Mo ₂ C-NWs	-	0.35 V 0.05 mA cm ⁻²	8.47 mAh cm ⁻² 0.1 mA cm ⁻²	165 cycles with a limited capacity of 0.5 mAh cm ⁻² at 0.1 mA cm ⁻²	[154]
Li-CO ₂	Pure CO ₂	Mo ₂ N-ZrO ₂ @NCNF	1 M LiTFSI/TEGDME	0.32 V 10 μA cm ⁻²	5,262.2 μAh cm ⁻² 20 μA cm ⁻²	165 cycles with a limited capacity of 0.5 mAh cm ⁻² at 50 μA cm ⁻²	[156]
Na-CO ₂	Pure CO ₂	Ru@CNT	1 M NaTFSI/TEGDME	1.5 V 100 mA g ⁻¹	20,277 mAh g ⁻¹ 100 mA g ⁻¹	100 cycles with a limited capacity of 500 mAh g ⁻¹ at 100 mA g ⁻¹	[95]
Na-CO ₂	Pure CO ₂	a-MCNTs	PVDF-HFP-4% SiO ₂ /NaClO ₄ -TEGDME	0.74 V 100 mA g ⁻¹	5,000 mAh g ⁻¹ 50 mA g ⁻¹	400 cycles with a capacity of 1,000 mAh g ⁻¹ at 500 mA g ⁻¹	[163]
Na-CO ₂	CO ₂ /O ₂	Porous carbon	1 M NaTFSI/SiO ₂ -IL-TFSI/PC	2.4 V 200 mA g ⁻¹	800 mAh g _{carbon} ⁻¹ 200 mA g ⁻¹	20 cycles with a cut-off capacity of 800 mAh g _{carbon} ⁻¹ at 200 mA g ⁻¹	[164]
Na-CO ₂	Pure CO ₂	NS-PC/CP	1 M NaClO ₄ /TEGDME	1.04 V 10 uA cm ⁻²	2,422 μAh cm ⁻² 40 μA cm ⁻²	100 cycles with energy efficiencies of ≈60%	[85]
Na-CO ₂	Pure CO ₂	RuCP	1 M NaPF ₆ /DIGLYME	1.8 V 100 μA cm ⁻²	2,788 μAh cm ⁻² 100 μA cm ⁻²	Over 350 cycles at 100 μA cm ⁻²	[98]
K-CO ₂	Pure CO ₂	B-NCNTs	1.0 M KTFSI/TEGDME	0.81 V at 50 mA g ⁻¹	9,436 mAh g ⁻¹ at 200 mA g ⁻¹	450 cycles or 3,100 h with a curtailing capacity of 500 mAh g ⁻¹	[77]

K-CO ₂	Pure CO ₂	N-CNT/RGO 3D network	1.0 M KTFPI TEGDME	-	-	250 cycles or 1,500 h with a cut-off capacity of 300 mA h g ⁻¹	[165]
Al-CO ₂	Pure CO ₂	NPG@PD	AlCl ₃ /[EMIm]Cl (M:M=1:3)	0.091 V 333 mA g ⁻¹	-	Over 30 cycles at 333 mA g ⁻¹	[107]
Al-CO ₂	CO ₂ /O ₂	Ketjenblack	[EMIm]Cl/AlCl ₃ (M:M=1:2)	1.4 V 70 mA g _{Carbon} ⁻¹	13,000 mAh g _{Carbon} ⁻¹ 70 mA g _{Carbon}	50 cycles with a capacity of 200 mAh g _{Carbon} ⁻¹ at 70 mA g _{Carbon}	[166]
Zn-CO ₂	Pure CO ₂	C-BN@600	0.1 M KHCO ₃	0.78 V 1 mA cm ⁻²	5.42 mW cm ⁻² 39.4 mA cm ⁻²	Over 800 cycles at 1 mA cm ⁻²	[88]
Zn-CO ₂	Pure CO ₂	Ni-N ₄ /Ni ₅	1 M KOH	0.393 V 50 mA cm ⁻²	-	425 h at 5 mA cm ⁻²	[123]

insufficient basic understanding of the catalyst mechanism of action in the development of practically usable M-CO₂ batteries. To address the challenges faced by carbon-based catalysts and their use as hosts for metal catalysts in M-CO₂ batteries, future research can focus on the following aspects:

- (1) Enhancing electrocatalytic activity: developing carbon-based catalysts with high-density active sites through heteroatom doping, defect engineering, and edge structure modifications can significantly enhance activity. Multifunctional active sites, such as bimetallic or metal-nonmetal combinations, can improve reaction efficiency by enabling synergistic effects. Precise structural control using advanced synthesis techniques, such as tailoring hierarchical designs in graphene or CNTs, can further optimize performance.
- (2) Improving catalyst stability and durability: stability can be enhanced by designing robust carbon frameworks with high corrosion resistance and strong metal-support interactions. Protective coatings or interface engineering can prevent metal nanoparticle detachment or agglomeration during operation. Additionally, understanding degradation mechanisms, such as poisoning and structural collapse under long-term use, will allow for the development of more durable catalysts.
- (3) Catalyst design and optimization strategies: leveraging computational methods and machine learning enables the rapid prediction of optimal catalyst compositions and configurations. Tailoring metal-carbon interfaces to enhance electron transfer and catalytic synergy is critical. Multi-scale porous structures can improve reactant mass transfer and active site accessibility, leading to higher catalytic efficiency and product selectivity.
- (4) Deepening understanding of catalytic mechanisms: advanced *in situ* characterization techniques, such as X-ray absorption spectroscopy and infrared spectroelectrochemistry, allow real-time monitoring of active site evolution and reaction pathways. Identifying intermediates and activation barriers will clarify reaction mechanisms. Integration of theoretical and experimental approaches can guide the rational design of next-generation catalysts.

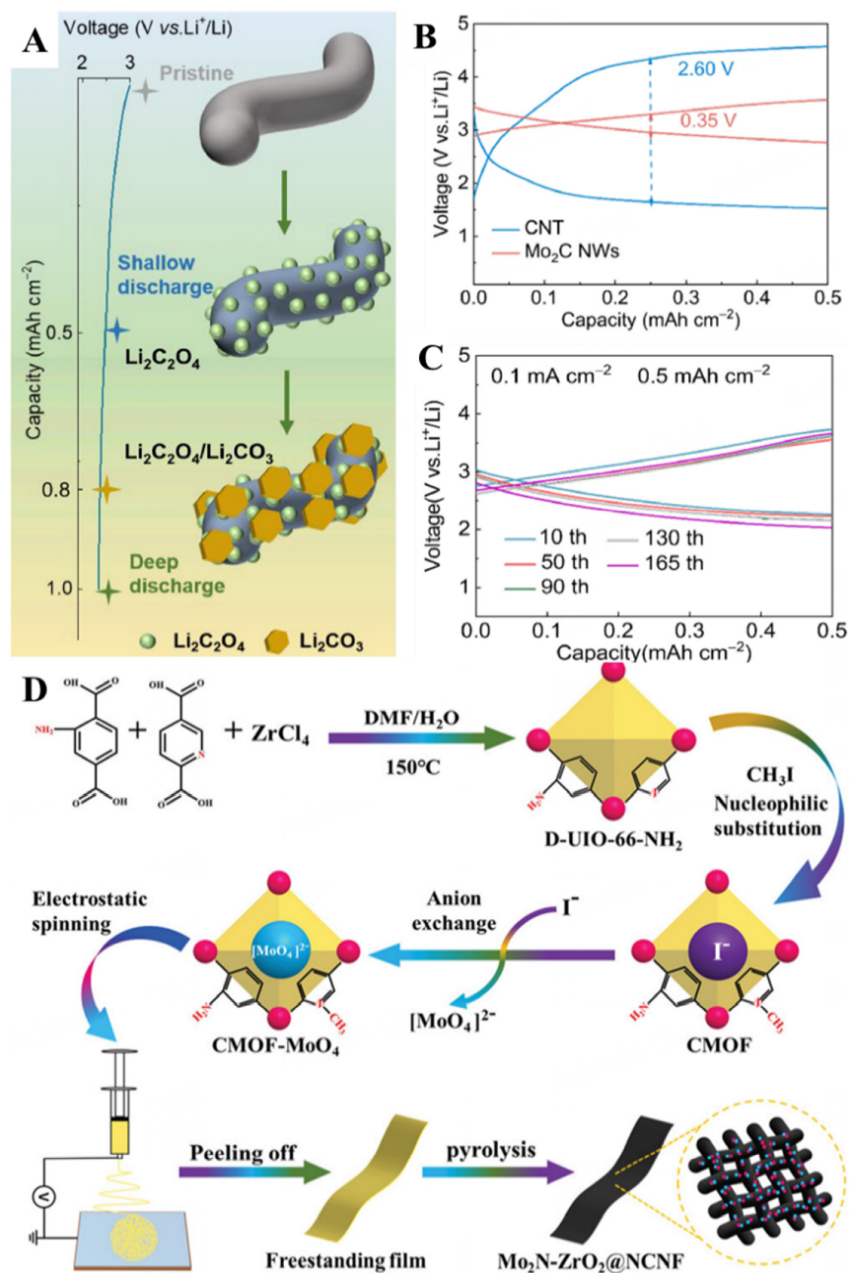


Figure 10. (A) Schematic of the discharge products in the Mo₂C-NWs cathode. (B) First cycle discharge/Charge curves of Li-CO₂ battery with CNT and Mo₂C-NWs electrodes. (C) Cycling performance of the Mo₂C-NWs electrode under CO₂. (D) Schematic of Mo₂N-ZrO₂@NCNF synthesis. These figures are reproduced with permission from Elsevier^[154] and Wiley^[156], respectively.

(5) Development of green and sustainable catalysts: replacing precious metals with earth-abundant alternatives, such as iron, cobalt, and nickel, can reduce costs. Biomass-derived carbon materials, as catalyst supports, enhance sustainability. Designing recyclable and regenerable catalysts with stable performance over multiple cycles is crucial for environmental and economic feasibility in large-scale applications.

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Authors' contributions

Conceived and wrote the manuscript: Dong, Y.; Xu, C.

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Conflicts of interest

All authors declared that there are no conflicts of interest.

Ethical approval and consent to participate

Not applicable.

Consent for publication

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